

1 REVISION 2, Correction date: 01aug 2023

2 **Multi-wavelength Raman spectroscopy of natural nanostructured carbons**

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9 **ABSTRACT**

10 There is an extensive range of carbon substances with poorly ordered structures that are not  
11 well understood. Yet they are important as indicators of conditions of related geological  
12 processes. The carbon minerals include nanocrystalline graphite, natural analogues of glass-like  
13 carbon (GLC) – shungite and impact ultrahigh-pressure GLC, recently discovered  
14 ultranocrystalline diamond, as well as natural carbon nanocomposites of diamond, lonsdaleite  
15 and graphite. Studying these natural carbon substances using a standard Raman approach with  
16 excitation by visible radiation may lead to a significant distortion of the understanding of their  
17 phase states. This paper presents in detail for the first time the spectral features of natural, poorly  
18 ordered and multiphase  $sp^2$ - $sp^3$  carbon composites by multi-wave Raman spectroscopy using  
19 laser excitations from visible to ultraviolet light applied to natural low-ordered carbon substances  
20 - nanocrystalline graphite and shungite, nanocrystalline and ultranocrystalline diamond, and  
21 multiphase carbon aggregates. The carbon state resolution advantages of ultraviolet Raman  
22 spectroscopy for phase analysis of nanostructured and poorly ordered polycomponent carbon

23 substances containing  $sp^2$ - and  $sp^3$ -carbons are presented. Raman spectroscopy with ultraviolet  
24 excitation can also be applied in the analysis of industrial carbon materials, such as glassy carbon  
25 and functional carbon nanocomposites, including ultranocrystalline diamond, lonsdaleite, and  
26 amorphous  $sp^3$ -carbon components.

27 **Keywords:** Raman spectroscopy, visible and ultraviolet excitation, natural nanostructured  
28 carbons

## 29 **Introduction**

30 Raman spectroscopy is currently highly sought after as a tool for studying carbon-based  
31 substances. Raman spectroscopy can readily identify crystalline substances such as diamond  
32 (Krishnan 1946; Ferrari 2002), graphite (Tan et al. 1999a; Tan et al. 2004; Saito 2010; Ferrari  
33 2007; Tuinstra, Koenig 1970), and fullerite (Buseck et al. 1992; Kovalevski 2008). Great  
34 opportunities are also open to study molecular (Shumilova et al. 2014, Tan et al. 1999b; Wang et  
35 al 2012), amorphous (Golubev et al 2019) and low-ordered carbon substances including soot  
36 (Sadezky et al. 2005), shungite (Isaenko et al., 2018; Golubev et al 2016), glassy carbon  
37 (Yasumaru et al. 2004), nanocrystalline graphite (Ferrari 2007; Gupta et al. 2003; Khanchuk et  
38 al. 2013; Plyusnina et al. 2013; Berdnikov et al 2014; Piskanec et al. 2004), diamond-like carbon  
39 (Ferrari 2002, Ferrari, Robertson 2001, 2004; Shumilova et al. 2013), nanocrystalline diamond  
40 (Chen et al. 2018; Davydov et al. 2004; Davydov et al. 2006; Goryainov et al 2014; Filik et al.  
41 2006; Praver et al. 2000; Kis et al., 2016; Shumilova et al. 2016; Mermoux 2017) and  
42 ultrananocrystalline diamond (Shenderova 2006; Shumilova et al. 2020). The analysis and  
43 interpretation of low-ordered substances are more complex, often necessitating the use of  
44 complementary techniques such as transmission electron microscopy (TEM) for structural and  
45 chemical analyses. Although amorphous carbon, nanocrystalline diamonds, and diamond-like  
46 carbon were studied by multi-wavelength Raman spectroscopy (Ferrari, Robertson 2001, 2004;

47 Filik et al. 2006), low-ordered/nanocrystalline carbonaceous matters with  $sp^2$  hybridization have  
48 mostly been investigated by visible Raman spectroscopy (Wopenka, Pasteris, 1993; Tuinstra,  
49 Koenig, 1970; Beyssac et al 2002).

50 When comparing Raman spectroscopy data of nanocrystalline diamonds, diamond-like  
51 carbon, and graphite-like substances, it has been observed that the Raman signal originating from  
52 the  $sp^3$  carbon state, which is excited by visible lasers in low-crystalline and amorphous  
53 substances, can be overlapped by the D-band of graphite-like carbon (Ferrari, Robertson 2004). It  
54 is known that a visible 633 nm laser beam excites a D-band centered at  $1330\text{ cm}^{-1}$ , which is a  
55 characteristic of low-ordered  $sp^2$  carbon substances, such as shungite (Golubev 2013; Golubev et  
56 al. 2016; Sychov et al. 2016) and nanocrystalline graphite (Danilova et al. 2015; Tan 2004;  
57 Merlen et al. 2017). The band is close to the diamond  $T_{2g}$  mode position at  $1332\text{ cm}^{-1}$  (Krishnan  
58 1946), which complicates the interpretation of the Raman spectra of poor-crystalline carbon  
59 substances with multiple phases, including nanocrystalline matter, due to the 50–250 times more  
60 intense scattering by  $sp^2$  carbon compared to  $sp^3$  sites (Ferrari, Robertson 2004). The latter results  
61 in very weak  $sp^3$ -carbon band in the spectra of poorly ordered carbon multiphase substances or  
62 even complete overlapping by D-band of  $sp^2$ -carbon component. For this reason, the  $sp^3$ -carbon  
63 band becomes invisible and may be neglected in the spectroscopic data interpretation of poorly  
64 ordered carbons. In addition, the circumstance may result in misinterpretation of the Raman  
65 spectra of diamond or amorphous tetrahedral diamond-like carbon (ta-C) instead of the D-band of  
66  $sp^2$  carbon.

67 The correct phase diagnostics of carbon substance is largely determined by an accurate  
68 interpretation of Raman spectrum bands around  $1330\text{ cm}^{-1}$ , which is complicated by different  
69 effects of the Raman shift of the D-band of graphite-type substances (Ferrari, Robertson 2001,  
70 2004). In the case of carbon multicomponent materials, it has been observed that the relative

71 intensities of certain Raman peaks can strongly depend on the wavelength of the laser used for  
72 excitation (Ferrari, Robertson 2001, 2004, Mermoux 2017). Thus, the effect could be used in the  
73 application of ultraviolet (UV) Raman spectroscopy to solve the problem of Raman bands of  $sp^2$   
74 and  $sp^3$  carbons overlapping in spectra range of near  $1330\text{ cm}^{-1}$  (Ferrari, Robertson 2001, 2004,  
75 Mermoux 2017). Indeed, UV Raman spectroscopy with excitation wavelengths ( $\lambda$ ) ranging from  
76 244 to 364 nm holds promise for the analysis of low-crystalline carbon substances.

77 UV Raman spectroscopy does not fundamentally differ from routine analysis using visible  
78 laser excitation. It has technical advantages that include the absence of fluorescence in the Raman  
79 active spectral range and the increase in sensitivity, which is proportional to  $1/\lambda^4$  ( $\lambda$  is the laser  
80 wavelength). However, due to the higher energy influence of ultraviolet radiation, carbon  
81 samples can experience more intense heating, which may result in structural alterations and even  
82 combustion in an air atmosphere (Mermoux, 2017), that requires accurate control of energy  
83 treatment. In addition, there are specific technical challenges associated with focusing invisible  
84 radiation in the optical system and on a sample, where the laser spot is increased to  $5\text{ }\mu\text{m}^2$ . This  
85 decrease in locality of the analysis leads to a larger area being probed during the measurement.  
86 Finally, it is worth noting that the instruments for UV Raman spectroscopy are significantly more  
87 expensive and more challenging to operate compared to those used for visible spectroscopy.

88 Ferrari and Robertson applied multiwavelength Raman spectroscopy in the analysis of low-  
89 ordered carbon substances, including amorphous carbon, nanocrystalline diamonds, and  
90 diamond-like carbon (Ferrari 2002, 2007; Ferrari, Robertson 2001, 2004). They demonstrated  
91 that the presence of a D-band in the Raman spectrum of disordered or amorphous carbons was  
92 influenced not only by the quantity of carbon in  $sp^2$ -hybridization but also by the nature of its  
93 structure. They also showed that with UV laser excitation, D-band was absent in the spectra of

94 crystalline graphites, but the residual D-band could be present in the UV Raman spectrum being  
95 shifted to  $1400\text{ cm}^{-1}$  only if  $\text{sp}^2$  carbon atoms were grouped to disordered rings, and in low-  
96 crystalline polymerized substances (Ferrari, Robertson 2001).

