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## REVISION 1

# **A high-pressure structural transition of norsethite-type BaFe(CO<sub>3</sub>)<sub>2</sub>: comparison with BaMg(CO<sub>3</sub>)<sub>2</sub> and BaMn(CO<sub>3</sub>)<sub>2</sub>**

Chengcheng He<sup>1,2</sup>, Chaoshuai Zhao<sup>3</sup>, Jianjun Jiang<sup>1</sup>, Pan Wang<sup>1</sup>, Heping Li<sup>1\*</sup>

1 Key Laboratory of High-temperature and High-pressure Study of the Earth's  
Interior, Institute of Geochemistry, Chinese Academy of Sciences, Guiyang, 550081,  
China

2 University of Chinese Academy of Sciences, Beijing, 100049, China

3 Center for High Pressure Science and Technology Advanced Research, Beijing  
100193, China

\* Corresponding author: [liheping@mail.gyig.ac.cn](mailto:liheping@mail.gyig.ac.cn)

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## Abstract

Investigations on the phase stability of the norsethite-type family ( $\text{BaMg}(\text{CO}_3)_2$ ,  $\text{BaMn}(\text{CO}_3)_2$ ,  $\text{BaFe}(\text{CO}_3)_2$ ) under high-pressure conditions are of great significance for understanding the structure and metal cationic ( $\text{Mg}^{2+}$ ,  $\text{Fe}^{2+}$ ,  $\text{Mn}^{2+}$ ) substitution mechanism in double divalent metal carbonates. The structural evolution and equation of state of  $\text{BaFe}(\text{CO}_3)_2$  were studied firstly at high pressure up to  $\sim 7.3$  GPa by synchrotron X-ray diffraction (XRD) in diamond anvil cell (DAC) in this study.  $\text{BaFe}(\text{CO}_3)_2$  undergoes a reversible phase transition from  $R\bar{3}m$  ( $\alpha$ -phase) to  $C2/c$  ( $\gamma$ -phase) space groups at  $\sim 3.0$  GPa. The fitted elastic parameters are  $V_0 = 377.79(2) \text{ \AA}^3$  and  $K_0 = 40.3(7)$  GPa for  $\alpha$ - $\text{BaFe}(\text{CO}_3)_2$ ,  $V_0 = 483.24(5) \text{ \AA}^3$  and  $K_0 = 91.2(24)$  GPa for  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$  using second-order Birch-Murnaghan equation of state (BM2-EoS). Besides, the vibrational properties and structural stability of complete norsethite-type minerals were also investigated firstly by Raman spectroscopy combined with DAC up to 11.1 GPa. Similar structural phase transitions occur in  $\text{BaMg}(\text{CO}_3)_2$ ,  $\text{BaFe}(\text{CO}_3)_2$ ,  $\text{BaMn}(\text{CO}_3)_2$  at 2.2-2.6, 2.6-3.7, and 3.7-4.1 GPa, respectively. The onset phase transition pressures of the norsethite-type family are much lower than that of dolomite-type  $\text{Ca}(\text{Mg,Fe,Mn})(\text{CO}_3)_2$  and calcite-type  $(\text{Mg,Fe,Mn})\text{CO}_3$  carbonates. These results provide new insights into the divalent cation substitution effects on the stability and structural evolution of carbonates under high-pressure conditions.

**Keywords:** Norsethite-type minerals, synchrotron X-ray diffraction, Raman spectroscopy, phase transition, diamond anvil cell

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## Introduction

48 Norsethite ( $\text{BaMg}(\text{CO}_3)_2$ ), which can be derived from dolomite ( $\text{CaMg}(\text{CO}_3)_2$ )  
49 by exchanging Ca with Ba (Lindner et al. 2017), has been studied extensively in  
50 recent years because of its rapid precipitation at ambient conditions compared with  
51 dolomite and magnesite ( $\text{MgCO}_3$ ) (Hood et al. 1974; Lippmann 1967, 1973; Böttcher  
52 et al. 1997; Böttcher 2000; Schmidt et al. 2013; Pimentel and Pina 2014, 2016; Liu  
53 and Li 2020). Moreover, as a typical double carbonate, the formation conditions and  
54 the potential influence on the global carbon cycle are of great significance for the  
55 carbonate geochemistry (Scheetz and White 1977; Effenberger and Zemmann 1985;  
56 Böttcher et al. 1997; Schmidt et al. 2013; Effenberger et al. 2014; Pippinger et al.  
57 2014; Zhuravlev and Atuchin 2020). The end-member of the norsethite-type family  
58 includes  $\text{BaMg}(\text{CO}_3)_2$ ,  $\text{BaMn}(\text{CO}_3)_2$ , and  $\text{BaFe}(\text{CO}_3)_2$  (Liang et al. 2021; Böttcher et  
59 al. 2022). All of them are typical layer structures with octahedra ( $\text{MO}_6$ ,  $\text{M} = \text{Mg}, \text{Fe},$   
60  $\text{Mn}$ ) and polyhedra ( $\text{BaO}_{12}$ ) located exactly one above the other, parallel to the [001]  
61 direction and separated by triangular  $\text{CO}_3$  groups (Böttcher et al. 1997, 2012; Böttcher  
62 2000; Pippinger et al. 2014; Liang et al. 2019, 2021), as shown in Figure 1(a).

63 The natural samples of norsethite and Mn-bearing norsethite were found in  
64 natural environments (e.g., Costanzo et al. 2006; Zidarov et al. 2009), and the Mg and  
65 Mn end-members of norsethite ( $\text{BaMg}(\text{CO}_3)_2$  and  $\text{BaMn}(\text{CO}_3)_2$ ) were also  
66 synthesized under high pressure-temperature ( $P$ - $T$ ) conditions (e.g., Böttcher et al.  
67 2012; Lindner et al. 2017, 2018; Liang et al. 2019). However, ferroan norsethite (e.g.,  
68  $\text{BaFe}(\text{CO}_3)_2$ ) was never reported until recently successfully synthesized at high  $P$ - $T$

69 conditions (Liang et al. 2021). Norsethite-type minerals serve as a crystal chemical  
70 and geochemical analogue for the dolomite-type minerals, which are regarded as one  
71 of the most prominent deep carbon carriers (Effenberger and Zemann 1985; Böttcher  
72 2000; Binck et al. 2020; Liang et al. 2021). Although the stability of  $\text{BaMg}(\text{CO}_3)_2$ ,  
73  $\text{BaMn}(\text{CO}_3)_2$  under high  $P$ - $T$  conditions were extensively studied using different  
74 methods (e.g., XRD, Raman spectroscopy, density functional theory) (Scheetz and  
75 White 1977; Effenberger and Zemann 1985; Böttcher et al. 1997; Schmidt et al. 2013;  
76 Effenberger et al. 2014; Pippinger et al. 2014; Lindner et al. 2017; Liang et al. 2019;  
77 Zhuravlev and Atuchin 2020), whereas the high-pressure behavior of  $\text{BaFe}(\text{CO}_3)_2$  is  
78 still unclear. Synchrotron XRD, combined with DAC is a well-known technique to *in*  
79 *situ* measure the structural evolution under high-pressure conditions (Liu et al. 2016;  
80 Fu et al. 2017), by which we can get some clues for possible existence conditions of  
81 ferroan norsethite from its high-pressure stability.

82 Partial cation substitution in carbonates is very common in natural samples, and  
83 the effects of substitution with various compositions and proportions on the physical  
84 and chemical properties of carbonates were extensively studied (e.g., Reeder and  
85 Dollase 1989; Lin et al. 2012; Palaich et al. 2015; Cerantola et al. 2017; Zhao et al.  
86 2020; Gabitov et al. 2021).  $\text{Mn}^{2+}$  or  $\text{Fe}^{2+}$  cations can often be incorporated into the  
87 crystal structures in double carbonates, such as kutnahorite, ankerite, and norsethite-  
88 type  $\text{BaMn}(\text{CO}_3)_2$  (Rividi et al. 2010; Efthimiopoulos et al. 2017; Wang et al. 2022).  
89 Previous studies showed that, in the dolomite group, cation substitution of  $\text{Mg}^{2+}$  by  
90  $\text{Mn}^{2+}$  could significantly lower the onset pressure of the phase transition (Wang et al.

91 2022), but this phenomenon was not obvious in the partial substitution of  $\text{Mg}^{2+}$  by  
92  $\text{Fe}^{2+}$  (Mao et al. 2011; Merlini et al. 2012). So how about the influence of cation  
93 substitution in the norsethite-type minerals? There is still no answer yet. Given  
94 Raman spectroscopy has classically been used as a very effective and convenient tool  
95 to make semiquantitative estimates of solid solutions (e.g., Rividi et al. 2010; Farsang  
96 et al. 2018; Dong et al. 2019; Efthimiopoulos et al. 2019; Binck et al. 2020), which  
97 could be used to further study the mechanisms of  $\text{Mg}^{2+}$ ,  $\text{Fe}^{2+}$ ,  $\text{Mn}^{2+}$  replacement in the  
98 norsethite-type carbonates.

