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4 **High-*P-T* phase relations of Al-bearing magnetite: Post-spinel phases as**  
5 **indicators for *P-T* conditions of formation of natural samples**

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7 LAURA UENVER-THIELE<sup>1</sup>  
8 (uerver-thiele@em.uni-frankfurt.de),

9 ALAN B. WOODLAND<sup>1</sup>,

10 NOBUYOSHI MIYAJIMA<sup>2</sup>,

11 TIZIANA BOFFA BALLARAN<sup>2</sup>,

12 EDITH ALIG<sup>3</sup>,

13 LOTHAR FINK<sup>3</sup>

14

15 <sup>1</sup> Institut für Geowissenschaften, Goethe-Universität Frankfurt, Altenhöferallee 1, D - 60438

16 Frankfurt am Main, Germany

17 <sup>2</sup> Bayerisches Geoinstitut, Universität Bayreuth, D - 95440 Bayreuth, Germany

18 <sup>3</sup> Institut für Anorganische u. Analytische Chemie, Goethe-Universität Frankfurt, Max-von-

19 Laue-Str. 7, D - 60438 Frankfurt am Main, Germany

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22 inclusion in diamond, shock metamorphism

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## ABSTRACT

The phase relations of Al-bearing magnetite were investigated between 6-22 GPa and 1100-1500 °C using a multi-anvil apparatus. This study demonstrates that the spinel-structured phase persists up to ~9-10 GPa at 1100-1400 °C irrespective of the amount of hercynite (FeAl<sub>2</sub>O<sub>4</sub>) component present (20, 40 or 60 mol%). At ~10 GPa the assemblage Fe<sub>2</sub>(Al,Fe)<sub>2</sub>O<sub>5</sub> + (Al,Fe)<sub>2</sub>O<sub>3</sub> forms and remains stable up to 16-20 GPa and 1200-1550 °C. Fe<sub>2</sub>(Al,Fe)<sub>2</sub>O<sub>5</sub> adopts the CaFe<sub>3</sub>O<sub>5</sub>-type structure with the *Cmcm* space group. At 18-22 GPa and *T* > 1300 °C the assemblage Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub> + (Al,Fe)<sub>2</sub>O<sub>3</sub> becomes stable. Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub> is isostructural with Fe<sub>7</sub>O<sub>9</sub>, having the monoclinic structure of the *C2/m* space group. At *T* < 1300 °C, Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub> + (Al,Fe)<sub>2</sub>O<sub>3</sub> gives way to the assemblage of an hp-Fe(Fe,Al)<sub>2</sub>O<sub>4</sub> + (Al,Fe)<sub>2</sub>O<sub>3</sub>. This hp-Fe(Fe,Al)<sub>2</sub>O<sub>4</sub> phase is unquenchable and a defect-bearing spinel-structured phase was recovered instead; containing numerous lamellae parallel to {100} or {113} planes and notably less Al than the initial starting composition. While low-pressure spinel can have complete solid solution between Fe<sup>3+</sup>-Al, the post-spinel phases have only very limited Al solubility, with a maximum of ~0.1 cpfu Al in hp-Fe(Fe,Al)<sub>2</sub>O<sub>4</sub>, ~0.3 cpfu in Fe<sub>2</sub>(Fe,Al)<sub>2</sub>O<sub>5</sub> and ~0.4 cpfu in Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub>, respectively. As a result, the phase relations of Fe(Fe<sub>0.8</sub>Al<sub>0.2</sub>)<sub>2</sub>O<sub>4</sub> can also be applied to bulk compositions richer in Al with the only difference being that larger amounts of a (Al,Fe)<sub>2</sub>O<sub>3</sub> phase are present.

Co-existing rhombohedral-structured phases demonstrate that the binary miscibility gap established at low pressure between hematite and corundum is still valid up to 20 GPa. Since iron oxides (e.g. magnetite) with variable Al contents are found in extraterrestrial rocks or as inclusions in diamond, constraints on their high-*P-T-fO*<sub>2</sub> stability might help to unravel their formation conditions.

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## INTRODUCTION

48 Along with iron, aluminum is another notable constituent of many oxides, hydroxides, as well  
49 as silicate phases. For example, spinel group minerals are well known as important carriers  
50 for Al, along with Fe. Such minerals occur widely in the Earth's mantle and crust and are also  
51 found as accessory phases in extraterrestrial rocks (e.g. Busche et al. 1971, 1972, Keil 2012;  
52 Krot, 2019). In addition to end-member compositions spinel ( $\text{MgAl}_2\text{O}_4$ ), hercynite ( $\text{FeAl}_2\text{O}_4$ ),  
53 chromite ( $\text{FeCr}_2\text{O}_4$ ), and magnetite ( $\text{Fe}^{2+}\text{Fe}^{3+}_2\text{O}_4$ ), spinel-structured phases are known for  
54 their ability to form a variety of solid solutions (e.g.  $\text{MgFe}_2\text{O}_4$ - $\text{Fe}_3\text{O}_4$ , Katayama and Iseda  
55 2002;  $\text{FeAl}_2\text{O}_4$ - $\text{Fe}_3\text{O}_4$ , Turnock and Eugster 1962). Of all the possible major constituents,  
56 iron, as a multi-valent element is sensitive to the prevailing redox conditions and thus can  
57 give insights into the local oxidation state within the Earth's interior. Although magnetite  
58 contains a significant amount of  $\text{Fe}^{3+}$  ( $\text{Fe}^{3+}/\text{Fe}_{\text{tot}}=2/3$ ), the incorporation of other trivalent  
59 cations such as Al or Cr can lower the  $\text{Fe}^{3+}$  content and thus stabilize this phase over a larger  
60 range in oxidation states.

61 Previous experimental studies on the phase relations of spinel-structured oxides have  
62 demonstrated that their stability is often restricted to pressures corresponding to the upper  
63 mantle (e.g. Schollenbruch et al. 2011; Woodland et al. 2012; Uenver-Thiele et al. 2017a,  
64 2017b; Ishii et al. 2014, 2015). At certain high-pressure (high- $P$ ) and high-temperature (high- $T$ )  
65 conditions the spinel-structured phase can i) break down into its constituent oxides (e.g.  
66  $\text{FeAl}_2\text{O}_4$ , Schollenbruch et al. 2011;  $\text{MgAl}_2\text{O}_4$ , Akaogi et al. 1999), ii) transform into a high- $P$   
67 polymorph (hp- $\text{O}_4$ ; e.g. Irfune et al. 1991, 2002; Funamori et al. 1998; Fei et al. 1999; Haavik  
68 et al. 2000; Levy et al. 2004; Chen et al. 2003a; Enomoto et al. 2009; Yong et al. 2012), or iii)  
69 produce an assemblage involving post-spinel phases with different stoichiometries [e.g.  
70  $\text{Fe}_4\text{O}_5$ , Lavina et al. 2011, Woodland et al. 2012;  $\text{Fe}_5\text{O}_6$ , Lavina and Meng 2015, Woodland et  
71 al. 2015, 2023;  $\text{Fe}_7\text{O}_9$ , Sinmyo et al. 2016;  $\text{Mg}_2\text{Al}_2\text{O}_5$ , Enomoto et al. 2009;  $\text{Mg}_2\text{Fe}_2\text{O}_5$ , Boffa  
72 Ballaran et al. 2015;  $(\text{Mg,Fe})_2\text{Fe}_2\text{O}_5$  and  $(\text{Fe,Mg})_2\text{Fe}_4\text{O}_9$ , Uenver-Thiele et al. 2017a, 2017b,