97 Moreover, it has been recently discovered (Osipov et al. 2018) that increasing the  
98 excitation energy makes  $\text{sp}^3$  and  $\text{sp}^2$  states more distinguishable. Thus, under UV excitation,  $\text{sp}^2$   
99 and  $\text{sp}^3$  carbon sites become more recognizable.

100 Despite detailed studies of different carbon materials, the spectroscopic characteristics of  
101 low-ordered graphite-like substances ( $\text{sp}^2$ -carbon), including nanocrystalline graphite and  
102 shungite, have not been previously examined in detail using UV excitation. In addition, the  
103 studies on mixtures of  $\text{sp}^2$  and  $\text{sp}^3$  carbons are limited.

104 In this paper, we present the results of our systematic studies of low-ordered natural carbon  
105 materials by UV Raman spectroscopy including  $\text{sp}^3$ ,  $\text{sp}^2$  carbons and their mixtures. Here we  
106 present the detailed analysis of UV Raman spectra of ultrananocrystalline diamond,  
107 nanocrystalline diamond, polyphase carbon aggregate, nanocrystalline graphite and shungite. The  
108 obtained spectroscopic characteristics demonstrate specific structure features and the perspectives  
109 for studying low-crystalline multiphase substances with  $\text{sp}^2$  and  $\text{sp}^3$  carbon states that are  
110 important for the identification of  $\text{sp}^3$  carbon in either natural specimens or synthesized materials.  
111 In particular, we have employed UV Raman spectroscopy to successfully analyze the natural  
112 aggregates of ultrananocrystalline diamond with nondiamond carbon on surfaces of diamond  
113 nanocrystallites.

114

### 115 **Samples and experimental methods**

116 In our previous studies, we provided Raman analysis with visible and UV laser excitations  
117 of natural samples of poor-ordered  $\text{sp}^2$  and  $\text{sp}^3$  carbons with known phase states by preliminary

118 analysis of their chemical composition and structure. Here we present the multiwave Raman data  
119 of natural samples from different objects (Fig. 1) – shungite (Shunga, Russia) (Golubev et al.  
120 2016; Melezhik et al 2004; Buseck et al. 1992, 1997 and others), nanocrystalline graphite from  
121 graphitic shists of the Nerkayu metamorphic complex in the Subpolar Urals (Russia) (Shumilova  
122 et al. 2016); natural ultrahigh-pressure (UHP) glass-like carbon (GLC), natural impact  
123 nanocrystalline diamond, ultrananocrystalline diamond and polyphase carbon aggregate from  
124 impactites of the Kara giant crater (Pay-Khoy, Russia). The carbons have been described in detail  
125 in our papers (Shumilova et al. 2014, 2018a, 2019, Ulyashev et al. 2018).

126       The visible Raman spectroscopy studies were conducted using LabRam HR800 (Horiba,  
127 Jobin Yvon) high-resolution Raman spectrometer at the Center of Collective Use «Geoscience»,  
128 the Institute of Geology, Federal Research Center of Komi Science Center of Ural Branch of  
129 Russian Academy of Sciences, Syktyvkar. Registration conditions: gratings – 1800 and 600  
130 g/mm; confocal hole size – 300 and 500  $\mu\text{m}$ ; slit – 100  $\mu\text{m}$ ; objectives –  $\times 50$ ,  $\times 100$ . We used an  
131 onboard He-Ne laser ( $\lambda=633$  nm, output power 2–20 mW) and Ar<sup>+</sup> laser ( $\lambda=515$  nm and  $\lambda=$   
132 488 nm, output power 1–100 mW). The duration of signal accumulation varied from 1 to 10  
133 seconds, and the number of measurements in one section of the spectral range was 3-20 times.  
134 The spectra were recorded at room temperature +20°C.

135       The UV Raman spectroscopy studies were carried out at the Keldysh Research Center  
136 (Moscow, Russia) using a T64000 spectrometer (Horiba) with the following registration  
137 conditions:  $\lambda=244$  nm; output power: 0.4–1.5 mW; confocal pinhole – 300  $\mu\text{m}$ ; grating – 2400  
138 g/mm; slit – 100  $\mu\text{m}$ ; objective –  $\times 40$  UVB. The signal accumulation period and number of  
139 measurements in one spectral range section were 10 sec and 20 times, respectively. The spectra  
140 were recorded at room temperature +20°C.

141 UV Raman spectroscopy studies of natural carbon polyphase aggregate were conducted at  
142 the St. Petersburg State University Resource Center «Geomodel» using a LabRam HR  
143 spectrometer (Horiba). Registration conditions:  $\lambda=325$  nm; output power – 5 mW; confocal  
144 pinhole – 300  $\mu\text{m}$ ; grating – 1800 g/mm; objective –  $\times 40$ . The signal accumulation period and  
145 number of measurements in one spectral range section were 2 sec and 60-80 times, respectively.  
146 The spectra were recorded at room temperature  $+20^{\circ}\text{C}$

147 Raman spectra were curve-fitted by LabSpec 5.36 software using the pseudo-Voigt  
148 function, which is a linear combination of a Gaussian curve and a Lorentzian curve, to determine  
149 peak positions, their full bandwidth at half maximum (FWHM) and their integral intensity (Table  
150 1). When comparing the band intensity in Raman spectra, the relative peak integral intensities  
151 were analyzed upon calculating them as a ratio of one peak area to a total area of all peaks in the  
152 spectrum. This value characterizes a peak fraction in Raman spectrum.

153 The Raman spectra of nanocrystalline graphite and shungite were deconvoluted by the  
154 method of Sadezky (2005) and Tan (2004) into the bands and their overtones and combinations –  
155 D4, D, D3, G, D2, 2D4, D4 + D, 2D, D3 + D, D + G, 2D2. UV Raman spectra of  
156 ultrananocrystalline diamonds were deconvoluted into the diamond line and additional bands at  
157 the fixed positions: 1050, 1254, 1590, 1640, 1740  $\text{cm}^{-1}$ . In the UV Raman spectra of  
158 nanocrystalline diamond, only the line corresponding to  $T_{2g}$  mode of diamond is distinguished.  
159 The visible Raman spectrum of the polyphase carbon aggregate was deconvoluted similarly into  
160 shungite bands, and the UV Raman spectrum was deconvoluted into shungite bands and diamond  
161 lines.

162 To determine the internal nanostructure features of the analyzed carbon substances, TEM  
163 and high-resolution TEM (HRTEM) images with corresponding electron diffraction patterns

164 were obtained (Shumilova et al., 2014, 2016, 2018a, 2019, Ulyashev et al. 2018). The TEM  
165 observations were conducted at the Central Facility for Electron Microscopy of RWTH Aachen  
166 University (Aachen, Germany) and the Institute of Geology of the FRC Komi SC RAS  
167 (Syktyvkar, Russia). The following TEM instruments were used: a high-resolution transmission  
168 electron microscope FEI TECNAI F20 equipped with an energy-dispersive X-ray spectrometer  
169 (EDX), an energy filter (GATAN imaging filter), a scanning transmission electron microscopy  
170 (STEM) unit, and a digital CCD-Camera, operated at 200 kV; and a transmission electron  
171 microscope Tesla BS-500, operated at 60 kV. The TEM samples were prepared by the powder  
172 method with copper grids with holey carbon films.

173

174

## Results

175 The recorded visible Raman spectra of shungite and nanocrystalline graphite samples  
176 clearly demonstrate both D and G bands of  $sp^2$  carbon, and there are also  $D_2$ ,  $D_3$  and  $D_4$  bands,  
177 and their overtones and combinations, observed depending on excitation laser radiation: 2450–  
178  $2500\text{ cm}^{-1}$  ( $D_4 + D$ ),  $2650\text{--}2700\text{ cm}^{-1}$  ( $2D$ ),  $2800\text{--}2850\text{ cm}^{-1}$  ( $D_3 + D$ ),  $2900\text{--}2950\text{ cm}^{-1}$  ( $D + G$ )  
179 and  $3180\text{--}3230\text{ cm}^{-1}$  ( $2D_2$ ). The spectra obtained through UV radiation exhibit a distinct  
180 difference compared to the visible Raman data, which will be presented below.

181

### Nanocrystalline graphite

182 Raman spectra of nanocrystalline graphite were recorded in ranges of  $100\text{--}4000\text{ cm}^{-1}$  for  
183 visible excitation and  $1000\text{--}4000\text{ cm}^{-1}$  for UV. The Raman spectra of graphite with 100–200 nm  
184 crystallite sizes (Fig. 2d) with visible excitation contain the bands – D, G,  $D_2$ ,  $D_4+D$ ,  $2D$ ,  $D+G$ ,  
185  $2D_2$  (Fig. 2a, 2b, Table 1). With decreasing the exciting radiation wavelength from 633 nm to 515  
186 nm, the D-band shifts from  $1332\text{ cm}^{-1}$  to  $1355\text{ cm}^{-1}$  with  $FWHM_D$  widening from 49 to  $57\text{ cm}^{-1}$ ,  
187 and two times decrease in its relative integral intensity from 42 to 21 %. At the same time, the



188 position of the G-band of graphite shifts slightly from 1581 to 1584  $\text{cm}^{-1}$ , where the  $\text{FWHM}_G$   
189 essentially does not change, and the contribution of the G-band to the total area of the spectrum  
190 bands also remains almost the same: 23 % and 25 %.