99 In this study, we firstly used *in situ* high-pressure XRD to characterize the  
100 structural changes of  $\text{BaFe}(\text{CO}_3)_2$ , calculated the isothermal compressibility, and  
101 compared it with other carbonate phases. Besides, the effects of metal cation radius on  
102 the high-pressure phase transition, Raman shifts, and pressure-induced mode shifts in  
103 the norsethite-type minerals were investigated by Raman spectroscopy. We also  
104 presented the pressure dependence and mode Grüneisen parameters of Raman-active  
105 bands for  $\text{BaFe}(\text{CO}_3)_2$ ,  $\text{BaMg}(\text{CO}_3)_2$ , and  $\text{BaMn}(\text{CO}_3)_2$ . These results provide new  
106 insights into the understanding of high-pressure behavior of the norsethite-type  
107 minerals.

108

## 109 **Materials and Methods**

110 High-purity single-crystal of the norsethite-type minerals  $\text{BaFe}(\text{CO}_3)_2$ ,  
111  $\text{BaMg}(\text{CO}_3)_2$ , and  $\text{BaMn}(\text{CO}_3)_2$  samples were synthesized at 3 GPa and 973 K for 2-  
112 12 h on a DS 6 × 600 t cubic anvil type apparatus. Norsethite-type crystals have a

113 trigonal space group of  $R\bar{3}m$ , and no superstructure reflection was observed in the X-  
114 ray images. The lattice parameters were refined to be  $a = 5.022(1) \text{ \AA}$ ,  $c = 16.752(1) \text{ \AA}$ ,  
115 and  $V = 365.85(8) \text{ \AA}^3$  for  $\text{BaMg}(\text{CO}_3)_2$ ;  $a = 5.092(1) \text{ \AA}$ ,  $c = 17.309(1) \text{ \AA}$ , and  $V =$   
116  $388.69(8) \text{ \AA}^3$  for  $\text{BaMn}(\text{CO}_3)_2$ ;  $a = 5.062(1) \text{ \AA}$ ,  $c = 17.027(1) \text{ \AA}$ , and  $V = 377.81(8) \text{ \AA}^3$   
117 for  $\text{BaFe}(\text{CO}_3)_2$  at ambient conditions (Liang et al. 2021). A more detailed synthetic  
118 method and characterization of three samples can be referred to Liang et al (2019,  
119 2021).

120 High-pressure experiments for both synchrotron XRD and Raman spectroscopy  
121 were conducted by a symmetric type of DAC equipped with a pair of 400  $\mu\text{m}$  culet-  
122 size diamond anvils. A thickness of 250  $\mu\text{m}$  rhenium gasket was pre-indented to  $\sim 60$   
123  $\mu\text{m}$ , and a diameter of around 160  $\mu\text{m}$  was drilled by a laser drilling machine as the  
124 sample chamber. A volume ratio of 4:1 methanol and ethanol mixture was selected as  
125 the pressure transmitting medium (PTM), which can provide a quasi-hydrostatic  
126 condition up to approximately 10 GPa (Klotz et al. 2009). Several tiny ruby ( $\text{Cr}^{3+}$ -  
127 doped  $\alpha\text{-Al}_2\text{O}_3$ ) spheres in Raman and gold (Au) powder in XRD experiments were  
128 loaded into the sample chamber for pressure calibration (Mao et al. 1986; Shen et al.  
129 2020), respectively. The pressures and uncertainties were calculated by the measured  
130 pressure sensor before and after data collection.

131 For high-pressure single-crystal Raman experiments, three double carbonates  
132 with good crystal morphology were loaded together into one sample chamber. Raman  
133 spectra were collected from 100 to 1200  $\text{cm}^{-1}$  on a Renishaw 2000 micro-confocal  
134 laser Raman spectrometer with a 2400 lines/mm diffraction grating in the Key  
135 Laboratory of High-Temperature and High-Pressure Laboratory Institute of  
136 Geochemistry, Chinese Academy of Sciences, Guiyang. The Raman spectrometer was  
137 calibrated using a single-crystal silicon wafer at 520  $\text{cm}^{-1}$  before experiments.

138 Samples were excited by an argon ion laser ( $\lambda = 514.5$  nm) operated at 20 mW with a  
139 focused laser spot of approximately 5  $\mu\text{m}$  in diameter through an SLM Plan 50  $\times$   
140 Olympus microscope objective, and a spectrometer with a liquid nitrogen-cooled  
141 CCD detector was used to collect the Raman data. The accurate peak positions were  
142 fitted by the *PeakFit* software.

143 For high-pressure synchrotron powder-crystal XRD experiments, dozens of  
144 colorless and transparent single crystals of neutral  $\text{BaFe}(\text{CO}_3)_2$  were picked out and  
145 ground with ethanol in an agate mortar for approximately 2 h to obtain homogeneous  
146 powder samples (Liang et al. 2021). To avoid powder scattering, powder samples  
147 were pressed slightly by two opposing diamond anvils to form a dense disk thinner  
148 than 20  $\mu\text{m}$ . Subsequently, a small piece of dense sample was loaded into the DAC  
149 sample chamber. High-pressure synchrotron XRD experiments were carried out at the  
150 BL15U1 beamline of the Shanghai Synchrotron Radiation Facility (SSRF). The  
151 incident synchrotron X-ray beam was monochromatized to a wavelength of 0.6199  $\text{\AA}$   
152 with a focused laser spot of approximately  $2 \times 3 \mu\text{m}^2$  area. The diffraction patterns of  
153 the samples were collected by a MAR-165 charge-coupled device (CCD) detector.  
154 The distance of the sample-to-detector and the geometrical parameters of the detector  
155 were calibrated with cerium dioxide ( $\text{CeO}_2$ ) powder. All 2-D diffraction patterns  
156 collected were integrated into conventional 1-D diffraction patterns by the *Fit2D*  
157 program as a function of  $2\theta$  (Hammersley et al. 1996). Diffraction data were collected  
158 at high pressures up to 7.3 GPa with intervals of 0.3-0.9 GPa. Whole XRD patterns  
159 were analyzed by the *GSAS* software package (Larson and Von Dreele 2000), and  
160 unit-cell parameters were refined by the Le Bail method (Le Bail et al. 1988).

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## Results and Discussion

## 163 **Equation of state of BaFe(CO<sub>3</sub>)<sub>2</sub> at high pressure**

164 The powder X-ray diffraction patterns of BaFe(CO<sub>3</sub>)<sub>2</sub> with increasing pressure at  
165 room temperature were shown in Figure 2 (a). All the XRD peaks of BaFe(CO<sub>3</sub>)<sub>2</sub> shift  
166 toward higher angles with increasing pressure below 2.8 GPa. Then the XRD patterns  
167 change significantly at 2.8 GPa, accompanied by the occurrence of several new peaks  
168 at ~8.5°, 9.1°, 14.3°, and 17° and the disappearance of peak at ~13°. These  
169 characteristics represent the onset phase transition of BaFe(CO<sub>3</sub>)<sub>2</sub>. Finally, the whole  
170 XRD peaks remain from 3.2 to 7.3 GPa.

171 The XRD patterns of BaFe(CO<sub>3</sub>)<sub>2</sub> were analyzed by the Le Bail refinements  
172 using the *GSAS* program. Below 2.8 GPa, the XRD patterns agree well with the phase  
173 at ambient conditions (Liang et al. 2021), which yields a trigonal structure (space  
174 group  $R\bar{3}m$ ) (Figure 2 (b)). Similar to  $\alpha$ -norsethite ( $\alpha$ -BaMg(CO<sub>3</sub>)<sub>2</sub>) (Pippinger et al.  
175 2014), the trigonal phase is referred to as  $\alpha$ -BaFe(CO<sub>3</sub>)<sub>2</sub> in the following paragraphs.  
176 Above 3.2 GPa, the powder XRD patterns of the new phase are similar to  $\gamma$ -norsethite  
177 ( $\gamma$ -BaMg(CO<sub>3</sub>)<sub>2</sub>) with all the diffraction peaks indexed to the  $\gamma$ -BaMg(CO<sub>3</sub>)<sub>2</sub>-type  
178 structure, which yields a monoclinic structure (space group *C2/c*) (Pippinger et al.  
179 2014). Two representative XRD patterns of  $\alpha$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phase at 2.0 GPa and 300  
180 K, and  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phase at 5.6 GPa and 300 K were exhibited in Figures 2 (b) and  
181 (c), respectively. The refinement results show great agreement with experimental  
182 values, which indicate the reliable structures of  $\alpha$ - and  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phases. At 2.8  
183 GPa, the existence of the characteristics of both  $\alpha$ - and  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phases is  
184 noteworthy, indicating the coexistence of these two phases, namely, the mixture of  $\alpha$   
185 +  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phase.