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73 2018; Fe<sub>9</sub>O<sub>11</sub>, Ishii et al. 2018]. In this respect, knowing the phase relations for the chemically  
74 different post-spinel phases/assemblages can help to constrain the formation conditions and  
75 evolution of certain high-pressure samples from the Earth's mantle, from shock-  
76 metamorphosed terrestrial rocks as well as from extraterrestrial samples, forming a kind of  
77 "oxide scale" as suggested by Zhang (2017). For instance, the experimentally determined  
78 stability fields of different post-spinel assemblages in the MgO-FeO-Fe<sub>2</sub>O<sub>3</sub> system over a  
79 range in *P* and *T* allowed Uenver-Thiele et al. (2017b) to add constraints on the petrological  
80 history of diamonds with inclusions of magnesioferrite (MgFe<sub>2</sub>O<sub>4</sub>). Moreover,  
81 magnesioferrite exsolutions also have been found in a number of ferropicriase (Mg,Fe)O  
82 inclusions in diamond independent of the Fe content in the ferropicriase (e.g. McCammon et  
83 al. 1998, Harte et al. 1999, Wirth et al. 2014, Kaminsky et al. 2015, Palot et al. 2016, Anzolini  
84 et al. 2019, Sharygin et al. 2021; Lorenzon et al. 2023). Such exsolution from ferropicriase  
85 may occur during ascent from the upper or lower mantle, or it may result from a back-reaction  
86 from a high-*P* precursor phase (e.g. Wirth et al. 2014, Anzolini et al. 2020). Identifying a  
87 high-*P* precursor phase often requires detailed investigation of the micro-textures of the  
88 "magnesioferrite" crystals (e.g. Jacob et al. 2016, Anzolini et al. 2020), since Fe-Mg post-  
89 spinel oxide phases can suffer from retrograde reaction during exhumation and transform or  
90 break down into a spinel-structured phase.

91 To the best of our knowledge, the first report of naturally preserved hp-phases was by Chen et  
92 al. (2003a, 2003b) who identified two high-*P* polymorphs of chromite with the CaFe<sub>2</sub>O<sub>4</sub>-type  
93 (CF; named "Chenmingite", IMA 2017-036, Ma and Tschauner 2017) and CaTi<sub>2</sub>O<sub>4</sub>-type (CT;  
94 named "Xieite", IMA 2007-056, Chen et al. 2008) structure preserved in the shocked Suizhou  
95 meteorite. Chen et al. (2003a, 2003b) demonstrated that these two hp-polymorphs must have  
96 formed under shock conditions of ~20-23 GPa and 1800-2000 °C. The CF-FeCr<sub>2</sub>O<sub>4</sub>  
97 (Chenmingite) was also reported from the highly shocked Tissint Martian meteorite and is

98 interpreted to have formed above 16-18 GPa and temperatures below  $\sim 1350$  °C (Ma et al.  
99 2019). In addition, occurrences of preserved post-spinel phases have been also reported in  
100 terrestrial samples, including in a shocked gneiss from the Xiuyan impact crater (CF-type  
101 structured hp-MgFe<sub>2</sub>O<sub>4</sub>, named Maohokite, IMA 2017-047; Chen et al. 2019) or as inclusions  
102 in diamond from the Juina area, Mato Grosso State, Brazil (CaTi<sub>2</sub>O<sub>4</sub>-type structured Mg-Cr-  
103 Fe oxide and orthorhombic hp-CaCr<sub>2</sub>O<sub>4</sub>, Kaminsky et al. 2015; Maohokite, Agrosi et al.  
104 2019). Pressure-temperature conditions of the Xiuyan impact event were estimated to be 25-  
105 45 GPa and 800-900 °C (Chen et al. 2019), while the inclusions in diamond constrained  
106 diamond formation to have occurred in the deep transition zone or lower mantle ( $\geq 18$  GPa;  
107 Kaminsky et al. 2015, Agrosi et al. 2019). Note that all these *P-T* constraints were based upon  
108 experimental data.

109 Considering that high-*P* spinels occurring in both terrestrial and extraterrestrial rocks often  
110 contain significant amounts of magnetite and hercynite components (e.g. as inclusions in  
111 diamond: Newhouse & Glass 1936, Hayman et al. 2005; Stachel and Harris 2008; Kaminsky  
112 et al. 2009, Wirth et al. 2014; Palot et al. 2016, Smith et al. 2018; in extraterrestrial rocks: e.g.  
113 Keil 2012, Busche et al. 1972), knowledge about their high-*P* and high-*T* phase relations  
114 allows us to better constrain the physico-chemical conditions of their formation, as well as  
115 that of their host material. However, to date only experimental data on the phase relations of  
116 the end-member FeAl<sub>2</sub>O<sub>4</sub> and FeFe<sub>2</sub>O<sub>4</sub> are available and these relations are in fact quite  
117 contrasting (Schollenbruch et al. 2011, Woodland et al. 2012). Therefore, experimental  
118 investigations of solid solutions along the FeFe<sub>2</sub>O<sub>4</sub>-FeAl<sub>2</sub>O<sub>4</sub> binary join are very important for  
119 understanding high-*P* phase stabilities in natural oxide samples containing both Fe<sup>3+</sup> and Al.  
120 Here, we report the results of an experimental study on how the presence of Al affects the  
121 phase relations in Fe(Fe,Al)<sub>2</sub>O<sub>4</sub>. Since natural samples of high-*P* origin mostly have relatively  
122 low Al contents, we focussed on a Fe<sup>3+</sup>-rich bulk composition (80 mol% Fe<sub>3</sub>O<sub>4</sub> – 20 mol%

123  $\text{FeAl}_2\text{O}_4$ ), although a number of reconnaissance experiments with higher Al-contents (i.e. 40  
124 mol% or 60 mol%  $\text{FeAl}_2\text{O}_4$ ) or with a different stoichiometry of  $\text{Fe}^{2+}_3\text{Fe}^{3+}_3\text{AlO}_9$  were also  
125 performed.

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## EXPERIMENTAL METHODS

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### 129 **Starting materials**