191 Under UV excitation the D-band in a Raman spectrum of nanocrystalline graphite is weak,  
192 its contribution to the total spectral intensity is only 3 %, and its position shifts from 1332  $\text{cm}^{-1}$   
193 (633 nm laser) to 1458  $\text{cm}^{-1}$  (244 nm laser) with  $\text{FWHM}_D$  increasing to 106  $\text{cm}^{-1}$ . At the same  
194 time, the G-band shifts slightly to 1575  $\text{cm}^{-1}$  with  $\text{FWHM}_G$  decreasing to 19  $\text{cm}^{-1}$ , but the relative  
195 integral intensity of the G-band increases substantially and reaches 84 %. In the higher spectral  
196 range under UV excitation, the combination of D+G bands and the nitrogen line of the ambient  
197 air at 2330  $\text{cm}^{-1}$  are visible, while other bands are absent in the UV Raman spectra (Fig. 2c).

## 198 **Shungite**

199 Raman spectra of shungite without microscopically observed crystalline planes (Fig. 3d)  
200 were recorded in the spectral range 100–4000  $\text{cm}^{-1}$  with visible excitation and 100–2600  $\text{cm}^{-1}$   
201 under UV excitation. In visible Raman spectra of shungite (Fig. 3a, 3b)  $D_4$ , D,  $D_3$ , G,  $D_2$  bands  
202 and  $D_4+D$ , 2D,  $D_3+D$ , D+G,  $2D_2$  sub-bands are recognized (Table 1). The position of a G-band  
203 depending on the used laser wavelength varies insignificantly – from 1598 to 1595  $\text{cm}^{-1}$ , and the  
204  $\text{FWHM}_G$  slightly increases from 51 to 57  $\text{cm}^{-1}$ . The contribution of the G-band to the total  
205 integral area of all bands in the spectrum also does not change.

206 In the meantime, the D-band behaves differently. Using 633 nm and 515 nm lasers we  
207 observe a D-band shift from 1329  $\text{cm}^{-1}$  to 1349  $\text{cm}^{-1}$ , with the corresponding  $\text{FWHM}_D$  decreasing  
208 from 82 to 71  $\text{cm}^{-1}$  and significant decrease in the relative integral intensity from 63 to 53 %.  
209 Under UV excitation, the D-band becomes unclear in the Raman spectrum of shungite. Its  
210 contribution to the total spectrum intensity is 9 %, and the position shifts from 1329  $\text{cm}^{-1}$  at a 633  
211 nm laser excitation to 1484  $\text{cm}^{-1}$  at a 244 nm laser with the  $\text{FWHM}_D$  192  $\text{cm}^{-1}$ . In contrast, the

212 relative G-band integral intensity increases several times (up to 65 %). In addition, under UV  
213 excitation, the shungite Raman spectra contain wide bands of a supporting glass substrate – 465,  
214 797, 1064  $\text{cm}^{-1}$ , the peaks of atmospheric oxygen – 1553  $\text{cm}^{-1}$  and nitrogen – 2328  $\text{cm}^{-1}$ ; other  
215 characteristic bands for the visible excitation listed above for shungite are absent (Fig. 3c).

216

### 217 **Ultra-high-pressure glass-like carbon**

218 Natural ultra-high pressure (UHP) glass-like carbon has been enriched from the Kara  
219 impactites, where it is found in association with impact ultrananocrystalline diamond (Shumilova  
220 et al. 2019). From HRTEM images, the carbon material appears to have a disordered structure,  
221 lacking a truly crystalline arrangement (Fig. 4d). The UHP GLC is characterized by the presence  
222 of hollow onion-like structures within an essentially amorphous carbon matrix, indicating the  
223 formation under extremely high pressure and temperature conditions (Ulyashev et al., 2018). The  
224 structural features differ from the above described shungite that has a low pressure origin.

225 Raman spectra of natural UHP GLC were recorded in the spectral range 100–4000  $\text{cm}^{-1}$   
226 with visible excitation and 500–2600  $\text{cm}^{-1}$  under UV excitation. In the visible Raman spectra of  
227 UHP GLC (Fig. 4a, 4b), the bands  $D_4$ , D,  $D_3$ , G and sub-bands 2D,  $D_3+D$ ,  $2D_2$  are recognized  
228 (Table 2). The position of a G-band depending on the laser wavelength used varies insignificantly  
229 from 1594 to 1600  $\text{cm}^{-1}$ , and the corresponding  $\text{FWHM}_G$  increases from 54 to 75  $\text{cm}^{-1}$ . The  
230 contribution of G-band to the total integral area of all bands in the spectrum increases from 9-13  
231 % (VIS excitation) to 48% (UV excitation).

232 At the same time, D-band behaves differently. Using 633 nm and 515 nm lasers, we  
233 observe a D-band shift from 1344  $\text{cm}^{-1}$  to 1350  $\text{cm}^{-1}$ , with a small change of the  $\text{FWHM}_D$  from  
234 121 to 118  $\text{cm}^{-1}$  and the relative integral intensity from 23 to 21 %. Under UV excitation, the D-  
235 band shifts to 1411  $\text{cm}^{-1}$  with a large increase in  $\text{FWHM}_D$  up to 329  $\text{cm}^{-1}$ . Its contribution to the

236 total spectrum intensity increases significantly from 9 to 52 %. In addition, under UV excitation,  
237 the UHP GLC Raman spectra include wide bands of a supporting glass substrate,  $789\text{ cm}^{-1}$ , and  
238 peak of atmospheric nitrogen,  $2333\text{ cm}^{-1}$ . (Fig. 4c).

239

#### 240 **Nanocrystalline diamond**

241 Raman spectra of nanocrystalline diamond with 20-30 nm crystallites (Fig. 5d) under  
242 visible laser radiation have a peak of diamond at  $1332\text{ cm}^{-1}$  (Fig 5a, 5b) which is overlapped by  
243 intense structureless luminescence in the whole range of Raman spectrum registration,  
244 significantly complicating the identification of a substance, and in some cases, making it  
245 impossible.

246 Under UV laser radiation, the luminescence in the spectrum is absent, thus a distinct Raman  
247 bands can be well observed (Table 3). The average value of the diamond  $T_{2g}$  mode position is  
248  $1330\text{ cm}^{-1}$  (Fig. 5c), which is close to that of a bulk high crystalline diamond. The measured  
249  $FWHM_{T_{2g}}$  is  $18\text{ cm}^{-1}$  depending on the size of the nanocrystallite, which is about 30-50 nm  
250 (Shumilova et al. 2018a).

#### 251 **Ultrananocrystalline diamond**

252 Previous studies (Chen et al. 2018; Praver et al. 2000; Davydov 2003) show that the visible  
253 Raman spectroscopy does not provide informative spectra of nanocrystalline and  
254 ultrananocrystalline diamonds because of their high luminescence, essentially or completely  
255 overlapping the Raman signal. We tested the abilities of UV Raman spectroscopy in the study of  
256 ultrananocrystalline diamond samples present as nanocrystalline aggregates with crystallites of 2-  
257 5 nm in size (Fig. 6d), separated from the Kara impactites by chemical dissolution (Shumilova et  
258 al. 2019). Under visible radiation, the Raman bands of carbon matter are overlapped by extremely  
259 high intensity structureless luminescence in the whole range of registration of Raman spectrum

260 (Fig. 6a, 6b), which leads to the impossibility of phase identification. The luminescence level  
261 exceeds the Raman signal by 2-3 orders of magnitude.

262 Under UV laser excitation, the luminescence is absent in the measured spectral range, and  
263 the Raman bands can be clearly observed (Fig. 6c). The UV Raman spectra of  
264 ultrananocrystalline diamonds have been deconvoluted into diamond line and additional bands at  
265 fixed positions: 1050, 1254, 1590, 1640, 1740  $\text{cm}^{-1}$  (Fig. 6 c, Table 3). In the observed spectra,  
266 the band at 1321  $\text{cm}^{-1}$  ( $\text{FWHM}_{\text{T}_{2g}}=39 \text{ cm}^{-1}$ ) corresponds to the  $\text{T}_{2g}$  diamond mode, shifted from  
267 the standard position, presumably due to heating of the nanocrystalline material during the laser  
268 excitation (Tan et al. 1998; Zouboulis, Grimsditch 1991). According to Ferrari and Robertson  
269 (2004), the observed band at 1050  $\text{cm}^{-1}$  ( $\text{FWHM} = 252 \text{ cm}^{-1}$ ) corresponds to a T-peak, which  
270 conforms to the presence of C–C vibrations in a  $\text{sp}^3$  carbon state. The observed band at 1254  $\text{cm}^{-1}$   
271 ( $\text{FWHM} = 225 \text{ cm}^{-1}$ ) is present in the Raman spectra of nanodiamonds only by UV excitation  
272 resulting from small vibrational domains (Osswald et al. 2009).