186 The pressure-volume (*P-V*) profiles of BaFe(CO<sub>3</sub>)<sub>2</sub> were shown in Figure 3. The  
187 *P-V* profiles of  $\alpha$ - and  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phases with the  $R\bar{3}m$  and *C2/c* structures were



188 well fitted by a BM2-EoS using the *EoSFit7c* software within the uncertainty of  
189 measurements (Birch 1947; Angel et al. 2014). The equation of BM-EoS was shown  
190 as follows:

$$191 \quad P = (3/2)K_0[(V_0/V)^{7/3} - (V_0/V)^{5/3}] \times \{1 + (3/4)(K'_0 - 4)[(V_0/V)^{2/3} - 1]\} \quad (1)$$

192 where  $V_0$ ,  $V$ ,  $K_0$ , and  $K'_0$  are the ambient-pressure unit-cell volume, high-pressure unit-  
193 cell volume, isothermal bulk modulus, and its pressure derivative ( $K'_0 = 4$ ) at ambient  
194 conditions. The fitting parameters are:  $V_0 = 377.79(2) \text{ \AA}^3$  and  $K_0 = 40.3(7) \text{ GPa}$  for  $\alpha$ -  
195  $\text{BaFe}(\text{CO}_3)_2$ ,  $V_0 = 483.24(5) \text{ \AA}^3$  and  $K_0 = 91.2(24) \text{ GPa}$  for  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$ . Compared  
196 to the norsethite-type family  $\text{BaMg}(\text{CO}_3)_2$ , the  $K_0$  of  $\alpha$ - $\text{BaFe}(\text{CO}_3)_2$  is distinctly  
197 smaller than that of  $\alpha$ - $\text{BaMg}(\text{CO}_3)_2$  ( $K_0 = 66.2 \text{ GPa}$ ) (Pippinger et al. 2014). Different  
198 from the comparable  $K_0$  values of dolomite-type carbonates, the  $K_0$  values are 95(1),  
199 92(1), and 85(6) for  $\text{CaMg}(\text{CO}_3)_2$ ,  $\text{CaMg}_{0.6}\text{Fe}_{0.4}(\text{CO}_3)_2$ , and  $\text{Ca}_{0.76}\text{Mn}_{1.24}(\text{CO}_3)_2$ ,  
200 respectively ( $K'_0 = 4$ ) (Merlini et al. 2017; Palaich et al. 2015). It may be attributed  
201 to the existence of superlattice in  $\alpha$ - $\text{BaMg}(\text{CO}_3)_2$ , which greatly improves its  
202 incompressibility (Helmersson et al. 1987; Pippinger et al. 2014; Dong et al. 2015).

203 The refined lattice parameters of  $\text{BaFe}(\text{CO}_3)_2$  at various pressures were listed in  
204 Table 1. Changes in the unit cell volumes and lattice parameters throughout our  
205 pressure range were depicted in Figures 3-4. The length of the  $a$ - and  $b$ -axes  
206 contracted by 0.03(1)  $\text{ \AA}$  while the  $c$ -axis contracted gradually by 0.57(1)  $\text{ \AA}$  up to  
207 pressures near 2.0 GPa. Above 2.8 GPa, the length of  $c$ -axis decreased by about 32%  
208 while that of  $a$ -axis increased by 72% (Figure 4), which results in the expansion of the  
209 unit cell volume by about 30% (Figure 3). For better comparison, we transform the

210 parameter values of the  $\alpha$ -phase to an equivalent monoclinic cell setting (Figure S1).  
211 All lattice parameters and volume as a function of pressure clearly show  
212 discontinuous evolution, which can be assigned to the structural phase transition from  
213  $\alpha$ -phase to  $\gamma$ -phase. In addition, the compressibility of  $\text{BaFe}(\text{CO}_3)_2$  becomes harder at  
214 high pressure, showing a nearly doubled bulk modulus  $K_0$  of  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$ . The  
215 calculated density of  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$  is also found to be higher than the original phase  
216 at ambient conditions accompanied 2% higher. Such a hardened behavior under high-  
217 pressure conditions is also found in other carbonates. The bulk moduli  $K_0$  are 92-95  
218 and 76-83 GPa for dolomite-I, -II phases, respectively (Merlini et al. 2017). The  
219 values are 112 and 175 GPa for high- and low-spin states of siderite (Ming et al.  
220 2011), 113 and 144 GPa for rhodochrosite  $\text{MnCO}_3$ -I and -II phases, (Liu et al. 2016),  
221 97 and 155 GPa for magnesite  $\text{MgCO}_3$ -I, -II phases, respectively (Maeda et al. 2017).

222 The axial compressibility and normalized lattice constants of  $\text{BaFe}(\text{CO}_3)_2$  as a  
223 function of pressure at room temperature were shown in Figure 4 (d). For  $\alpha$ -  
224  $\text{BaFe}(\text{CO}_3)_2$ , our fitted axial modules to  $a$ ,  $b$ , and  $c$  are  $K_{a0} = K_{b0} = 324(22)$  GPa and  
225  $K_{c0} = 56(2)$  GPa, which indicate a remarkable compressional anisotropy with the  
226 compressibility of the  $c$ -axis being almost six times softer than that of other axes. The  
227 great compressional anisotropy can be attributed to the rigid unit of the  $\text{CO}_3$  groups,  
228 which is a coplanar arrangement and nearly parallel to the (0001) plane direction. As  
229 shown in Figure 1, the coplanar carbonate groups are out-of-plane tilt under high-  
230 pressure conditions (Pippinger et al. 2014), which result in the difference in the  
231 symmetries of the two phases and the decrease in compressional anisotropy for  $\gamma$ -

232 BaFe(CO<sub>3</sub>)<sub>2</sub> with  $K_{a0} = 287(22)$  GPa,  $K_{b0} = 504(72)$  GPa, and  $K_{c0} = 207(16)$  GPa. The  
233  $\beta$  angle steadily increases under compression within the pressure range (Figure S1(d)).  
234 The  $a$ -axis shows a softening behavior at high pressure, and the  $b$ - and  $c$ -axes show  
235 the opposite behavior.

236

237 **Raman spectroscopy of the norsethite-type family (BaFe(CO<sub>3</sub>)<sub>2</sub>, BaMg(CO<sub>3</sub>)<sub>2</sub>,**  
238 **BaMnCO<sub>3</sub>)<sub>2</sub>) at high pressure**

239 To better compare the high-pressure behaviors of BaFe(CO<sub>3</sub>)<sub>2</sub>, BaMg(CO<sub>3</sub>)<sub>2</sub>,  
240 BaMn(CO<sub>3</sub>)<sub>2</sub>, Raman spectra were obtained up to 11.1 GPa on the crystals of the  
241 complete norsethite-type family. The expansion of the vibration was obtained through  
242 factor-group analysis for norsethite-type crystals (Scheetz and White 1977; Zhuravlev  
243 and Atuchin 2020):

$$244 \quad \tau_{tot} = 3A_{1g}(R) + 2A_{1u} + A_{2g} + 5A_{2u}(IR) + 4E_g(R) + 6E_u(IR) \quad (2)$$

245 There are 7 Raman-active modes, which contain internal vibrations of the carbonate  
246 group and external lattice modes. The vibrations of the carbonate group (~700 to 1450  
247 cm<sup>-1</sup>) include fundamental symmetric and antisymmetric stretching ( $\nu_1$ ,  $\nu_3$ ), out-of-  
248 plane bending ( $\nu_2$ ) and in-plane bending ( $\nu_4$ ), among which  $\nu_1$  is the strongest  
249 characteristic peak. In addition, the external lattice modes (< 400 cm<sup>-1</sup>), derived  
250 mainly from the translational lattice mode  $T$  and librational lattice mode  $L$ , could also  
251 be obtained (Gillet et al. 1993). Similar Raman spectra of the norsethite-type family  
252 were observed in the wavenumber range from 100 to 1200 cm<sup>-1</sup> at ambient conditions  
253 (Figures 5, S2, S4). A weak peak ( $\nu_2$ ) can be observed in the Raman spectrum of  
254 BaMg(CO<sub>3</sub>)<sub>2</sub> at 880 cm<sup>-1</sup>, which is not found in BaMn(CO<sub>3</sub>)<sub>2</sub> and BaFe(CO<sub>3</sub>)<sub>2</sub>. The  $\nu_2$   
255 internal mode belongs to the infrared-active mode, and chemical substitution affects  
256 its activity. Such a phenomenon could also be observed in the dolomite system (Gillet

257 et al. 1993; Gunasekaran et al. 2006). Compared with previous studies of related  
258 Raman-active modes for the norsethite-type family, the slight differences in  
259 wavenumber for observed modes are due to the weak intensity, peak overlap,  
260 orientation dependence, and synthetic environment (Scheetz and White 1977; Schmidt  
261 et al. 2013; Liang et al. 2019; 2021).