130 Three different starting mixtures with compositions (i) 80 mol%  $\text{Fe}_3\text{O}_4$  – 20 mol%  $\text{FeAl}_2\text{O}_4$ ,  
131 (ii) 60 mol%  $\text{Fe}_3\text{O}_4$  – 40 mol%  $\text{FeAl}_2\text{O}_4$  and (iii) 40 mol%  $\text{Fe}_3\text{O}_4$  – 60 mol%  $\text{FeAl}_2\text{O}_4$  were  
132 employed in our high- $P$  experiments. A further starting composition of  $\text{Fe}^{2+}_3\text{Fe}^{3+}_3\text{AlO}_9$   
133 stoichiometry was employed to directly test for the extent of Al solubility in the post-spinel  
134 phase  $\text{Fe}_7\text{O}_9$ . Magnetite ( $\text{FeFe}_2\text{O}_4$ ) and hercynite ( $\text{FeAl}_2\text{O}_4$ ) were both pre-synthesised in a  
135 CO-CO<sub>2</sub> gas-mixing furnace at 1 atm under controlled  $f\text{O}_2$ . Magnetite was produced by  
136 reducing hematite at 1300 °C and  $\log f\text{O}_2 = -5.5$ , which should yield an essentially  
137 stoichiometric composition (Diekmann 1982). For the synthesis of hercynite, a stoichiometric  
138 mixture of  $\text{Al}_2\text{O}_3$  and  $\text{Fe}_2\text{O}_3$  was ground together, pressed into pellets and equilibrated at 1300  
139 °C. Initially, the oxygen fugacity was set at  $\log f\text{O}_2 = -4$  in order to first produce a mixture of  
140 wüstite coexisting with a hercynite-magnetite solid solution. After 6 h, the sample was  
141 quenched in water, reground and pressed into pellets for further sintering for 5 h under the  
142 same conditions. This procedure was repeated for additional two cycles, but with a stepwise  
143 shift to more reducing conditions ( $\log f\text{O}_2 = -9$  for 4 h;  $\log f\text{O}_2 = -11.4$  for 1 h) to ensure all of  
144 the iron is  $\text{Fe}^{2+}$  in the final step. A direct synthesis at low  $f\text{O}_2$  tends to produce a mixture of  
145 metallic Fe and corundum, which then does not react efficiently to hercynite. Several crystals  
146 of magnetite and hercynite were checked for homogeneity and chemical composition by  
147 electron microprobe (EPMA). Analysis of X-ray powder diffraction patterns gave a unit-cell  
148 parameter for magnetite and hercynite of  $a_0 = 8.3966(6)$  Å or  $a_0 = 8.1502(1)$  Å, respectively.

149 The starting material of  $\text{Fe}^{2+}_3\text{Fe}^{3+}_3\text{AlO}_9$  was composed of a stoichiometric mixture of  
150  $\text{FeAl}_2\text{O}_4$ ,  $\text{FeFe}_2\text{O}_4$ ,  $\text{Fe}_2\text{O}_3$  and  $\text{Fe}^0$  in a 1:2:2:1 molar ratio. To prevent adsorption of moisture,  
151 the starting materials were stored in a desiccator.

152

### 153 **Experimental procedure**

154 High-*P* and high-*T* experiments were performed at the Goethe Universität Frankfurt and the  
155 Bayerisches Geoinstitut in Bayreuth over a *P-T* range of 6-22 GPa and 1200-1600 °C (see  
156 Table 1). Experimental run conditions and run products are listed in Table 1. Experiments up  
157 to 14 GPa and some at 18 GPa were performed in Frankfurt using an 800t Walker-type multi-  
158 anvil apparatus (Walker et al. 1990). Experiments carried out at the Bayerisches Geoinstitut  
159 Bayreuth were conducted with 500t, 1000t or 5000t split-sphere Kawai-type multi-anvil  
160 presses (Kawai and Endo 1970). Mineral phase transitions (e.g.  $\alpha\text{-Mg}_2\text{SiO}_4\text{-}\beta\text{-Mg}_2\text{SiO}_4$ ,  $\beta\text{-}$   
161  $\text{Mg}_2\text{SiO}_4\text{-}\gamma\text{-Mg}_2\text{SiO}_4$ , coesite-stishovite,  $\text{CaGeO}_3$  garnet-perovskite,  $\text{MgSiO}_3$  ilmenite-  
162 perovskite) were used for pressure calibration with details described in Brey et al. (2008) and  
163 Keppler and Frost (2005). Different assembly sizes were used depending on the desired  
164 pressure. Up to 14 GPa,  $\text{Cr}_2\text{O}_3$ -doped MgO octahedra were employed along with tungsten  
165 carbide (WC) cubes having 8 mm truncation edges (so-called 14/8 assembly). Higher-  
166 pressure experiments were conducted using 10/4, 10/5, 18/8 or 18/11 assemblies/truncation  
167 lengths. The pressure cells in Frankfurt have a Re-foil heater, whereas  $\text{LaCrO}_3$  was employed  
168 as a heater in the experiments at the Bayerisches Geoinstitut. The heater was placed inside  
169 either in a  $\text{ZrO}_2$  or a  $\text{LaCrO}_3$  (only for 10/4 experiments with Re-foil heater) sleeve, which  
170 acts as a thermal insulator. While the Re-foil had direct contact with the WC cubes, a  
171 molybdenum ring and disc was inserted at the top or the bottom of the  $\text{LaCrO}_3$  heater to  
172 ensure contact to the WC cubes. The sample capsule/s and thermocouple were surrounded by  
173 MgO sleeves to avoid direct contact with the furnace. The starting material was usually  
174 loaded into Ag-foil capsules, however, several experiments employed Pt-foil when the desired

175 temperature exceeded the melting point of Ag. Capsule dimensions depended on the  
176 particular assembly employed with 1.2 to 2 mm diameter and 1.2 to 1.7 mm length. Either a  
177  $W_5/Re_{95}-W_{26}/Re_{74}$  thermocouple (800t multi-anvil press) or a  $W_3/Re_{97}-W_{25}/Re_{75}$   
178 thermocouple (1000t, 5000t multi-anvil press) was inserted axially from the top of the  
179 octahedron to monitor the temperature, with the electromotive force uncorrected for pressure.  
180 Further details of the experimental setup are described in Brey et al. (2008) and Keppler and  
181 Frost (2005). Uncertainties in pressure and temperature are  $\pm 0.5$  GPa and  $\pm 30-50$  °C,  
182 respectively (Keppler and Frost 2005). The experiments were conducted first by cold  
183 pressurization, followed by heating to the desired temperature at a rate of 50 °C/min. The  
184 experiment was terminated by turning off the power while maintaining the pressure.  
185 Depending on the experimental  $P-T$  conditions, the duration varied from 1 to 3 hrs (see Table  
186 1). After quenching, decompression was immediately initiated.

187

#### 188 **Analytical methods**

189 The recovered samples were analysed with electron microprobe (EPMA), powder X-ray  
190 diffraction and/or transmission electron microscope (TEM). For chemical analyses, several  
191 fragments of the recovered sample were mounted in epoxy, polished and carbon-coated.  
192 Measurements were carried out with a five-spectrometer JEOL JXA-8530F plus Hyperprobe  
193 at the University of Frankfurt operating in the wavelength-dispersive mode with an  
194 acceleration voltage of 15 kV, probe current of 20 nA and a spot size of 1  $\mu$ m. Pure  $Fe_2O_3$ ,  
195  $Al_2O_3$  and Pt metal were employed as primary standards. Integration times for Al were 20 s  
196 on the peak and 10 s on the background while 40 s on the peak and 20 s on the background  
197 were adjusted for Fe and Pt. A CITZAF algorithm was employed for the matrix correction  
198 (Armstrong 1993). For investigating microtextures and verifying the homogeneity of the  
199 mineral grains, backscatter electron images were taken (e.g. Fig. 1a-d).



200 Further phase identification by X-ray diffraction was performed using an STOE Stadi P  
201 diffractometer equipped with a linear PSD or a Mythen detector and a Ge(111)  
202 monochromator at the University of Frankfurt operating at 45 kV and 35 mA and using MoK $\alpha$   
203 ( $\lambda=0.70926$  Å) radiation. Recovered sample material and small amounts of silicon which  
204 served as internal standard were ground together and mounted in a 0.5 mm diameter glass  
205 capillary. Measurements were performed in Debye-Scherrer mode in transmission geometry  
206 between 1-100° 2 $\theta$ . Determination of the unit-cell parameters was obtained by full-pattern  
207 refinement using the general structure analysis system (GSAS; Larson and von Dreele 1994)  
208 software package and the EXPGUI interface of Toby (2001).