273 By spectra deconvolution, we recognize several additional non-diamond bands  
274 corresponding to  $\text{sp}^2$  carbon and some radicals. Among them is a well recognized G-band at 1590  
275  $\text{cm}^{-1}$  of  $\text{sp}^2$  carbon ( $\text{FWHM}=103 \text{ cm}^{-1}$ ). The band at 1640  $\text{cm}^{-1}$  ( $\text{FWHM}=192 \text{ cm}^{-1}$ ) may belong  
276 to the G peak within ta-carbon (Ferrari & Robertson, 2004) or to O–H group vibrations. The band  
277 at 1740  $\text{cm}^{-1}$  ( $\text{FWHM}=64 \text{ cm}^{-1}$ ) presumably corresponds to C=O vibrations (Mochalin et al.  
278 2009). The presence of C=O groups in the studied samples was confirmed by IR spectroscopy  
279 (Shumilova et al. 2019).

## 280 **Natural polyphase carbon aggregates**

281 We further studied natural polyphase carbon aggregate (Fig. 7c) separated from the Kara  
282 impactites by chemical dissolution. According to HRTEM data (Ulyashev et al. 2018), it contains  
283 impact UHP glassy carbon, nanocrystalline diamond and newly formed graphite. The study of

284 natural polyphase aggregates by visible Raman spectroscopy allowed obtaining informative  
285 Raman spectra (Fig. 7a), which turned out to be similar to the spectra of shungite, where the G-  
286 band is at  $1597\text{ cm}^{-1}$  (FWHM =  $58\text{ cm}^{-1}$ ), and D-band has a maximum at  $1345\text{ cm}^{-1}$  (FWHM =  $83$   
287  $\text{cm}^{-1}$ ). Additionally, in the spectrum, D<sub>4</sub>- and D<sub>3</sub>-bands were registered with positions  $1231\text{ cm}^{-1}$   
288 and  $1513\text{ cm}^{-1}$ , respectively (Table 4).

289 Raman spectrum of the carbon polyphase aggregate was recorded under 325 nm ultraviolet  
290 laser excitation (Fig. 7b). Deconvolution of the spectrum revealed three bands: the D-band at  
291  $1414\text{ cm}^{-1}$  (FWHM =  $170\text{ cm}^{-1}$ ), the G-band at  $1581\text{ cm}^{-1}$  (FWHM =  $70\text{ cm}^{-1}$ ) and the diamond  
292 line at  $1325\text{ cm}^{-1}$  (FWHM =  $21\text{ cm}^{-1}$ ). Thus, the use of ultraviolet laser (325 nm) allowed the  
293 detection of diamond in the polyphase carbon aggregate, in addition to shungite, previously  
294 discovered by a visible laser (515 nm).

## 295 Discussion

296 ***D-band nature in sp<sup>2</sup>-carbons.*** It is known that the Raman spectrum of graphite-like carbon  
297 substances contains an intense G-band with a maximum at  $1570\text{--}1610\text{ cm}^{-1}$  (Tan et al. 1999a;  
298 Tan et al. 2004; Saito 2010; Ferrari 2007; Tuinstra, Koenig 1970) corresponding to the  
299 fundamental Raman mode E<sub>2g</sub> (Tan et al. 2004), which is responsible for vibrations of carbon  
300 atoms inside the graphite layer. In addition, the spectra of disordered graphite or poor-structured  
301 sp<sup>2</sup>-carbon have an intensive D-band with a band position in the range of  $1320\text{--}1360\text{ cm}^{-1}$ . While  
302 the nature of G-band has been unequivocally established, the nature of D-band remains unclear,  
303 though it has been extensively studied. Initially, the D-band was explained by a breathing mode  
304 of sp<sup>2</sup>-carbon atoms in carbon rings (Tuinstra, Koenig 1970). The band later received another  
305 interpretation, however.

306 The term “D” - band is coined from the English word, “disordered”. It is important to note  
307 that the D-band is absent in the Raman spectrum of an ideal graphite crystal. Its presence and

308 intensity are associated with a degree of graphite disorder, depending on the size of the graphite  
309 crystallite (Reich, Thomsen 2004). It was also noted that the position of D-band depends on the  
310 exciting radiation energy. Under exciting by a 515 nm laser, its position is  $1350 \text{ cm}^{-1}$ ; when  
311 excited by a 633 nm laser, the D-band is characterized by a shift to the shortwave direction,  $1330$   
312  $\text{cm}^{-1}$  (Reich, Thomsen 2004).

313 Sood et al. (1998) explained the D-band by double resonance caused by defects in the band  
314 structure above slit  $\Delta \approx 1 \text{ eV}$ , leading to the dependence of the phonon wave vector  $q$ , and hence,  
315 the phonon frequency on the incident light energy  $E_1$ , as  $q \sim (E_1 - \Delta)^{1/2}$ . Tan et al. (2004) found  
316 discrepancies between the Stokes and anti-Stokes frequencies of the D-band, but they could not  
317 provide a reason for this. For the bands of the second-order in graphite, an unusual dependence  
318 on the excitation energy was found (Reich, Thomsen 2004). At higher energy laser excitation, the  
319 second-order bands are shifted twice faster compared to the first-order bands. At the same time,  
320 the effect is not caused by graphite structure defects.

321 Analyzing the modern theoretical models, Ferrari has shown that the D-band is associated  
322 with a longitudinal optical photon (LO phonon) around the K point (Brillouin zone) and is  
323 activated by double resonance. Its scattering depends on the excitation energy due to the Kohn  
324 anomaly at the K point (Ferrari 2007; Kohn 1959). The Kohn anomaly is described in detail in  
325 the semiconductor physics, where the Fermi surface is considered (Kohn 1959; Hasdeo et al  
326 2016).

327 Pocsik et al. (1998) demonstrated the D-band shift in a Raman spectrum of graphite under  
328 excitation by lasers with different energies ( $\lambda$  from 1068 to 300 nm), and it was shown that the  
329 position of the D-band (dispersion  $40\text{--}50 \text{ cm}^{-1}/\text{eV}$ ) at excitation with 244 nm laser is in the range  
330 of  $1455\text{--}1485 \text{ cm}^{-1}$ . The shift of the D-band observed in our study with nanostructured  $\text{sp}^2$ -

331 carbon leads to an increase in the wave number for the samples of nanocrystalline graphite and  
332 shungite at 112–155  $\text{cm}^{-1}$ , and the decreases in the band intensity are probably associated with  
333 the scattering effect that depends on the excitation energy due to the Kohn anomalies at K point  
334 (Kohn 1959).

335 In studying low-ordered graphite-like substances by UV Raman spectroscopy, we observed  
336 that the relative integral intensity of the D-band decreased by 2–14 times and its position shifted  
337 up to 112–155  $\text{cm}^{-1}$  towards increasing the wavenumber. Correspondingly, the relative intensity  
338 of the G-band increased by 2–4 times and had a slight shift of 6–11  $\text{cm}^{-1}$  towards the  
339 wavenumber decreasing. Thus, a clearly distinguished D-band in the visible-Raman spectra was  
340 barely noticeable in UV excitation spectra.

341 It is clear from the above data that the decrease in the wavelength of the excitation from  
342 visible to ultra-violet light results in a substantial “simplification” of the Raman spectrum in the  
343  $\text{sp}^2$  carbon crystalline and noncrystalline substances. Under UV excitation, the G-band, as  
344 directly related to the vibrations of carbon atoms inside a graphene layer, is demonstrated more  
345 clearly, and its relative integrated intensity increases significantly. On the contrary, the D-band in  
346 UV Raman spectra has a resonant nature and shifts in  $\text{sp}^2$ -carbons from 1330  $\text{cm}^{-1}$  (633 nm laser)  
347 to 1444–1484  $\text{cm}^{-1}$  (244 nm laser) and becomes almost invisible in a UV spectrum relative to the  
348 G-band. Note that the shift of the D-band position observed under UV radiation cannot be  
349 associated with laser heating during the process of Raman scattering excitation, since heating  
350 would result in band shifts towards decreasing the wavenumber (Tan et al. 1999a).