262 The high-pressure Raman spectra of single-crystal of  $\text{BaFe}(\text{CO}_3)_2$  were collected  
263 in the diamond window of 100-1200  $\text{cm}^{-1}$  at ambient temperatures. The pressure  
264 dependence of the Raman spectra and shifts of  $\text{BaFe}(\text{CO}_3)_2$  modes were shown in  
265 Figures 5 and 6. Obviously, with increasing pressure, all the Raman bands of  
266  $\text{BaFe}(\text{CO}_3)_2$  show blue shifts due to the decrease of corresponding bond length  
267 (Ba/Fe/C-O) caused by crystal shrinkage. When the pressures of  $\text{BaFe}(\text{CO}_3)_2$  up to 2.6  
268 GPa, the  $\nu_4$  mode splits into two individual components, then with further  
269 compression,  $T$  and  $L$  modes begin to split into two and three peaks, respectively, and  
270 no new peak occurs after 3.7 GPa. Meanwhile, the pressure-induced frequency shifts  
271 for the measurable modes all show a discontinuous evolution at 2.6 GPa, a transition  
272 area at 2.6 to 3.7 GPa, and a stable linear relation with pressure after 3.7 GPa.  
273 Referring to the XRD results of  $\text{BaFe}(\text{CO}_3)_2$  in this study, both of these significant  
274 changes could be attributed to the occurrence of structural phase transition from  $\alpha$ -  
275  $\text{BaFe}(\text{CO}_3)_2$  to  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$ . Combined the high-pressure results of XRD and  
276 Raman spectra, the more accurate phase transition range of  $\text{BaFe}(\text{CO}_3)_2$  from  $\alpha$ -phase  
277 to  $\gamma$ -phase should be 2.6 to 3.2 GPa, and  $\alpha$ -phase and  $\gamma$ -phase coexist in this pressure  
278 range. XRD results reveal that the  $\text{CO}_3$  groups are out-of-plane tile at high pressure,  
279 which results in the vibrational environment change. So the  $T$  mode splits into three  
280 peaks ( $T'$ ,  $T''$ ,  $T'''$ ), and the pressure coefficient ( $\partial\nu_i/\partial P$ ) changes from 2.03(11) to  
281 1.89(5), 6.10(6), 9.45(4)  $\text{cm}^{-1}/\text{GPa}$ , respectively. The  $L$  mode splits into two modes

282 ( $L', L''$ ) and the  $\partial v_i / \partial P$  values decrease from 8.05(31) to 5.09(9) and 7.37(7)  $\text{cm}^{-1}/\text{GPa}$ ,  
283 respectively. The  $v_4$  mode becomes two peaks ( $v'_4, v''_4$ ) with the  $\partial v_i / \partial P$  from 2.48(9)  
284  $\text{cm}^{-1}/\text{GPa}$  changing into 0.81(5) and 3.01(7)  $\text{cm}^{-1}/\text{GPa}$ , respectively. No split was  
285 observed in mode  $v_1$  but the pressure coefficient decreases obviously from 5.71(2) to  
286 2.04(3)  $\text{cm}^{-1}/\text{GPa}$  (Table 2).

287 The evolution and the pressure dependence of the Raman modes of  $\text{BaMg}(\text{CO}_3)_2$   
288 and  $\text{BaMn}(\text{CO}_3)_2$  were showed in Figures S2-S5. From Raman spectra and shifts at  
289 high pressure, similar structural phase transitions can be also observed in  
290  $\text{BaMg}(\text{CO}_3)_2$  and  $\text{BaMn}(\text{CO}_3)_2$  at 2.2-2.6 GPa and 3.7-4.1 GPa, respectively. These  
291 results are comparable to the previous works of high-pressure XRD and Raman  
292 experiments. Pippinger et al. (2014) reported that a structural transition occurred from  
293 an  $R\bar{3}m$  to  $C2/c$  structure at approximately 2.32 GPa in  $\text{BaMg}(\text{CO}_3)_2$  by the single-  
294 crystal XRD experiment. Liang et al. (2019) found a similar structural phase transition  
295 may occur in  $\text{BaMn}(\text{CO}_3)_2$  between 3.0 and 3.8 GPa by Raman scattering  
296 measurement. The slight difference of phase transition pressures may be attributed to  
297 the uncertainty of different detecting methods. When the pressures reach 2.6 and 4.1  
298 GPa for the  $\text{BaMg}(\text{CO}_3)_2$  and  $\text{BaMn}(\text{CO}_3)_2$ , respectively, the  $T$  and  $L$  modes undergo  
299 various degrees of splitting, with the values of the slope  $\partial v_i / \partial P$  for each Raman  
300 vibration changing drastically. Meanwhile, the  $v_4$  mode splits into two new modes,  
301 and the slope  $\partial v_i / \partial P$  of  $v_1$  mode exhibits an obvious decrease. The  $v_2$  mode of  
302 norsethite also shows a discontinuous change. The detailed pressure coefficients of  
303  $\text{BaMg}(\text{CO}_3)_2$ , and  $\text{BaMn}(\text{CO}_3)_2$  are listed in Tables S1 and S2, respectively.  
304 Reversibility of the phase transition of the norsethite-type minerals from  $\alpha$ -phase to  $\gamma$ -  
305 phase has been confirmed by the consistency between decompressed spectra and  
306 initial spectra. In addition, the similar phase transition seems complete faster in

307 BaMg(CO<sub>3</sub>)<sub>2</sub> and BaMn(CO<sub>3</sub>)<sub>2</sub> than in BaFe(CO<sub>3</sub>)<sub>2</sub>, with no two-phase coexistence  
308 observed in the former two.

309 We calculate the isothermal mode Grüneisen  $\gamma_{iT}$  to further understand the  
310 elasticity properties of the norsethite-type minerals at high pressure, which is  
311 calculated as follows (Gillet et al. 1989):

$$312 \quad \gamma_{iT} = \frac{K_T}{v_{i0}} \left( \frac{\partial v_i}{\partial P} \right)_T \quad (3)$$

313 where  $v_{i0}$  and  $\partial v_i / \partial P$  are the frequencies fitted linearly by  $v_i = v_{i0} + (\partial v_i / \partial P) \times P$ ,  $v_{i0}$  is  
314 the frequency of mode  $v_i$  at 0 GPa, and  $P$  is pressure.  $K_T$  is the bulk modulus with  
315 values of 40.3(7) and 91.2(24) GPa for  $\alpha$ - and  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> (this study), and 66.2(23)  
316 and 41.9(4) GPa for  $\alpha$ - and  $\gamma$ -BaMg(CO<sub>3</sub>)<sub>2</sub>, respectively (Pippinger et al. 2014). The  
317 results are listed in Tables 2 and S1. The average  $\gamma_{iT}$  values of CO<sub>3</sub> group (0.17 for  
318 BaMg(CO<sub>3</sub>)<sub>2</sub>, 0.18 for BaFe(CO<sub>3</sub>)<sub>2</sub>) are much smaller than that of the external lattice  
319 modes (1.685 for BaMg(CO<sub>3</sub>)<sub>2</sub>, 1.095 for BaFe(CO<sub>3</sub>)<sub>2</sub>) in low-pressure phase. In  $\gamma$ -  
320 phase, the difference of compressibility gets larger with a smaller  $\gamma_{iT}$  value of CO<sub>3</sub>  
321 group (0.06 for BaMg(CO<sub>3</sub>)<sub>2</sub>, 0.22 for BaFe(CO<sub>3</sub>)<sub>2</sub>) and a larger  $\gamma_{iT}$  value of the  
322 external lattice vibrations (2.09 for BaMg(CO<sub>3</sub>)<sub>2</sub>, 3.81 for BaFe(CO<sub>3</sub>)<sub>2</sub>).  $\gamma_{iT}$  values are  
323 small for the C-O bonds, which reflects their relative incompressibility and weak  
324 expansivity. Hence, the weaker bonds, corresponding to the external lattice modes at  
325 low frequency, are more affected by increasing pressure. The change in volume is  
326 mainly caused by the weaker bonds. It is obvious that the carbonate groups in  
327 BaMg(CO<sub>3</sub>)<sub>2</sub> and BaFe(CO<sub>3</sub>)<sub>2</sub> are much more incompressible than cation octahedra,  
328 which is consistent with other carbonates (e.g., MgCO<sub>3</sub> from Liang et al. 2018).