209 Two samples (S7316; Z1953o) had (i) not enough sample material for powder X-ray  
210 diffraction and/or (ii) included phases with breakdown textures so that TEM analyses were  
211 necessary. Those run products were investigated using a FEI Titan G2 80-200 S/TEM  
212 equipped with 4 SDD energy-dispersive X-ray spectrometers (EDS) and a Philips CM20FEG  
213 with a pure Ge EDS, operating at 200 kV at the Bayerisches Geoinstitut Bayreuth (Germany).

214 Some of the recovered sample material was cut and polished to make a thin section. The thin  
215 section was placed on a 3-mm-sized Mo grid and thinned to electron transparency by Ar-ion  
216 milling at accelerating voltages of 3.5 kV and an angle of incident of 8° using a precision ion  
217 polishing system (Gatan, model 691). The sample foil was carbon-coated to reduce charging  
218 and then investigated by imaging microtextures, selected area electron diffraction (SAED)  
219 and energy-dispersive X-ray (EDX) analyses. In addition, a part of the recovered material of  
220 sample Z1953o was polished on the surface and thinned to electron transparency by using a  
221 focused Ga ion beam (FIB) milling machine (FEI, Scios Dual Beam system) at the  
222 Bayerisches Geoinstitut Bayreuth (Germany).

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## RESULTS AND DISCUSSION

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## 226 **Experimental run products**

227 The resulting phase assemblages, as well as chemical compositions and lattice parameters of  
228 the individual phases, are summarized in Tables 1, 2 and 3. A total of 20 experiments with the  
229  $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.8}\text{Al}_{0.2})_2\text{O}_4$  starting composition were performed over a  $P$ - $T$  range of 6-22 GPa and  
230 1100-1550 °C. A limited number of experiments were performed with higher Al contents  
231 corresponding to stoichiometries of  $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.6}\text{Al}_{0.4})_2\text{O}_4$  and  $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.4}\text{Al}_{0.6})_2\text{O}_4$  (Table 1).  
232 These experiments were performed to clarify the extent of Al solubility in the high- $P$  post-  
233 spinel phases. Following the identification of a phase with  $(\text{Fe}^{2+})_3(\text{Fe}^{3+},\text{Al})_4\text{O}_9$  stoichiometry  
234 ( $\text{O}_9$ -phase; see text below), two further experiments with a starting composition of  
235  $\text{Fe}^{2+}_2\text{Fe}^{3+}_2\text{AlO}_9$  were specifically performed to investigate the maximum amount of Al that  
236 can be incorporated into this phase.

237 Backscattered electron (BSE) imaging of the run products confirms noticeable grain growth  
238 (2-60  $\mu\text{m}$  in size) with complete recrystallization into chemically homogenous grains. Straight  
239 grain boundaries and triple junctions often with grain boundary angles of  $\sim 120^\circ$  in the  
240 polycrystalline samples indicate that an “equilibrium-fabric” was reached during the  
241 experiments (Fig. 1a-d; Passchier and Trouw, 2005). The apparent even distribution of the  
242 newly formed phases (Fig. 1), indicates that the starting material was well homogenized. One  
243 exception is sample Z1953o, which includes an additional phase that exhibits internal textures  
244 of very fine-grained crystals (Fig. 1d) indicating that this phase dissociated during quench  
245 (unquenchable phase; named hereafter UQ-phase). The  $\text{Fe}^{2+}$ -bearing oxide phases always  
246 form the largest grains while interstitial corundum-hematite solid solutions are much smaller  
247 in size.

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## 249 **Phase relations of $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.8}\text{Al}_{0.2})_2\text{O}_4$**

250 ***Breakdown into  $Fe_2(Fe,Al)_2O_5 + (Al,Fe)_2O_3$***

251 Low-pressure experiments demonstrate that the stability of the spinel-structured phase persists  
252 up to about 9-10 GPa at temperatures of 1200-1400 °C (Fig. 2). In most cases, this phase has  
253 a composition similar to that of the initial starting mixture (see Table 2). At higher pressure,  
254 the spinel-structured phase breaks down to a multi-phase assemblage as indicated by BSE  
255 imaging and X-ray powder diffraction. Analysis of diffraction patterns (Online Materials Fig.  
256 OM1) reveals an  $A_2B_2O_5$  phase with a  $CaFe_3O_5$ -type structure (space group  $Cmcm$ , Lavina et  
257 al. 2011; Boffa Ballaran et al. 2015) coexisting with one or two corundum-type structured  
258 phases (space group  $R-3c$ ) having  $(Fe,Al)_2O_3$  stoichiometry. The  $O_5$ -structured phase is  
259 equivalent to those found in the  $Fe^{2+}$ - $Fe^{3+}$  and  $Mg$ - $Fe^{3+}$  endmember systems (e.g. Woodland  
260 et al. 2012; Uenver-Thiele et al. 2017a, 2017b). Siderite appeared in some experiments in  
261 trace amounts, barely detectable in most powder diffraction patterns. We attribute its presence  
262 to the reaction of adsorbed  $CO_2$  with the fine-grained starting material powders, which  
263 implies minor reduction of  $Fe^{3+}$  to  $Fe^{2+}$  during the experiment. In experiment M815, tiny  
264 amounts of a  $Fe_5O_6$  structured phase (Lavina and Meng 2015; Woodland et al. 2015, 2023)  
265 were identified in the powder XRD pattern. Its presence might be also attributed to a small  
266 degree of reduction during the experiment.

267 In terms of composition, the  $O_5$ -phase contains relatively low amounts of Al, with only up to  
268 0.3 cpdf present, corresponding to 15 mol%  $Fe_2Al_2O_5$  component (Table 2). Thus, this phase  
269 incorporated proportionally less Al than present in the starting material (i.e.  $(Al/(Al+Fe^{3+})) =$   
270 0.15 compared to 0.2 in the starting composition). The Al poorer and  $Fe^{2+}$  richer  
271 stoichiometry of the  $O_5$ -phase compared to the starting bulk composition causes an excess of  
272 trivalent cations (Al and  $Fe^{3+}$ ) that stabilises the coexisting corundum-hematite solid solutions  
273 (hereafter referred to as  $cor_{ss}$  or  $hem_{ss}$  depending on the Al/Fe ratio, Table 1). In many  
274 experiments two separate phases were present, indicating that the large miscibility gap along

275 the  $\text{Al}_2\text{O}_3\text{-Fe}_2\text{O}_3$  binary documented by Feenstra et al. (2005) at <4.0 GPa and 1300 °C  
276 persists to at least 18 GPa (Table 1). For instance, this is manifested in the coexistence of  
277 hematite with up to 23.5 mol%  $\text{Al}_2\text{O}_3$  ( $\text{hem}_{\text{ss}}$ ) and corundum with up to 10.5 mol%  $\text{Fe}_2\text{O}_3$   
278 ( $\text{cor}_{\text{ss}}$ ) in many of our experiments (see Table 1 and Table 3).