351 ***Raman features of nanostructured  $\text{sp}^2$ -carbons.*** Comparing the spectra of shungite and  
352 nanocrystalline graphite under UV excitation, we can draw the following conclusions. At first  
353 glance, the UV Raman spectra are similar. If a slight shift in the position of the G-band can be  
354 attributed to thermal heating during the registration of Raman spectra, then the FWHM of the G-

355 band differs significantly; the FWHMs for shungite and nanocrystalline graphite are 39 cm<sup>-1</sup>  
356 and 29 cm<sup>-1</sup>, respectively (Table 2). Thus, the main distinguishing feature of shungite from  
357 nanocrystalline graphite is the FWHM of the G-band. For the shungite, we observe a very low D  
358 band (1484 cm<sup>-1</sup>), as occurred in graphite (Ferrari & Robertson, 2001, 2004).

359 Our studied UHP glass-like carbon samples, which were formed at impact conditions in  
360 tight genetic and spatial relation with nano- and ultrananocrystalline diamond, display different  
361 UV Raman spectra (Fig. 4c) from those of the above described low pressure shungite. The UHP  
362 GLC is characterized by a quite intensive wide Raman band centered at 1411 cm<sup>-1</sup> that can be  
363 assigned to the D-band of sp<sup>2</sup> carbon, which should be absent at UV excitation (Ferrari &  
364 Robertson, 2001, 2004). Ferrari and J. Robertson (2001, 2004) showed that the D-peak intensity  
365 strongly depends on the order of sp<sup>2</sup> sites and the D-peak could be observed in sp<sup>2</sup>-carbons  
366 presented with disordered rings structures. Our TEM observation revealed an essentially  
367 amorphous structure at the atomic level (Fig. 4d), explaining the relatively high intensity of D-  
368 band (Fig. 4c).

369 Thus, according to Ferrari & Robertson (2001, 2004) and our TEM data of the analyzed  
370 carbons, we can conclude that the observed D-band in natural glass-like carbons (low pressure  
371 shungite and impact ultrahigh-pressure GLC) is indicative of defect structures with varying  
372 degrees of carbon ring disorder.

373 ***Raman features of nanostructured sp<sup>3</sup>-carbons.*** Our results demonstrate excellent  
374 analytical capabilities for the characterization of ultrananocrystalline diamonds with crystallite  
375 sizes of 2–5 nm (Fig. 5, Table 3). The diamond type has been actively studied as one of the most  
376 novel types of carbon materials (Chen et al. 2018; Davydov et al. 2004; Davydov et al. 2006;  
377 Shenderova, Gruen 2006).



378 It should be particularly noted that UV excitation for Raman spectra does not significantly  
379 change the position of a fundamental  $sp^3$ -carbon band compared to the excited spectra by  
380 radiation of the visible light, which was previously discussed in detail by Ferrari and Robertson  
381 (Ferrari 2007; Ferrari, Robertson 2001, 2004). However, the experimentally measured position of  
382 a  $T_{2g}$ -mode of diamond essentially depends on the size of the diamond nanocrystallites, with  
383 overheating of the ultrananocrystalline diamond. We argue here that to exclude a temperature  
384 shift of a  $T_{2g}$ -mode of diamond due to laser heating during spectra collection, especially for  
385 nanocrystalline samples, the position of the diamond line in the Raman spectrum has to be  
386 carefully controlled. For accurate measurements, it is necessary to record Raman spectra at the  
387 lowest possible laser power. This provides the possibility of obtaining spectra without a peak  
388 shift caused by laser overheating. The decrease in laser power leads to significant increase of  
389 exposure time. However, only under these conditions can we be sure that the shift of the diamond  
390 line is associated with the structural features of the sample and not with over-heating during the  
391 registration of the Raman spectrum.

392 In this study, we were able to obtain Raman spectra with minimal or no thermal shift of  
393 the bands. This was achieved by using a low power of the laser radiation, which helped in  
394 maintaining the integrity of the Raman signal decreases. To minimize the impact of noise in the  
395 spectra, we increased the exposure time by increasing the number of spectra counts within the  
396 shortest possible time, achieving a signal-to-noise ratio of at least 3. This approach can be used  
397 for accurate Raman measurements aimed at analyzing some structural features of nanocomposite  
398  $sp^3$ -carbons with either visible or UV excitations.

399 At the same time, we must note that nanodiamonds and diamond-like carbons (DLC) have  
400 additional bands in their Raman spectra resulting from a large number of  $sp^2$  sets at their edge  
401 boundaries and amorphous carbon matrix of DLC and specific bands corresponding to the

402 amorphous state of  $sp^3$  sets (Ferrari & Robertson, 2004). The additional bands are presented with  
403 D and G bands of  $sp^2$  carbon, similar to that described above for  $sp^2$  monophasic substances.

404 From the measured UV Raman spectra, the ultrananocrystalline diamond is presented not  
405 only with tiny-sized diamond crystallites but also with tetrahedral amorphous diamond-like  
406 carbon (ta-carbon), which has a high content of  $sp^3$  carbon sites characterized by T and G bands of  
407 DLC, at  $1050\text{ cm}^{-1}$  (FWHM =  $252\text{ cm}^{-1}$ ) and  $1640\text{ cm}^{-1}$  (FWHM= $192\text{ cm}^{-1}$ ), respectively (Table  
408 3). It is especially important that the mentioned T band can be visible only under UV excitation.  
409 According to simulations, it corresponds to C–C  $sp^3$  vibration density of states of ta-carbon  
410 (Ferrari & Robertson, 2004). From the data of ultrananocrystalline diamond (Ferrari and  
411 Robertson 2004), it is expected that the DLC matrix is presented with a high content of  $sp^3$   
412 carbon (about 85 %). For a more certain interpretation of the additional bands, additional TEM  
413 studies with electron energy loss spectroscopy are needed.

414 ***Aggregates of nanostructured  $sp^2$  and  $sp^3$  carbons.*** Our data clearly demonstrate that UV  
415 excitation provides a technical possibility to analyze differently aggregated  $sp^2$  and  $sp^3$  carbons  
416 from nanocrystalline tight growths to ultrananocrystalline diamond with the possible presence of  
417 diamond-like amorphous tetrahedral  $sp^3$  carbon (Fig. 7). It is also possible to analyze structural  
418 features of nanostructural carbons, including C-H and C=O vibrations and other hetero-elemental  
419 bonding within carbon substances. The use of UV Raman spectroscopy does not require the  
420 preparation of special specimens and can be a convenient and user-friendly method in addition to  
421 TEM studies of nanostructured carbons.

422

423

### **Implications**

424 The Raman spectral features of natural low-ordered carbon substances excited by lasers of  
425 different wavelengths have been demonstrated. For the first time we have provided a detailed

426 description of the spectroscopic characteristics obtained under UV radiation for natural  
427 nanocrystalline graphite, shungite, UHP glass-like carbon, as well as for natural nanocrystalline  
428 and ultrananocrystalline diamonds with crystallite sizes of 2-5 nm and polyphase carbon  
429 aggregates. The results demonstrate new perspectives for future studies of natural low-ordered  
430  $sp^2$  carbons using Raman spectroscopy with UV excitation.

431         This study has the potential to significantly enhance our understanding of the structural  
432 characteristics and polymeric features of low-structured natural carbon matter with varying  $sp^2$ -  
433 site orders. A significant decrease in intensity and a considerable shift of the D-band in the Raman  
434 spectra of low-ordered  $sp^2$ -carbon substances using UV excitation allow for the distinction  
435 between  $sp^2$  and  $sp^3$  carbons when they coexist in nano- and ultrananocrystalline individual and  
436 polyphase samples. The latter is explained by the constancy of  $sp^3$ -carbon upon excitation of  
437 Raman scattering by both visible and ultraviolet radiations. UV Raman spectroscopy offers  
438 convenient and user-friendly features that make it an excellent tool for studying nanostructured  
439 carbons. It can effectively analyze the phase composition and ordering characteristics of low-  
440 ordered multicomponent natural carbon substances, including  $sp^3$ - and  $sp^2$ -carbons in polymeric  
441 or nanocrystalline substances. Additionally, UV Raman spectroscopy is highly suitable for  
442 studying nanocrystalline and ultrananocrystalline diamonds, as well as polyphase tight  
443 intergrowths. Its detailed spectroscopic characteristics enable precise characterization and  
444 differentiation of different carbon structures, making it a valuable technique for future research in  
445 this field.

446         Finally, the use of UV Raman spectroscopy may provide a more accurate understanding of  
447 the real phase compositions of nanostructured natural carbons for their potential technological  
448 applications.