329

330

## Implications

331 Cationic substitution is an important factor to affect the evolution and stability of  
332 crystal structure, phase transition pressure, the positions and frequency shifts of  
333 spectroscopic peaks for different carbonate structures. The phase transition pressures  
334 of  $\text{BaMg}(\text{CO}_3)_2$ ,  $\text{BaFe}(\text{CO}_3)_2$ , and  $\text{BaMn}(\text{CO}_3)_2$  are 2.4(2), 2.7(5), and 3.9(2) GPa in  
335 this study, respectively. Compared with this study and previous results (Lin et al.  
336 2012; Spivak et al. 2014; Cerantola et al. 2015; Merlini et al. 2015; Palaich et al.  
337 2015; Fu et al. 2017; Merlini et al. 2017; Vennari and Williams 2018; Binck et al.  
338 2020; Chariton et al. 2020; Wang et al. 2022), the phase stability of norsethite-type  
339  $\text{Ba}(\text{Mg,Fe,Mn})(\text{CO}_3)_2$ , dolomite-type  $\text{Ca}(\text{Mg,Fe,Mn})(\text{CO}_3)_2$ , and calcite-type  
340  $(\text{Mg,Fe,Mn})\text{CO}_3$  carbonates at ambient temperature were presented in Figure 7. The  
341 effective cation radii of  $\text{Ba}^{2+}$ ,  $\text{Ca}^{2+}$ ,  $\text{Mn}^{2+}$ ,  $\text{Fe}^{2+}$ , and  $\text{Mg}^{2+}$  are 1.35, 1.00, 0.83, 0.78,  
342 and 0.72 Å at ambient conditions, respectively (Shannon and Prewitt 1969). With the  
343 addition of the norsethite-type members, it is more clear that a smaller metal cation  
344 tends to stabilize the trigonal structure to higher pressure (Wang et al. 2022), and the  
345 phase transition pressures are much lower for norsethite-type carbonates than that of  
346 calcite- and dolomite-type carbonates. A cation octahedron with a smaller cation has  
347 shorter bonds and smaller compressibility, it thus can delay the polyhedral tilting and  
348 distortion at high pressure (Ross and Reeder 1992; Santillán and Williams 2004;  
349 Pippinger et al. 2014). However, different from the linear trend reported previously  
350 (Wang et al. 2015), the relationship tends to be nonlinear in the norsethite-type  
351 minerals join. Taking a closer look at the effects of ionic radii on the phase transition  
352 of different carbonate structures, the substitution of  $\text{Mg}^{2+}$ ,  $\text{Fe}^{2+}$ , and  $\text{Mn}^{2+}$  could  
353 change the kinetic barriers of phase transition by changing the effective cation radii.  
354 The onset phase transition pressures to high-pressure phases increase with cations  
355 radii for norsethite-type minerals, while the results are opposite for both calcite- and

356 dolomite-type carbonates. We attributed this to the structure difference. In particular,  
357 the norsethite-type carbonates are composed of octahedral (Mg,Fe,Mn)O<sub>6</sub>, trigonal  
358 planar (CO<sub>3</sub>)<sup>2-</sup>, and polyhedra BaO<sub>12</sub> groups (Liang et al. 2019), while the dolomite-  
359 type carbonates are composed by alternation layers of octahedral (Mg,Fe,Mn)O<sub>6</sub>,  
360 (CO<sub>3</sub>)<sup>2-</sup>, and octahedral CaO<sub>6</sub> groups (Binck et al. 2020). The ionic radii display a  
361 significant difference between (Mg, Fe, Mn)<sup>2+</sup> (0.72-0.83 Å) and Ba<sup>2+</sup> (1.35 Å) in  
362 norsethite-type carbonates, while the values of (Mg, Fe, Mn)<sup>2+</sup> are close to Ca<sup>2+</sup> (1.00  
363 Å) in dolomite-type carbonates. Although the atomic distribution of the norsethite-  
364 type carbonates is topologically related to that of the dolomite-type minerals, the  
365 larger ratio between the radii of the Ba<sup>2+</sup>/(Mg<sup>2+</sup>, Mn<sup>2+</sup>, Fe<sup>2+</sup>) ions compared to that of  
366 Ca<sup>2+</sup>/(Mg<sup>2+</sup>, Mn<sup>2+</sup>, Fe<sup>2+</sup>) causes a distinct rotation of the carbonate group around  
367 (0001) plane (Liang et al. 2019). The difference in structure may lead to the change of  
368 high-pressure phase transition mechanism of cation (Mg<sup>2+</sup>, Mn<sup>2+</sup>, Fe<sup>2+</sup>) substitution.  
369 These results provide new insights into the divalent cation substitution effects on the  
370 stability and structural evolution of carbonates under high-pressure conditions.

371 The effect of ionic radii on phase transition pressures have been found not only in  
372 carbonate minerals, but also in other materials, such as rare-earth vanadates (Gong et  
373 al. 2018), gallium-bearing sphalerite-structure compounds (Zhang et al. 2021), and  
374 lead-free niobate ceramics (Yan et al. 2019). Consistent with the results in this study,  
375 the substance with a smaller ionic radius has a higher structural stability in  
376 isostructural materials. Once the relationship between the ionic radius and the phase  
377 transition pressure in a crystal structure is confirmed, the high-pressure structural  
378 stability of isostructural materials might be predicted. Moreover, the ionic radius can  
379 be added as the optimization variables to improve the high-pressure properties of the  
380 resistant materials (Yan et al. 2019), which may provide a possible way for high-



381 pressure material synthesis and design.

382

383

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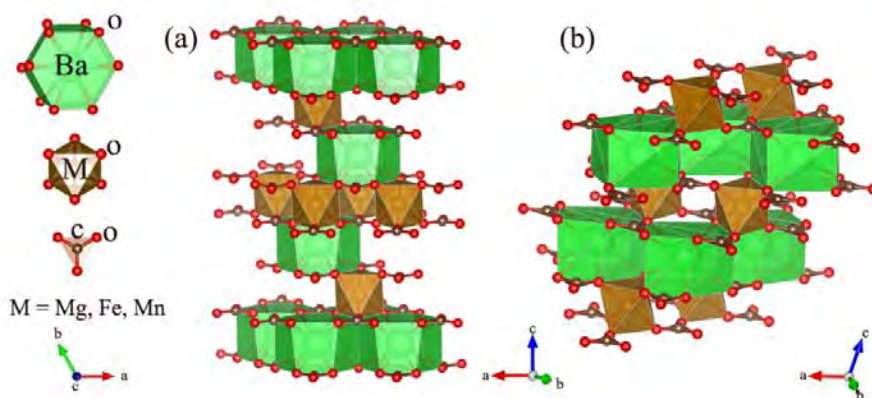
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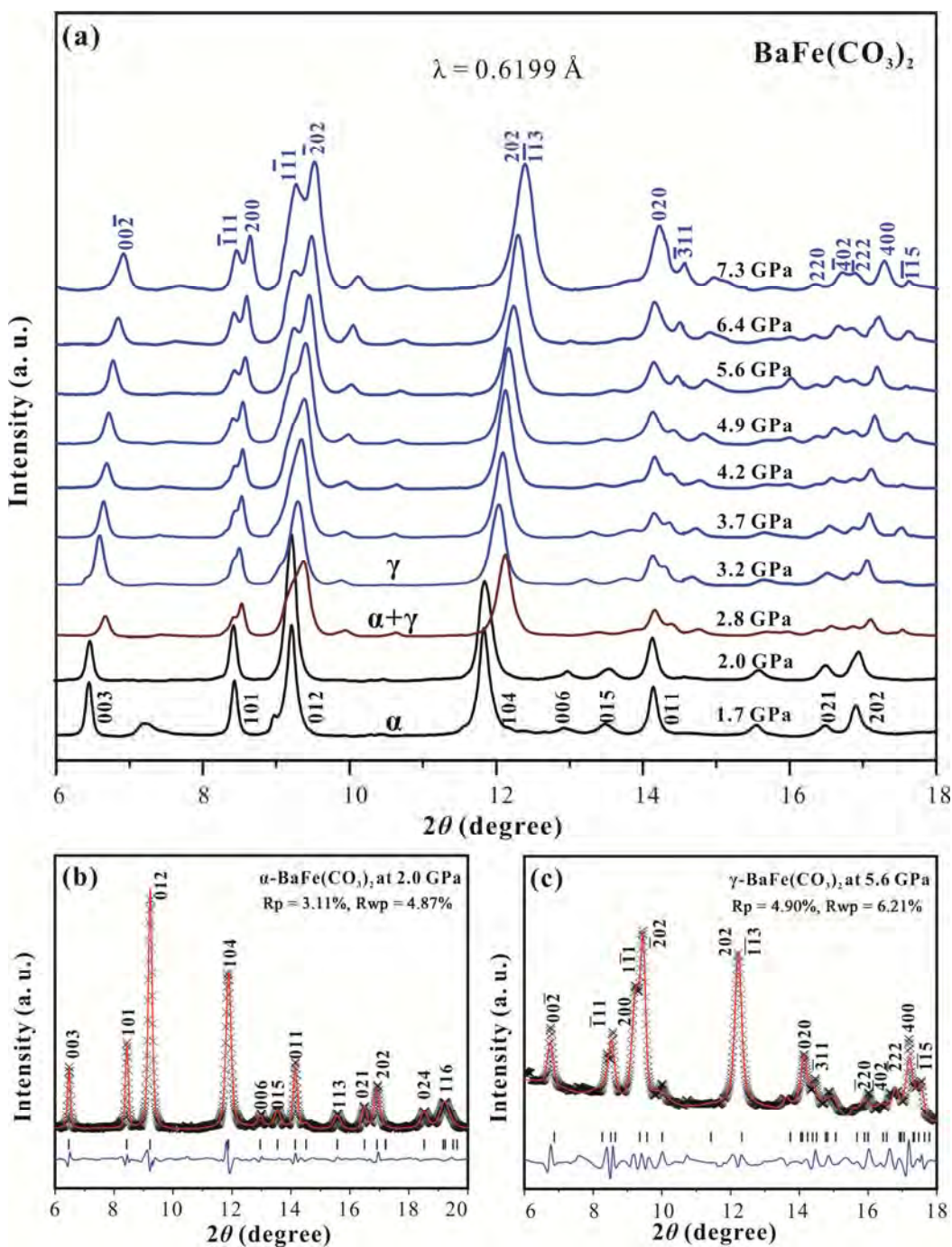
595 **Figure Captions**