279

### 280 *Stability of $\text{Fe}^{2+}_3(\text{Fe}^{3+},\text{Al})_4\text{O}_9 + (\text{Al},\text{Fe})_2\text{O}_3$*

281 In the  $P$ - $T$  range of 16-20 GPa and 1200-1550 °C, the maximum stability of the  $\text{O}_5$ -phase is  
282 reached and above such pressures another phase assemblage becomes stable (Figs. 1 and 2).  
283 The XRD patterns of run products from four experiments (Z2040u, M841, H5668, H5745)  
284 indicate the presence of a  $\text{Fe}_7\text{O}_9$ -structured phase (Sinmyo et al. 2016) (see Fig. 1b, Online  
285 Materials Fig. OM2, and Table 1). In experiment Z2040u and H5668,  $\text{Fe}_3(\text{Al},\text{Fe})_4\text{O}_9$   
286 represents the majority phase together with  $\text{hem}_{\text{ss}}$  and/or  $\text{cor}_{\text{ss}}$  (Fig. 1b). The sample is well  
287 crystallized with no apparent reaction zones (Fig. 1b). Coexisting  $\text{cor}_{\text{ss}}$ , also occurring as  
288 inclusions in the  $\text{O}_9$ -phase indicate that the  $\text{O}_9$ -phase was saturated in Al. Thus, under the  
289 conditions of our experiments, it appears that only up to 10.5 mol % of a  $\text{Fe}_3\text{Al}_4\text{O}_9$  component  
290 can be incorporated into  $\text{Fe}_7\text{O}_9$ .

291 The appearance of  $\text{Fe}_3(\text{Fe},\text{Al})_4\text{O}_9$  is notable since no  $\text{O}_9$ -phase has been observed in the phase  
292 relations of either the  $\text{Fe}_3\text{O}_4$  or  $\text{FeAl}_2\text{O}_4$  endmembers (e.g. Woodland et al. 2012,  
293 Schollenbruch et al. 2011). However, it is likely that further new studies of different spinel  
294 compositions will also reveal the stability of an  $\text{O}_9$ -phase. For example, Uenver-Thiele et al.  
295 (2017b) reported the existence of  $\text{Mg}_3\text{Fe}_4\text{O}_9$  and  $\text{Mg}_{1.5}\text{Fe}_{1.5}\text{Fe}_4\text{O}_9$  in the  $\text{MgFe}_2\text{O}_4$  and  
296  $\text{Mg}_{0.5}\text{Fe}^{2+}_{0.5}\text{Fe}_2\text{O}_4$  systems, respectively. More recently, Ishii et al. (2020) also studied the  
297 high- $T$  and high- $P$  phase relations of  $\text{MgFe}_2\text{O}_4$ . Their study confirmed the existence of an  $\text{O}_9$ -  
298 phase + hem assemblage, but at a somewhat different  $P$ - $T$  range (Ishii et al. 2020) than  
299 reported earlier. In any case, both studies of Mg-bearing compositions indicate that an  $\text{O}_9$ -  
300 structured phase becomes stable at pressures beyond the stability of an  $\text{M}_4\text{O}_5 + \text{hem}$

301 assemblage and its stability appears to be limited to a narrow range in  $P$ - $T$  space. Here, in Al-  
302 bearing compositions, we also find that the stability of  $\text{Fe}_3(\text{Fe,Al})_4\text{O}_9$  is limited to a narrow  
303 range in  $P$  and  $T$  beginning at  $\sim 18$  GPa (at 1300-1400 °C) and widens to higher  $P$  and  $T$  (at 20  
304 GPa  $>1300$  °C but  $<1550$  °C; see Fig. 2).

305

### 306 *Appearance of a hp-polymorph*

307 As previously described,  $\text{Fe}_3(\text{Fe,Al})_4\text{O}_9$  becomes stable at  $P \geq 18$  GPa and  $T \geq 1300$  °C. At 18  
308 GPa and 1300 °C (M841) run products include an  $\text{O}_9$ -phase together with  $\text{cor}_{\text{ss}}$  as well as an  
309 additional phase. At lower  $T$  the  $\text{O}_9$ -phase completely disappears. Analysis of the XRD  
310 patterns suggests that a spinel-structured phase with a lattice parameter of  $a = 8.3696(4) -$   
311  $8.3830(1)$  Å is present. However, low-pressure experiments indicate that the spinel-structured  
312  $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.8}\text{Al}_{0.2})_2\text{O}_4$  phase has its maximum pressure stability at 9-11 GPa with a lattice  
313 parameter of  $a = 8.3379(1)$  Å (e.g. sample M840; Table 1, Fig. 2). Chemical analyses by  
314 EPMA and TEM (see Table 2) reveal that this phase (at  $\geq 18$  GPa) has an extremely low Al  
315 content, approaching an almost pure magnetite composition [ $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.95}\text{Al}_{0.05})_2\text{O}_4$ ], which is  
316 consistent with the observed lattice parameter. However, since experimental studies have  
317 demonstrated that magnetite is also only stable up to 9-10 GPa (e.g. Woodland et al. 2012,  
318 Fig. 2) this phase in experiments at  $\geq 18$  GPa must be a product of a retrograde back-reaction  
319 from a high- $P$  phase during quenching. To clarify the presence or even the nature of the  
320 precursor phase of the magnetite, sample S7316 and Z1953o were investigated by TEM, with  
321 sample Z1953o being discussed in more detail since it contains not only a spinel-structured  
322 phase but also an UQ-phase (see Figs. 1d and 3a). Striations within this magnetite-rich spinel  
323 observed by BSE imaging were investigated in detail and proved to be numerous lamellae  
324 (Fig. 3). Some of these lamellae have an orientation corresponding to the  $\{113\}$  plane in the  
325 magnetite (Fig. 3c), which is consistent with the orientation of  $\{113\}$  twins. The twinning

326 along {113} usually reflects a back reaction from a galena-like archetype structure belonging  
327 to the lillianite homologies series (Makovicky 1977) and as such may result from the back-  
328 reaction of a hp-phase like Fe<sub>4</sub>O<sub>5</sub> or Fe<sub>5</sub>O<sub>6</sub> (Schollenbruch et al. 2011; Woodland et al. 2012,  
329 2023; Myhill et al. 2016). Since the lamellae within the magnetite of sample Z1953o are not  
330 only on the {113} plane but also on the {100} plane and Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub> is quenchable without  
331 any retrograde microtextures, we interpreted this defect magnetite (dmt) phase to be indicative  
332 of a previous high-*P* orthorhombic O<sub>4</sub>-polymorph (e.g., Haavik et al. 2000). However,  
333 although the {100} and {113} lamellae seem to have twin-like features, their twin law  
334 remains to be determined and statistically proven, and the stability of this hp-polymorph at *P*  
335 and *T* needs to be verified by in-situ high-*P-T* experiments.

336 Detailed TEM observations of the UQ-phase confirm that it results from the dissociation into  
337 two phases (Fig. 4) during quench. The new nanometer-sized crystals are vermicularly  
338 intergrown, a texture that is commonly known for exsolution processes within minerals (e.g.  
339 feldspar). Moreover, TEM observations indicate that the crystals are in an epitaxial relation  
340 with the larger surrounding crystals, which is not expected if they crystallized from a melt. As  
341 can be seen from EDS chemical map illustrated in Figure 4b the dissociation products are  
342 either Al- or Fe-rich. The electron diffraction patterns of the Al-dominant phase are consistent  
343 with a corundum structure (Fig. 4c), while those of the Fe-dominant phase correspond to a  
344 spinel structure (i.e., magnetite) (Fig. 4d). It is noteworthy that this magnetite within the UQ-  
345 domain differs from the dmt described above. For instance, the magnetite within the UQ-  
346 phase does not have the characteristic relict features expected for transformation from a  
347 previous hp-polymorph. This phase is also richer in Al-content than the dmt phase.