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450

### Acknowledgments and funding

451 The authors are thankful to Valeriy Kazakov for help in UV Raman measurements, Joachim  
452 Mayer for help in preliminary TEM studies, Mikhail Filippov for providing samples of shungite  
453 and to Sergey Tkachev for valuable comments. The study was carried out with support of the  
454 project NIR 122040600009-2.

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626

627 **Figure 1.** Geographic position of the studied objects. (1) The Nerkayu complex (N65° E61° with  
628 nanocrystalline graphite. (2) Shun'ga deposit (N62° E35°) with shungite carbon; (3) Kara  
629 astrobleme (N69° E64°) with impact nano- and ultrananocrystalline diamond, impact natural  
630 UHP glass-like carbon.

631

632 **Figure 2.** Nanocrystalline graphite. Raman spectra at different wavelengths excitation (a) 633  
633 nm, (b) 515 nm, and (c) 244 nm. Additional line detected by UV laser radiation is N2-2330 cm<sup>-1</sup>  
634 (room nitrogen); (d) magnified range 1000–2000 cm<sup>-1</sup> of spectrum c; (e) TEM image in bright  
635 field mode (BF) with corresponding electron diffraction pattern (ED).

636

637 **Figure 3.** Shungite. Raman spectra at different laser wavelengths excitation (a) 633 nm, (b) 515  
638 nm, (c) 244 nm. Additional lines detected by UV laser radiation include SiO<sub>2</sub> glass: 465 cm<sup>-1</sup>,  
639 797 cm<sup>-1</sup>, 1064 cm<sup>-1</sup>; O2-1553cm<sup>-1</sup>; N2-2328 cm<sup>-1</sup>. **{auth: ok?}** (d) TEM image in BF mode  
640 with corresponding ED pattern.

641

642 **Figure 4.** Natural carbon glass-like carbon: Raman spectra at different laser wavelengths  
 643 excitation: (a) 633 nm, (b) 515 nm, (c) 244 nm. Additional lines detected by UV laser radiation  
 644 in SiO<sub>2</sub> glass: 797, N2-2333 cm<sup>-1</sup>. (d) TEM image in BF mode with corresponding ED pattern  
 645 (the TEM image is used from (Ulyashev et al. 2018) by permission of the editorial board of  
 646 Vestnik of Geosciences).

647  
 648 **Figure 5.** Natural nanocrystalline diamond. Raman spectra at different laser wavelengths  
 649 excitation (a) 633 nm, (b) 515 nm, (c) 244 nm. (d) TEM image in BF mode with corresponding  
 650 ED pattern.

651  
 652 **Figure 6.** Natural ultrananocrystalline diamond. Raman spectra at different laser wavelengths  
 653 excitation: (a) 633 nm, (b) 515 nm, (c) 244 nm. (d) TEM image in BF mode with corresponding  
 654 ED pattern.

655  
 656  
 657 **Figure 7.** Natural carbon polyphase aggregate of ultrananocrystalline diamond (UNCD) and  
 658 UHP glass-like carbon (GLC). Raman spectra from the aggregate at different laser wavelengths  
 659 excitation (a) 515 nm; (b) 325 nm. (c) TEM image in BF mode (the TEM image is used from  
 660 (Ulyashev et al. 2018) by permission of the editorial board of Vestnik of Geosciences).

661  
 662 **Table 1.** Raman spectra components of nanostructured sp<sup>2</sup> carbon substances

Nanocrystalline graphite									
Laser wavelength	633 nm			515 nm			244 nm		
Band	P	FWHM	A (%)	P	FWHM	A (%)	P	FWHM	A (%)
D	1332	49	42	1355	57	21	1458	106	3
G	1581	24	23	1584	26	25	1575	29	85
D <sub>2</sub>	1617	23	4	1622	32	5			
D <sub>4</sub> +D	2459	154	8	2454	213	9			
2D	2667	88	15	2710	93	24			
D+G	2921	130	4	2946	214	14	3125	146	12
2D <sub>2</sub>	3242	300	5	3249	22	2			
Shungite									
Laser wavelength	633 nm			515 nm			244 nm		
Band	P	FWHM	A (%)	P	FWHM	A (%)	P	FWHM	A (%)
D <sub>4</sub>	1171	85	4	1172	126	3			
D	1329	82	63	1349	71	53	1484	192	12
D <sub>3</sub>	1549	131	6	1532	105	7			
G	1598	51	17	1595	57	18	1587	39	88
D <sub>2</sub>	1622	21	1	1618	14	0			
D <sub>4</sub> +D	2482	150	1	2502	138	1			
2D	2643	191	3	2684	159	9			
D <sub>3</sub> +D	2807	100	0	2819	114	1			
D+G	2913	146	5	2937	126	6			
2D <sub>2</sub>	3183	102	1	3190	109	1			

663 *Notes:* P = band position (cm<sup>-1</sup>), FWHM = full bandwidth at half maximum (cm<sup>-1</sup>), A (%) =  
 664 relative integral intensity of the band (the ratio of the band area to the total area of the bands of

665 the spectrum), laser wavelength = the wavelength of the exciting radiation.

666

667 Table 2. Raman spectra components of ultrahigh-pressure glass-like carbon

668

Ultrahigh-pressure glass-like carbon									
Laser wavelength	633 nm			515 nm			244 nm		
Band	P	FWHM	A (%)	P	FWHM	A (%)	P	FWHM	A (%)
D <sub>4</sub>	1243	167	19	1277	194	9			
D	1344	121	23	1350	118	21	1411	329	52
D <sub>3</sub>	1531	253	34	1523	279	28			
G	1598	55	9	1600	54	13	1594	75	48
2D	2629	378	9	2662	316	12			
D <sub>3</sub> +D	2894	300	6	2919	312	16			
2D <sub>2</sub>	3181	87	0	3192	102	2			

669 *Notes:* P = band position (cm<sup>-1</sup>), FWHM = full bandwidth at half maximum (cm<sup>-1</sup>), A (%) =  
 670 relative integral intensity of the band (the ratio of the band area to the total area of the bands of  
 671 the spectrum), laser wavelength = the wavelength of the exciting radiation.

672

673

674

**Table 3.** Raman spectra components of nanostructured diamond under UV excitation (244 nm laser)

Nanocrystalline diamond (sp <sup>3</sup> -carbon) (8 samples)			
Crystallite size (nm)	Band position range (average), cm <sup>-1</sup>	FWHM range (average), cm <sup>-1</sup>	Band interpretation
30-50	1329–1332 (1330)	12–27 (18)	T <sub>2g</sub> -diamond
Ultrananocrystalline diamond (mixture of sp <sup>3</sup> - and sp <sup>2</sup> -carbons) (6 samples)			
Crystallite size (nm)	Band position range (average), cm <sup>-1</sup>	FWHM range (average), cm <sup>-1</sup>	Band interpretation
2-5	1318–1322 (1321)	41–54 (47)	T <sub>2g</sub> -diamond Raman mode
	1050	227–287 (252)	T-peak (C–C vibrations in a sp <sup>3</sup> carbon, sp <sup>3</sup> vibration density of states of ta-carbon)
	1254	217–237 (225)	Nanodiamond (small vibrational domains)
	1590	87–117 (103)	G-band of sp <sup>2</sup> carbon
	1640	169–212 (192)	G-band in DLC (ta-carbon)
	1740	45–87 (64)	C=O vibrations

675 *Notes:* n = number of studied samples, FWHM = full bandwidth at half maximum (cm<sup>-1</sup>)