596

597 **Figure 1.** Crystal structures of  $\alpha$ -BaFe(CO<sub>3</sub>)<sub>2</sub> (a) and  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub> phase (b).

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599

600 **Figure 2.** High-pressure XRD patterns of BaFe(CO<sub>3</sub>)<sub>2</sub> (a), representative XRD

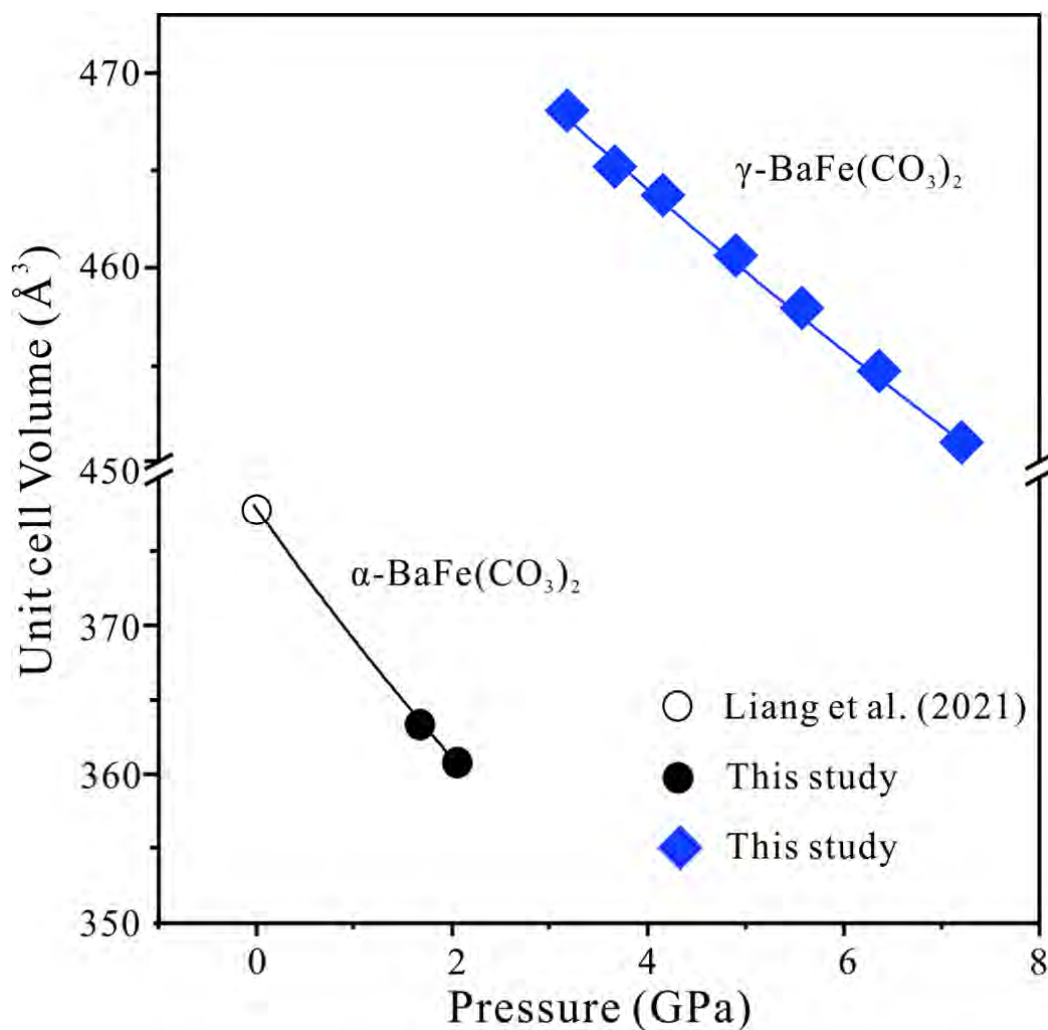
601 patterns of α-BaFe(CO<sub>3</sub>)<sub>2</sub> phase at 2.0 GPa and 300 K (b), and γ-BaFe(CO<sub>3</sub>)<sub>2</sub> phase at

602 5.6 GPa and 300 K (c). The black, red, and blue solid lines represent the α-, α + γ-

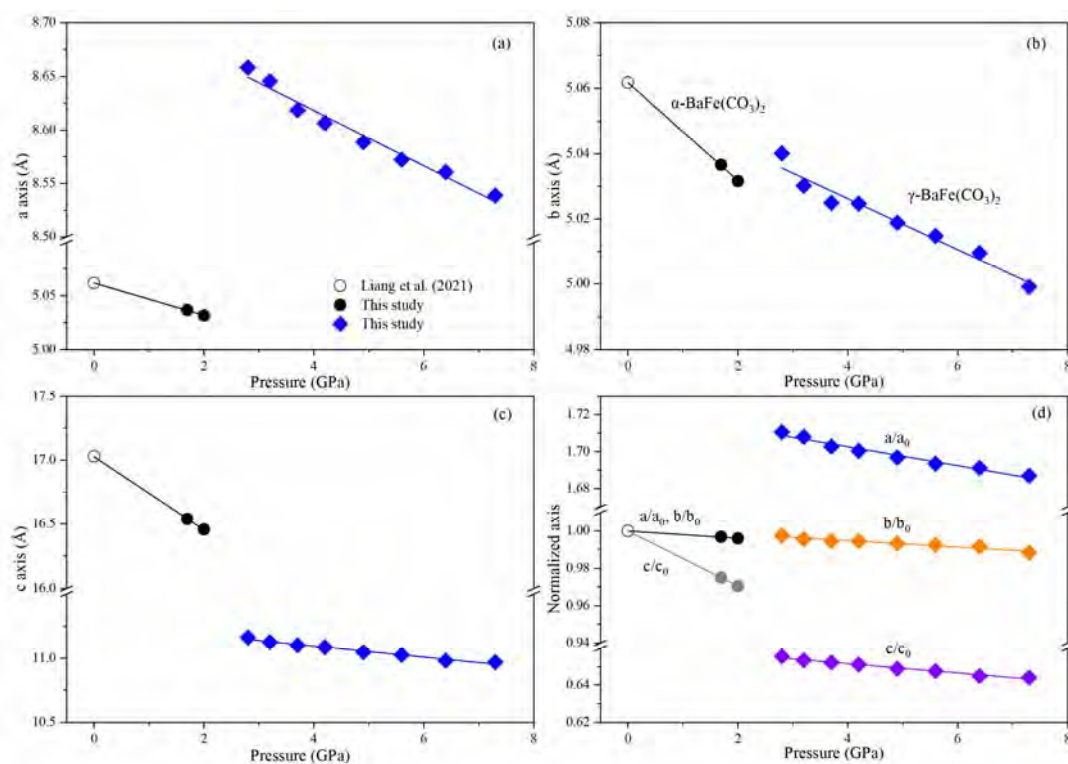
603 (coexistence of α-BaFe(CO<sub>3</sub>)<sub>2</sub> and γ-BaFe(CO<sub>3</sub>)<sub>2</sub> phase), and γ-BaFe(CO<sub>3</sub>)<sub>2</sub>. Le Bail

604 profile fits of the structural model of BaFe(CO<sub>3</sub>)<sub>2</sub> with: α-phase (*R* $\bar{3}$ *m*) at 0–2 GPa and

605  $\gamma$ -phase ( $C2/c$ ) at 3.2–7.3 GPa GPa. The black forks, red solid lines, and blue lines  
606 stand for the observed, calculated, and deviation diffraction patterns, respectively. The  
607 short black lines are the standardized diffraction peak positions of  $\alpha$ - and  $\gamma$ -  
608  $\text{BaFe}(\text{CO}_3)_2$  phase, respectively.