348

## 349 **Crystal-chemical behaviour of high-*P* phases**

### 350 ***Influence of Al on the stability of hp-phases***

351 A few experiments were performed with magnetite solid solutions having higher Al contents.  
352 Experiments with starting compositions of  $\text{Fe}(\text{Fe}_{0.6}\text{Al}_{0.4})_2\text{O}_4$  or  $\text{Fe}(\text{Fe}_{0.4}\text{Al}_{0.6})_2\text{O}_4$  produced  
353 post-spinel assemblages similar to those obtained with lower bulk Al contents, such as  
354  $\text{Fe}_2(\text{Fe,Al})_2\text{O}_5 + (\text{Fe,Al})_2\text{O}_3$  or  $\text{Fe}_3(\text{Fe,Al})_4\text{O}_9 + (\text{Fe,Al})_2\text{O}_3$  (Table 1). The appearance of these  
355 assemblages occurred at about the same  $P$ - $T$  range as for the  $\text{Fe}(\text{Fe}_{0.8}\text{Al}_{0.2})_2\text{O}_4$  bulk  
356 composition. This is due to the limited incorporation of Al in the  $\text{O}_5$  and  $\text{O}_9$  phases so that  
357 their compositions do not differ significantly from those synthesized with a  $\text{Fe}(\text{Fe}_{0.8}\text{Al}_{0.2})_2\text{O}_4$   
358 bulk composition. Thus, phase relations of  $\text{Fe}(\text{Fe}_{0.8}\text{Al}_{0.2})_2\text{O}_4$  (shown in Fig. 2) are also valid at  
359 higher bulk Al contents, the only difference being a higher proportion of  $\text{cor}_{\text{ss}}$  and  $\text{hem}_{\text{ss}}$  in the  
360 assemblage.

361

#### 362 *$\text{Fe}_4\text{O}_5$ - $\text{Fe}_2\text{Al}_2\text{O}_5$ solid solutions*

363 The molar volume of the  $\text{Fe}_4\text{O}_5$  phase decreases strongly and linearly with increasing Al  
364 content (Fig. 5a). This behaviour can be explained by the substitution of the smaller  $\text{Al}^{3+}$   
365 cation for  $\text{Fe}^{3+}$  (ionic radius = 53.5 vs. 64.5 pm, respectively, Shannon 1976). The decrease in  
366 molar volume is mostly due to changes in the  $b$ - and  $c$ -parameters (Fig. 5c-d). The  $a$ -lattice  
367 parameter does not significantly change with composition (Fig. 5b). This implies that little to  
368 no Al substitution occurs on the prismatic M3 site and is consistent with the observation of  
369 Boffa Ballaran et al. (2015) that this site is dominated by divalent cations. As already  
370 indicated,  $\text{Al}^{3+}$  substitution for  $\text{Fe}^{3+}$  in the  $\text{O}_5$ -phase is quite limited. The reason for this may  
371 be that the observed large changes in unit-cell parameters with Al incorporation act to  
372 destabilise the crystal structure. This contrasts with the successful synthesis of a  $\text{Fe}_2\text{Cr}_2\text{O}_5$   
373 endmember (Ishii et al. 2014), probably because  $\text{Cr}^{3+}$  has an ionic radius much more similar  
374 to that of  $\text{Fe}^{3+}$  (61.5 pm, Shannon 1976).

375

#### 376 *Al-substitution in $\text{Fe}_7\text{O}_9$*

377 The crystal structure of  $\text{Fe}_7\text{O}_9$  is monoclinic and comprises four different crystallographic  
378 sites for cations (Fe1, Fe2, Fe3, Fe4); Fe1, Fe2 and Fe3 are edge-sharing octahedra and the  
379 Fe4 site is trigonal prismatic (Sinmyo et al. 2016). Sinmyo et al. (2016) indicate that the Fe4  
380 site is almost fully occupied by  $\text{Fe}^{2+}$ . Therefore, we consider that in our  $\text{Fe}_3(\text{Al,Fe})_4\text{O}_9$  phase,  
381 Al will substitute for  $\text{Fe}^{3+}$  at the Fe1, Fe2 and Fe3 octahedral sites whereas the Fe4 site will be  
382 occupied by  $\text{Fe}^{2+}$ . This distribution is consistent with our limited lattice parameter data for Al-  
383 bearing  $\text{Fe}_7\text{O}_9$ . The  $b$ -parameter, which is mostly controlled by the height of the trigonal  
384 prism, does not vary significantly with Al content. On the other hand, we find that even small  
385 amounts of Al (e.g. 0.22-0.33 cpfu; see Table 2) substitution lead to a significant shortening  
386 in the  $a$ - and  $c$ -parameters. The  $a$ -parameter is the direction of the edge-sharing octahedral  
387 strips containing the Fe1, Fe2 and Fe3 sites in the structure and the stacking direction of these  
388 strips is along the  $c$ -axis (Sinmyo et al. 2016). Based upon the coexistence with a corundum  
389 solid solution in our experiments, it appears that the maximum Al content that the  $\text{O}_9$ -phase  
390 can accommodate is even more limited than observed for the  $\text{O}_5$ -phase.

391

### 392 *$\text{Fe}_2\text{O}_3$ - $\text{Al}_2\text{O}_3$ solid solutions*

393 In most experiments,  $\text{Fe}^{3+}$ -bearing corundum coexists with other high- $P$  Fe-Al-oxides (Table  
394 1). In five experiments, both Fe-bearing corundum and Al-bearing hematite are present (Table  
395 1). The chemical compositions and lattice parameters for these two phases are given in Table  
396 3. It is notable that the chemical compositions of coexisting  $\text{hem}_{\text{ss}}$  and  $\text{cor}_{\text{ss}}$  from our  
397 experiments are faithfully consistent with the binary miscibility gap observed at much lower  
398 pressures of  $\leq 4$  GPa (Fig. 6, Feenstra et al. 2005; Turnock & Eugster 1962; Atlas & Sumida  
399 1958). In addition, our unit-cell–composition systematics agree very well with those reported  
400 by Feenstra et al. (2005) (Fig. 7). This implies that the mixing behaviour along the hematite–  
401 corundum binary described by Feenstra et al. (2005) is still valid for pressures up to 20 GPa.



402 In some experiments, the cor<sub>ss</sub> does not coexist with a hem<sub>ss</sub> and is therefore not necessarily  
403 Fe<sup>3+</sup>-saturated.