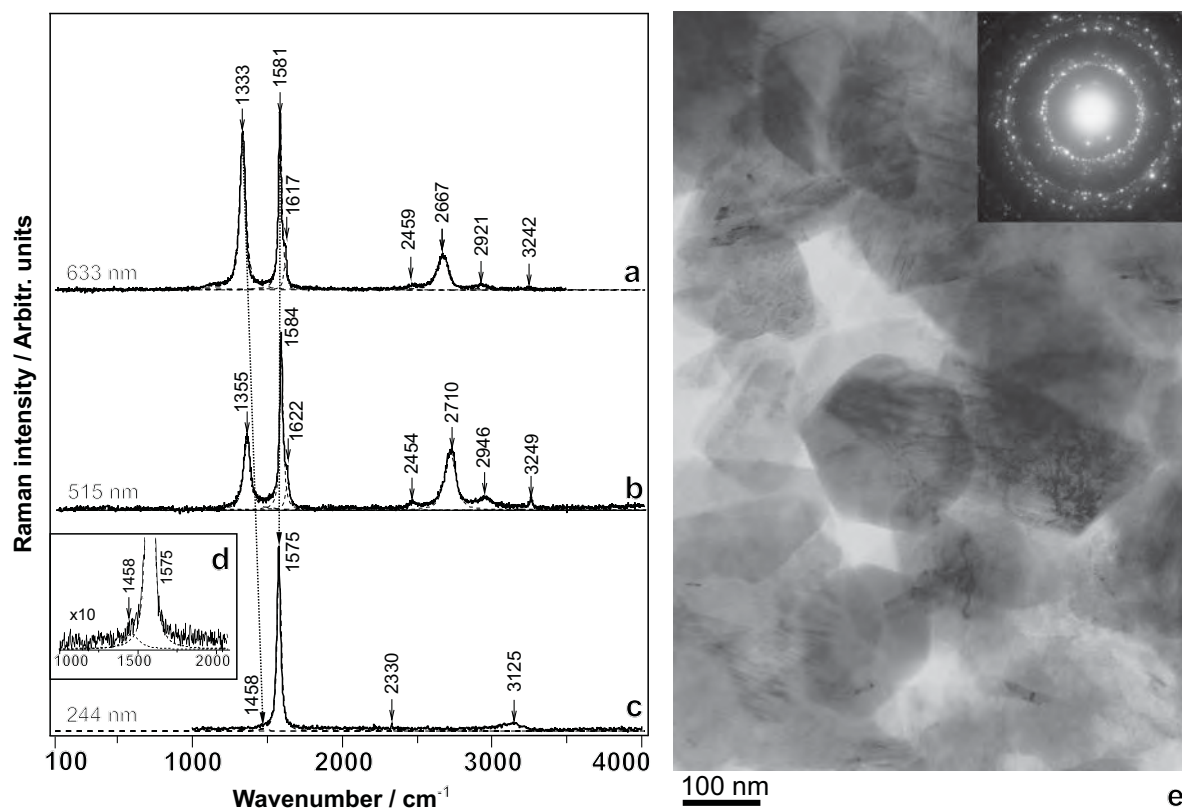
676

677 **Table 4.** Raman spectra components of polyphase carbon aggregate under visible and UV  
678 excitation

Laser wavelength	Band position (cm <sup>-1</sup> )	FWHM (cm <sup>-1</sup> )	A (%)	Band interpretation
325 nm	1325	21	8	T <sub>2g</sub> -diamond (sp <sup>3</sup> carbon)
	1415	137	38	D-band of sp <sup>2</sup> carbon
	1583	59	54	G-band of sp <sup>2</sup> carbon
514 nm	1231	217	15	D <sub>4</sub> -band of sp <sup>2</sup> carbon
	1345	83	39	D-band of sp <sup>2</sup> carbon
	1513	219	27	D <sub>3</sub> -band of sp <sup>2</sup> carbon
	1597	58	19	G-band of sp <sup>2</sup> carbon

679  
680 *Notes:* FWHM = full bandwidth at half maximum (cm<sup>-1</sup>), A (%) = relative integral intensity of  
681 the band (the ratio of the band area to the total area of the bands of the spectrum), laser  
682 wavelength = the wavelength of the exciting radiation.  
683







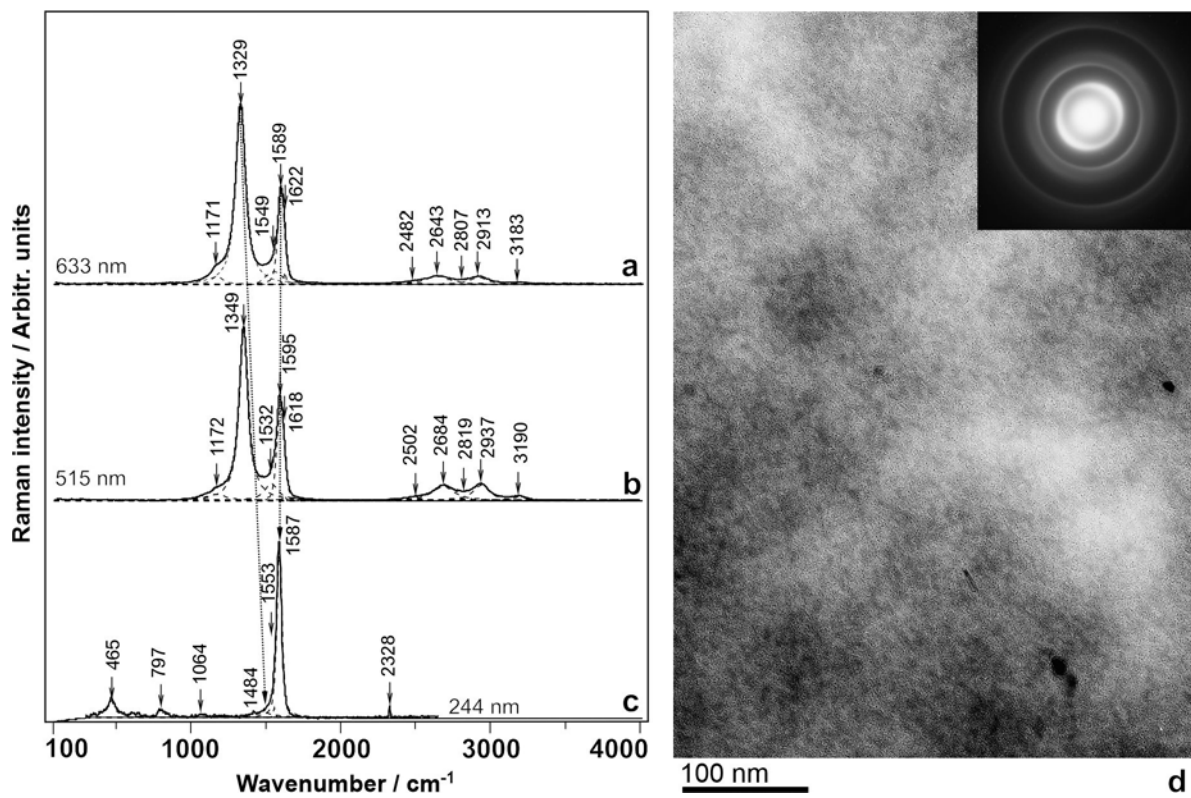


Figure 3.

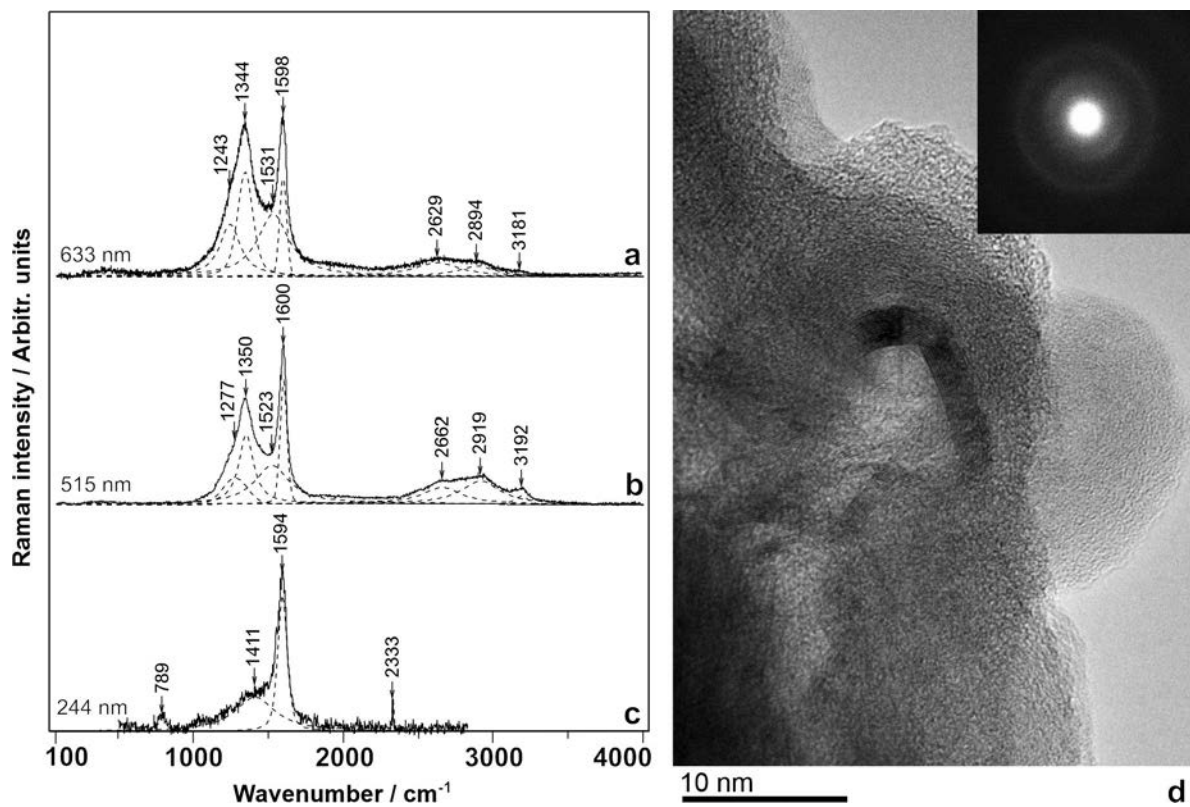


Figure 4.