609  
610 **Figure 3.** Pressure-volume profiles of  $\text{BaFe}(\text{CO}_3)_2$ . The black solid circles:  $\alpha$ -  
611  $\text{BaFe}(\text{CO}_3)_2$  phase, this study; blue solid diamonds:  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$ , this study; black  
612 open circle:  $\alpha$ - $\text{BaFe}(\text{CO}_3)_2$  phase, Liang et al. (2021). The black and blue solid curves:  
613 the BM2-EoS fit of  $\alpha$ - $\text{BaFe}(\text{CO}_3)_2$  and  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$ , respectively. Error bars smaller  
614 than symbols are not shown for clarity.



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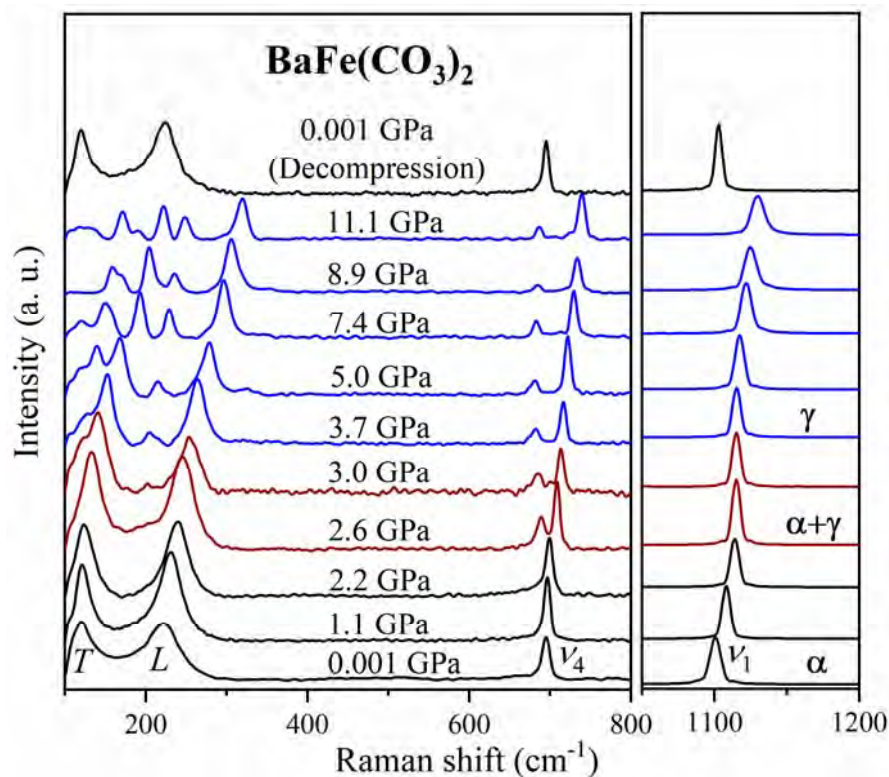
617 **Figure 4.** Second-order Birch–Murnaghan fittings to the compression pressure-lattice

618 parameters (a–c) and normalized lattice parameters (d) of BaFe(CO<sub>3</sub>)<sub>2</sub>. The black,

619 blue, orange and violet solid curves represent the BM2-EoS fittings of lattice

620 parameters. Error bars smaller than symbols are not shown for clarity.

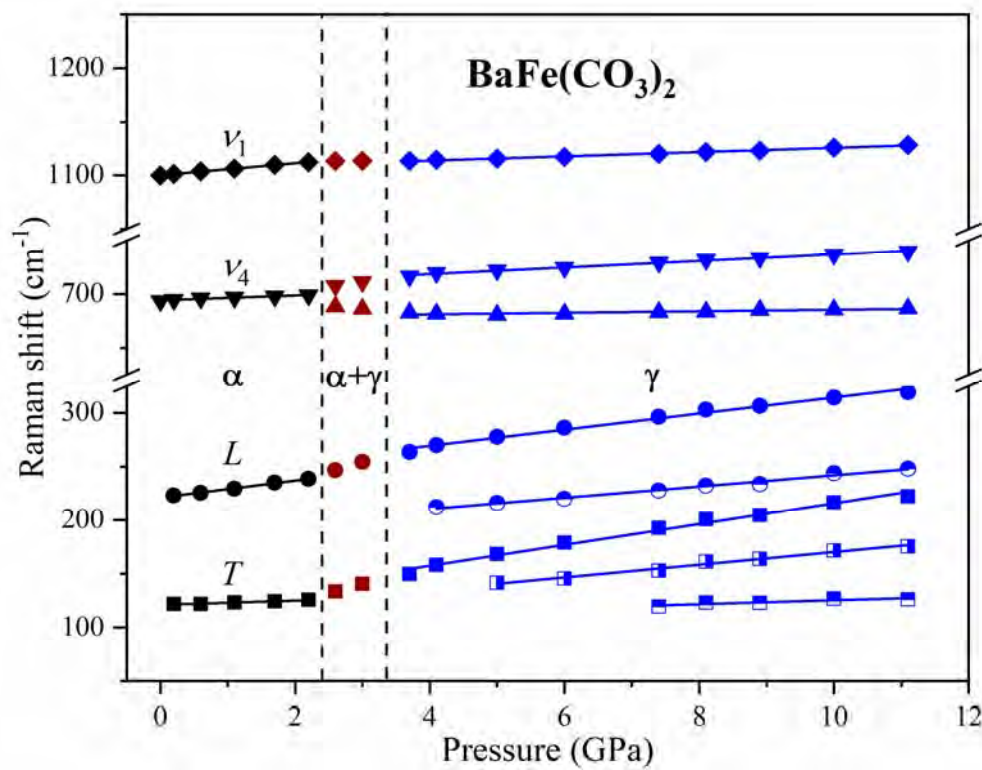
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622

623 **Figure 5.** Representative Raman spectra of  $\text{BaFe}(\text{CO}_3)_2$  at high pressures and room  
624 temperature. The black, red, and blue solid lines represent the  $\alpha$ -,  $\alpha + \gamma$ - (coexistence  
625 of  $\alpha$ - $\text{BaFe}(\text{CO}_3)_2$  and  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$  phase), and  $\gamma$ - $\text{BaFe}(\text{CO}_3)_2$ . The Raman spectrum  
626 of recovered sample (after decompression from 11.1 GPa) is consistent with that of  $\alpha$ -  
627  $\text{BaFe}(\text{CO}_3)_2$  phase at ambient conditions, which indicates a reversible phase transition  
628 of  $\text{BaFe}(\text{CO}_3)_2$ .