404

405

## IMPLICATIONS

406 Recent experimental syntheses of new high-pressure post-spinel phases with stoichiometries  
407 of M<sub>4</sub>O<sub>5</sub>, M<sub>5</sub>O<sub>6</sub>, M<sub>7</sub>O<sub>9</sub>, and M<sub>9</sub>O<sub>11</sub> give rise to the question if these phases might occur in  
408 natural samples. Since they are all stable over certain ranges in *P* and *T*, they could serve as  
409 “index phases” indicative of the conditions the host sample experienced (e.g. deep mantle or  
410 shock-metamorphosed samples). For instance, diamonds and their inclusions provide a unique  
411 opportunity to directly study pristine material from great depths in the Earth’s mantle. Several  
412 studies report exotic non-ultramafic (non-silicate) inclusions from uncertain or controversial  
413 depths (e.g. Hayman et al. 2005). In some cases, magnetite has been identified either as  
414 discrete grains or associated with other phases such as wüstite, metallic alloy, hematite or  
415 pyrrhotite (e.g. Anzolini et al. 2019, 2020, Shirey et al. 2019, Jacob et al. 2011, 2016, Wirth et  
416 al. 2014, Hayman et al. 2005, Stachel et al. 1998, Hutchison 1997). Since end-member  
417 magnetite is stable only up to ~10 GPa, the logical interpretation is that such inclusions  
418 formed in the upper mantle. However, the occurrence of particular microstructures in  
419 magnetite can be an indicator that it formed as a breakdown product from a precursor phase  
420 (e.g. Jacob et al. 2016, Anzolini et al. 2020). For example, Jacob et al. (2016) reported  
421 framesitic diamond with inclusions of pyrrhotite that had a partially developed  
422 nanocrystalline reaction rim of magnetite. Based on the epitaxial growth of pyrrhotite,  
423 magnetite and diamond, Jacob et al. (2016) concluded that magnetite formed at the expense of  
424 pyrrhotite directly, followed by the precipitation of diamond on this magnetite corona. In  
425 performing Transmission Kikuchi Diffraction (TKD) analyses, they identified twinning on the  
426 {311} planes (note, the {311} plane is equivalent to the {113} plane (Fig.3c) in the cubic

427 system), a textural feature shown to occur as a consequence of a breakdown of  $\text{Fe}_4\text{O}_5 + \text{Fe}_2\text{O}_3$   
428 or  $\text{Fe}_4\text{O}_5 + \text{O}_2$  (Woodland et al. 2012, Myhill et al. 2016). Thus, Uenver-Thiele et al. (2017b)  
429 interpreted this twinning in the natural magnetite to be the first evidence for the natural  
430 occurrence of an  $\text{O}_5$ -phase in a mantle sample. Since magnetite both with and without  $\{311\}$   
431 twinning was present in the diamond, Uenver-Thiele et al. (2017b) considered the origin of  
432 these inclusions to lie in the upper mantle between 8-10 GPa, conditions at which  $\text{Fe}_4\text{O}_5$  and  
433 magnetite can coexist (Myhill et al. 2016; Schollenbruch et al. 2011; Woodland et al. 2012).  
434 Nevertheless, twinning on the  $\{311\}$  planes in magnetite can also result from the retrograde  
435 reaction of other iron oxide phases such as  $\text{Fe}_5\text{O}_6$  (Woodland et al. 2023). On the other hand,  
436 the  $\{100\}$  and  $\{113\}$  lamellae in our recovered defect magnetite, which we interpret to have  
437 formed from a  $\text{hp-O}_4$  phase at  $P$  and  $T$  (see above), can also help to distinguish one precursor  
438 candidate from another. This means that detailed documentation of twinning features and  
439 lamellae textures are important in future studies of experimental and natural samples.

440 The results of this present work also demonstrate that the composition of phases in the  $\text{Fe}^{2+}$ -  
441  $\text{Fe}^{3+}$ -Al-oxide system can help to discriminate whether a spinel-structured phase formed from  
442 a precursor phase or not. In contrast to spinel-structured phases, the high- $P$  post-spinel phases  
443 in this study are found to incorporate only minor amounts of Al ( $\sim 0.1$  cpfu Al in a  $\text{hp-O}_4$   
444 polymorph,  $\sim 0.3$  cpfu Al in  $\text{A}_2\text{B}_2\text{O}_5$  or  $\sim 0.4$  cpfu in  $\text{A}_4\text{B}_3\text{O}_9$ ). This implies that any spinel that  
445 formed from a higher- $P$  precursor phase must also have a low Al content. In fact, spinels  
446 occurring as inclusions in diamond generally have low Al contents (Hayman et al. 2005;  
447 Wirth et al. 2014; Palot et al. 2016; Anzolini et al. 2020), but this may also be due to  
448 formation in a low-Al environment. For such compositions, our current results indicate that  
449 the phase relations of  $\text{Mg}_{0.5}\text{Fe}_{0.5}\text{Fe}_2\text{O}_4$  presented by Uenver-Thiele et al. (2017b) are still  
450 applicable. On the other hand, Wirth et al. (2014) reported Al-rich spinels with compositions  
451  $\sim (\text{Mg}_{0.73}\text{Fe}_{0.25}\text{Mn}_{0.02})(\text{Fe}_{0.44}\text{Al}_{1.17}\text{Cr}_{0.39})\text{O}_4$  occurring together with magnetite-magnesioferrite

452 solid solutions within a magnesiowüstite (MW) inclusion in a diamond from Brazil. Since the  
453 Al-rich spinels were found to be in epitaxial relationships with the host MW and the nearly  
454 Al-free magnesioferrite/magnetite, Wirth et al. (2014) considered that their exsolution from  
455 the MW host must have taken place under the same *P-T* conditions predicted for the  
456 magnesioferrite/magnetite. Our current study indicates that this Al-rich spinel could not have  
457 initially formed as a post-spinel phase. However, the high Mg content reported by Wirth et al.  
458 (2014) may also suggest that in this case the spinel structure was stabilized to higher  
459 pressures, up to ~15-16 GPa at ~1200-1600 °C (Akaogi et al. 1999). The effect of Mg on  
460 phase relations is the subject of an ongoing study (Uenver-Thiele et al. in prep). In any case,  
461 in Al-rich bulk compositions at *P-T* conditions above spinel stability, an additional Al-rich  
462 phase like corundum or garnet would be expected to coexist with low-Al oxide phases, like  
463 those described here.

464

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756

## FIGURE CAPTION

757 Figure 1. Backscatter electron images of the experimental run products showing a) the  
758 assemblage of well-crystallized  $\text{cor}_{\text{ss}} + \text{hem}_{\text{ss}} + \text{Fe}_2(\text{Fe,Al})_2\text{O}_5$ , b) coexistence of large crystals  
759 of  $\text{Fe}_3(\text{Fe,Al})_4\text{O}_9$  with interstitial  $\text{cor}_{\text{ss}} + \text{hem}_{\text{ss}}$ , c) crystals of  $\text{cor}_{\text{ss}}$  and a striated high-pressure  
760 post-spinel phase, presumably  $\text{hp-O}_4$ , that have transformed during quenching into defect  
761 magnetite (dmt; see text for details), d) run products of Z1953o with an UQ-phase coexisting  
762 with the same dmt as in (c). The dmt has slightly variable contrast due to effects of  
763 composition as well as orientation, and is full of striations, comparable to those observed in  
764 (c). Note that the UQ-phase differs from dmt by its texture (see text below for details).  
765 Abbreviations: hematite-corundum solid solutions are referred to as  $\text{cor}_{\text{ss}}$  or  $\text{hem}_{\text{ss}}$ ; wü=  
766 wüstite; UQ-phase= unquenchable phase;  $[\text{hp-O}_4]$  = spinel phase with a high magnetite  
767 component formed presumably by retrograde reaction from a high- $P$  polymorph during  
768 quench.