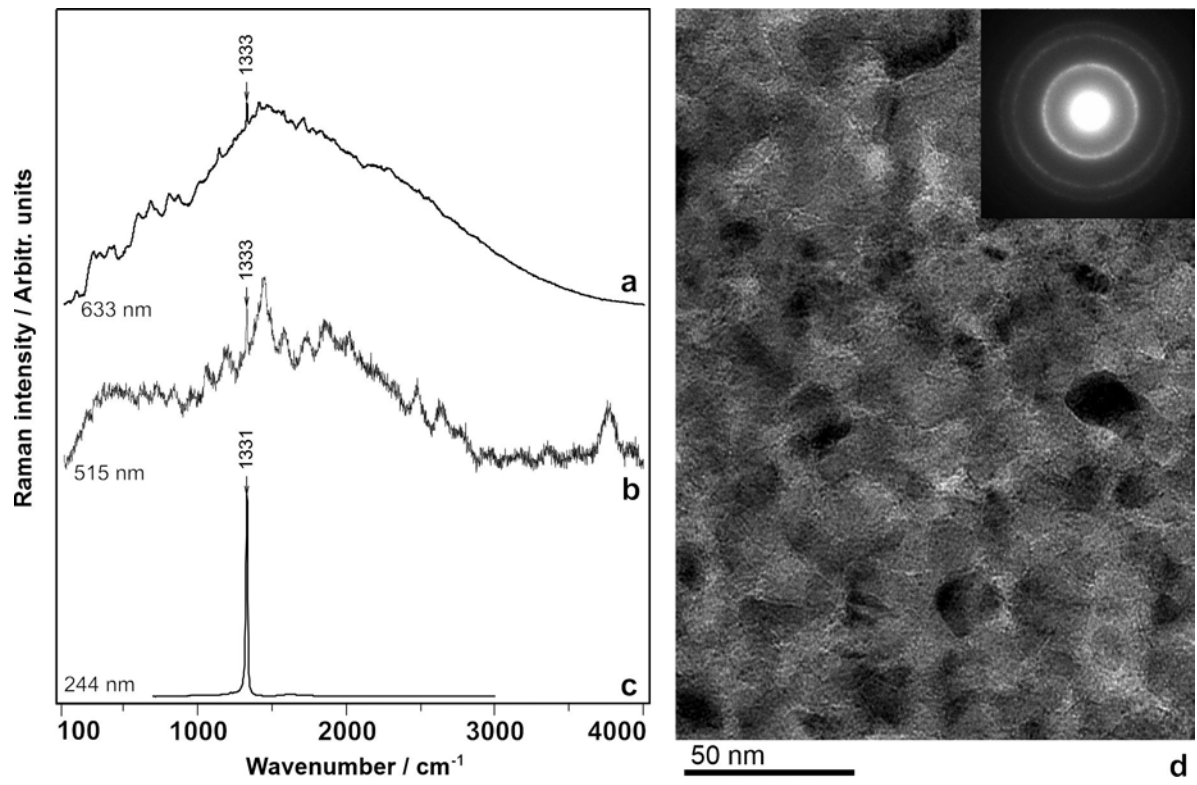


Figure 5.

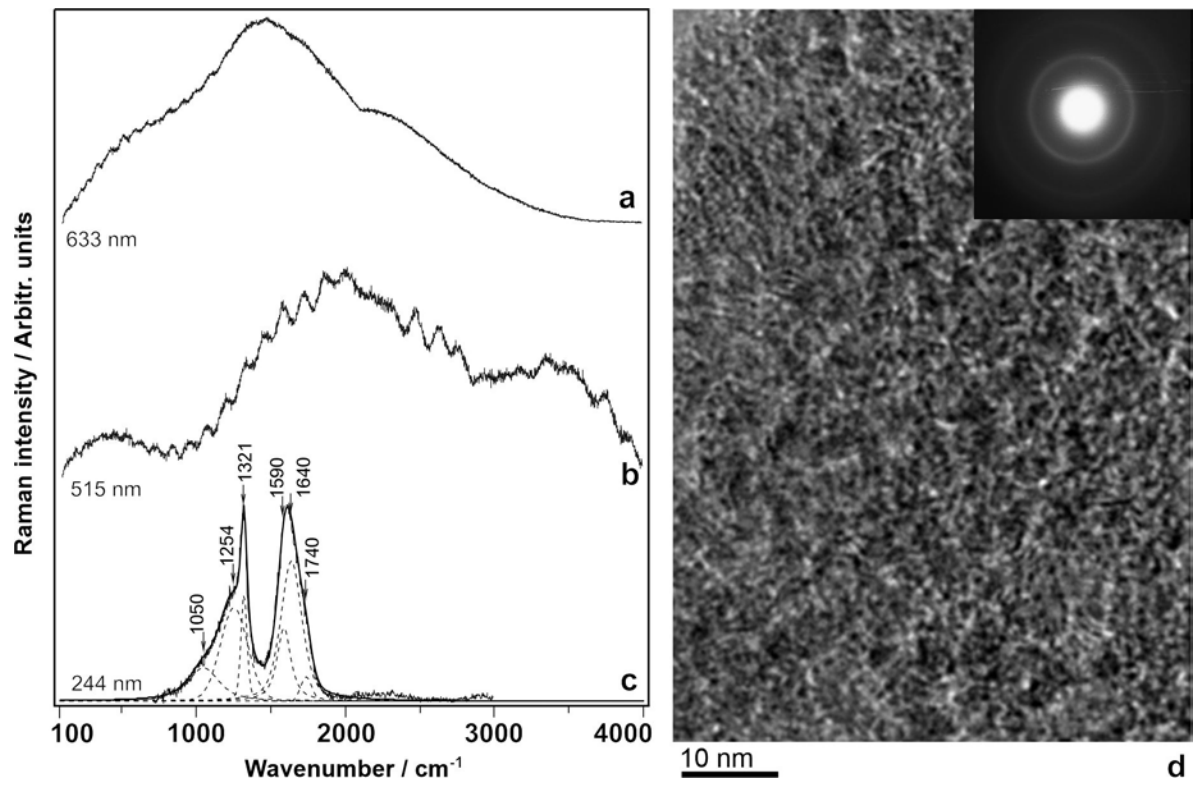


Figure 6.

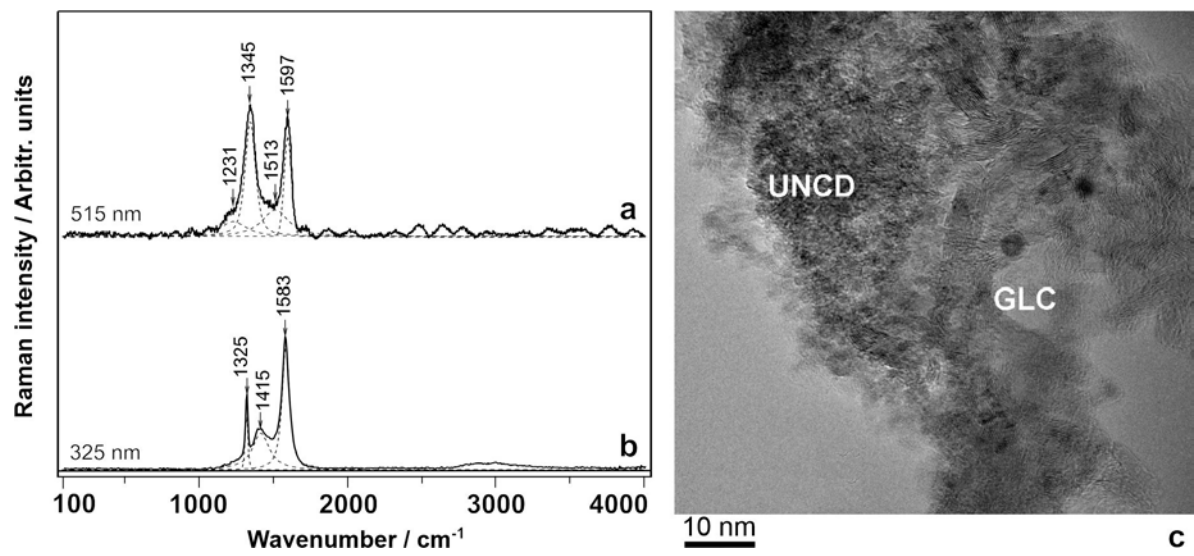


Figure 7.