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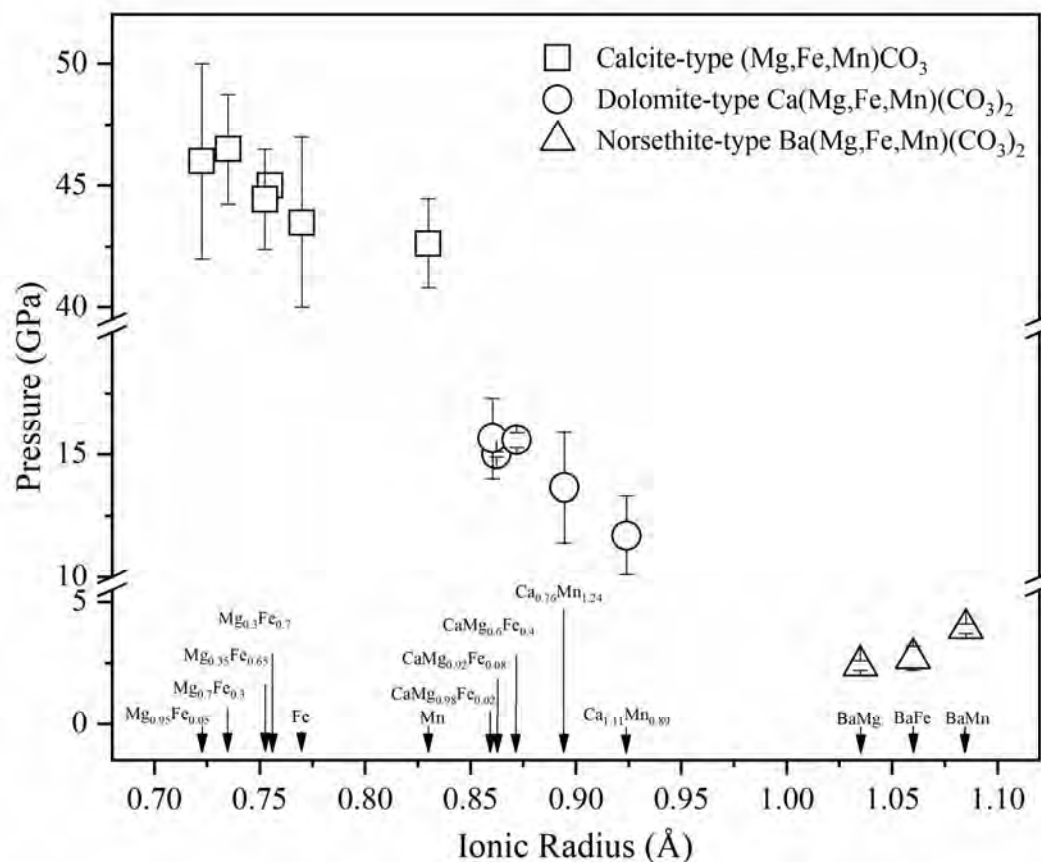


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631 **Figure 6.** Representative Raman shifts of  $\text{BaFe}(\text{CO}_3)_2$  at high pressures and room  
632 temperature. Error bars smaller than symbols are not shown for clarity.

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634

635 **Figure 7.** Phase stability of calcite-type  $(\text{Mg,Fe,Mn})\text{CO}_3$ , dolomite-type  
636  $\text{Ca}(\text{Mg,Fe,Mn})(\text{CO}_3)_2$ , norsethite-type  $\text{Ba}(\text{Mg,Fe,Mn})(\text{CO}_3)_2$  carbonates at ambient  
637 temperature. The data are derived from this study for  $\text{BaMg}(\text{CO}_3)_2$ ,  $\text{BaMn}(\text{CO}_3)_2$ ,  
638  $\text{BaFe}(\text{CO}_3)_2$ , for  $\text{Mg}_{0.95}\text{Fe}_{0.05}\text{CO}_3$  from Spivak et al. 2014, for  $\text{Mg}_{0.7}\text{Fe}_{0.3}\text{CO}_3$  from  
639 Chariton et al. 2020, for  $\text{Mg}_{0.35}\text{Fe}_{0.65}\text{CO}_3$  from Fu et al. 2017, for  $\text{Mg}_{0.3}\text{Fe}_{0.7}\text{CO}_3$  from  
640 Lin et al. 2012, for  $\text{FeCO}_3$  from Cerantola et al. 2015, for  $\text{MnCO}_3$  from Merlini et al.  
641 2015, for  $\text{CaMg}_{0.98}\text{Fe}_{0.02}(\text{CO}_3)_2$  from Binck et al. 2020, for  $\text{CaMg}_{0.92}\text{Fe}_{0.08}(\text{CO}_3)_2$  from  
642 Vennari and Williams 2018, for  $\text{CaMg}_{0.6}\text{Fe}_{0.4}(\text{CO}_3)_2$  from Merlini et al. 2017, for  
643  $\text{Ca}_{0.76}\text{Mn}_{1.24}(\text{CO}_3)_2$  from Palaich et al. 2015, and for  $\text{Ca}_{1.11}\text{Mn}_{0.89}(\text{CO}_3)_2$  from Wang et  
644 al. 2022. The cation radii of  $\text{Ba}^{2+}$ ,  $\text{Ca}^{2+}$ ,  $\text{Mn}^{2+}$ ,  $\text{Fe}^{2+}$ , and  $\text{Mg}^{2+}$  are 1.35, 1.00, 0.83,  
645 0.78, and 0.72 Å, respectively (Shannon and Prewitt 1969).

646

647 **Table 1.** Lattice constants and unit cell volumes of the  $\alpha$ -BaFe(CO<sub>3</sub>)<sub>2</sub> and  $\gamma$ -  
648 BaFe(CO<sub>3</sub>)<sub>2</sub> phase at high pressure and 300 K.

Phase	P (GPa)	<i>a</i> (Å)	<i>b</i> (Å)	<i>c</i> (Å)	$\beta$ (°)	<i>V</i> (Å <sup>3</sup> )	$\rho$ (g/cm <sup>3</sup> )
	0 <sup>a</sup>	5.062(1)	5.062(1)	17.027 (1)	-	377.79 (2)	3.34 (1)
$\alpha$ -phase ( <i>R</i> $\bar{3}m$ )	1.7	5.037 (1)	5.037 (1)	16.540 (1)	-	363.34 (18)	3.47 (2)
	2.0	5.032 (2)	5.032 (2)	16.456 (1)	-	360.79 (09)	3.50 (1)
	3.2	8.646 (2)	5.030 (1)	11.127 (3)	104.742 (2)	467.97 (12)	3.59 (4)
	3.7	8.619 (2)	5.025 (2)	11.104 (3)	104.751 (2)	465.06 (14)	3.61 (6)
	4.2	8.607 (2)	5.025 (2)	11.087 (3)	104.791 (2)	463.57 (13)	3.63 (1)
$\gamma$ -phase ( <i>C</i> 2/ <i>c</i> )	4.9	8.589 (2)	5.019 (2)	11.048 (3)	104.852 (2)	460.33 (14)	3.65 (4)
	5.6	8.572 (3)	5.015 (3)	11.027 (3)	105.023 (3)	457.83 (15)	3.67 (3)
	6.4	8.561 (3)	5.010 (3)	10.981 (3)	105.151 (3)	454.55 (15)	3.70 (1)
	7.3	8.538 (3)	4.993 (3)	10.968 (3)	105.341 (3)	450.90 (14)	3.73 (1)

649 **Notes:** In  $\alpha$ -phase,  $\alpha = \beta = 90^\circ$ ,  $\gamma = 120^\circ$ . In  $\gamma$ -phase,  $\alpha = \gamma = 90^\circ$ . <sup>a</sup>: Liang et al. (2021).

650

651 **Table 2.** Vibrational parameters of BaFe(CO<sub>3</sub>)<sub>2</sub> at high pressures and room  
652 temperature.

Raman modes	$\alpha$ -BaFe(CO <sub>3</sub> ) <sub>2</sub> (0–2.2 GPa)			$\gamma$ -BaFe(CO <sub>3</sub> ) <sub>2</sub> (3.7–11.1 GPa)		
	$\nu_{0i}$	$dv_i/dP$	$\gamma_i$	$\nu_{0i}$	$dv_i/dP$	$\gamma_i$
<i>T</i>	123	2.03(11)	0.68	107	1.89(5)	1.62
				111	6.10(6)	5.08
				118	9.45(4)	7.12
<i>L</i>	221	8.05(31)	1.51	189	5.09(9)	2.44
				239	7.37(7)	2.79
$\nu_4$	694	2.48(9)	0.15	678	0.81(5)	0.11
				706	3.01(7)	0.39
$\nu_1$	1100	5.71 (2)	0.21	1106	2.04(3)	0.17

653 **Notes:**  $\nu_{0i}$  are in cm<sup>-1</sup>,  $dv_i/dP$  are in cm<sup>-1</sup>/GPa. The reference frequency at room pressure ( $\nu_{i0}$ ) and pressure  
654 coefficients,  $\partial\nu_i/\partial P$ , were used to calculate the mode Grüneisen parameters ( $\gamma_i$ ) using the fitted  $K_{T0}$  values obtained  
655 in this study from BM2-EoS fitting:  $K_0=41.15(8)$  GPa for  $\alpha$ -BaFe(CO<sub>3</sub>)<sub>2</sub>,  $K_0=91.4(19)$  GPa for  $\gamma$ -BaFe(CO<sub>3</sub>)<sub>2</sub>.

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