769

770 Figure 2. Phase stabilities for the bulk composition  $\text{Fe}^{2+}(\text{Fe}^{3+}_{0.8}\text{Al}_{0.2})_2\text{O}_4$ . The boundaries of  
771 the assemblage stability fields are delineated by solid lines. The dashed black line indicates  
772 the breakdown reaction for magnetite which was observed in situ by Schollenbruch et al.  
773 (2011) and later identified by Woodland et al. (2012). Also shown is the position of the  
774 breakdown reaction for hercynite (dotted line) from Schollenbruch et al. (2010). The errors in  
775  $P$  and  $T$  are approximately equal to the size of the symbol. Note: Experiment Z1953o at 20  
776 GPa and 1300 °C contains an additional UQ-phase (see also Fig. 1d).

777

778 Figure 3. (a) Bright-field (BF) TEM image of sample Z1953o illustrating the assemblage of  
779 dmt + UQ-phase. (b) BF-TEM image of the dmt phase [right grain in the BF-TEM in (a)] with  
780 numerous lamellae normal to the projection of the  $\{100\}$  plane. The inset shows the

781 corresponding precession selected area electron diffraction (SAED) pattern (circle), indicating  
782 a superimposition of the  $\langle 011 \rangle$  and the  $\langle 121 \rangle$  zone axes of magnetite. The two zone axes  
783 have a common direction along  $\langle 011 \rangle$  (the diagonal direction from the lower left to upper  
784 right in the electron diffraction pattern). Note that the indexing is with the  $\langle 011 \rangle$  zone axis of  
785 magnetite. (c) Dark-field TEM image (with  $g$ -vectors of 400 from the matrix and twins) of the  
786 lamella-bearing dmt phase [the center one in image (a)] displaying  $\{113\}$  twin lamellae. The  
787 inset shows a SAED pattern indicating the twin relation.

788

789 Figure 4. (a) High-angle annular dark-field (HAADF) STEM image illustrating the dmt phase  
790 (left-hand side) next to the fine-grained unquenchable domains (right-hand side) and (b) EDS  
791 chemical map of the UQ-domain indicating Al-dominant and Fe-dominant dissociation  
792 products. (c) BF-TEM images of the Al-dominant phase indexed with a corundum structure  
793 (Crn) shown in the lower right inset. (d) BF-TEM images of the Fe-dominant phase. A  
794 selected area electron diffraction (SAED) pattern was taken in the area indicated by the white  
795 circle that is consistent with magnetite. Notably, this magnetite does not have the features of  
796 a defect magnetite structure.

797

798 Figure 5. Lattice parameters of  $\text{Fe}_2\text{Fe}_2\text{O}_5\text{-Fe}_2\text{Al}_2\text{O}_5$  solid solutions as a function of Al content  
799 (cpfu). These solid solutions adopt the same  $\text{CaFe}_3\text{O}_5$ -type structure (Cmcm space group) as  
800  $\text{Mg}_2\text{Fe}_2\text{O}_5$  reported by Boffa Ballaran et al. (2015). Variation in a) molar volume, b)  $a$ -  
801 parameter, c)  $b$ -parameter and d)  $c$ -parameter with the amount of Al in the crystal structure.  
802 For comparison, lattice parameters of the endmember  $\text{Fe}_4\text{O}_5$  from the study by Wooland et al.  
803 (2022) are also shown. Error bars are within the size of the symbols.

804



805 Figure 6. Plot of the  $\text{Fe}^{3+}/(\text{Fe}^{3+}+\text{Al})$  of Al-bearing hematite and Fe-bearing corundum as a  
806 function of synthesis temperature. Experimental results obtained in the studies by Feenstra et  
807 al. (2005) at 1-40 kbar, Turnock & Eugster (1962) at 2-4 kbar and Atlas & Sumida (1958) at 1  
808 bar are also shown for comparison.

809

810 Figure 7. Lattice parameter a (a), lattice parameter c (b) as well as volume (c) plotted versus  
811 chemical composition of corundum-hematite solid solutions. Insets show enlargements.  
812 Dashed lines show linear mixing between synthetic  $\text{Al}_2\text{O}_3$  and  $\text{Fe}_2\text{O}_3$  (Finger and Hazen,  
813 1980; Newnham and de Hann, 1962).

814

**Table 1. Experimental conditions and run products**

Experiment	Pressure [GPa]	Temperature [C°]	Duration [h]	Phase assemblage
<b>Fe(Al<sub>0.2</sub>Fe<sub>0.8</sub>)<sub>2</sub>O<sub>4</sub></b>				
M798*	6	1200	3	O <sub>4</sub>
M808	8	1300	2.5	O <sub>4</sub>
M830	9	1400	1	O <sub>4</sub>
M840	9	1100	3	O <sub>4</sub>
M804*	11	1300	2	O <sub>5</sub> + cor <sub>ss</sub> + FeO
M789	12	1300	1.5	O <sub>5</sub> + hem <sub>ss</sub> + cor <sub>ss</sub>
M788	14	1200	2.5	O <sub>5</sub> + hem <sub>ss</sub> + cor <sub>ss</sub>
M790	14	1400	1.5	O <sub>5</sub> + hem <sub>ss</sub> + cor <sub>ss</sub>
Z2039u*	15	1200	3	O <sub>5</sub> + cor <sub>ss</sub> + (sid)
Z2031o*	16	1200	3	O <sub>5</sub> + cor <sub>ss</sub> + (hem <sub>ss</sub> ) + (sid)
Z2030o*	16	1500	1	O <sub>5</sub> + cor <sub>ss</sub> + (sid)
M823*	18	1200	2.5	[hp-O <sub>4</sub> ] + cor <sub>ss</sub>
M825*	18	1300	2	O <sub>5</sub> + cor <sub>ss</sub>
M841	18	1300	2	O <sub>9</sub> + [hp-O <sub>4</sub> ] + cor <sub>ss</sub>
Z2040u*	18	1400	1	O <sub>9</sub> + hem <sub>ss</sub> + cor <sub>ss</sub>
Z1953o	20	1300	1	[hp-O <sub>4</sub> ] + cor <sub>ss</sub> + UQ
S7317	20	1550	1.5	O <sub>5</sub> + cor <sub>ss</sub>
H5668	20	1400	1.5	O <sub>9</sub> + cor <sub>ss</sub>
S7316	22	1300	1.5	[hp-O <sub>4</sub> ] + cor <sub>ss</sub>
H5745	22	1450	1 min	O <sub>9</sub> + cor <sub>ss</sub>
<b>Fe(Al<sub>0.4</sub>Fe<sub>0.6</sub>)<sub>2</sub>O<sub>4</sub></b>				
M815*	14	1400	1.5	O <sub>5</sub> + cor <sub>ss</sub> + (O <sub>6</sub> )
M822	18	~1250	2	O <sub>9</sub> + cor <sub>ss</sub> + (sid)
<b>Fe(Al<sub>0.6</sub>Fe<sub>0.4</sub>)<sub>2</sub>O<sub>4</sub></b>				
M842	10	1200		O <sub>4</sub> + (O <sub>5</sub> ) + (hem <sub>ss</sub> ) + (cor <sub>ss</sub> )
M826	13	1200	3	O <sub>5</sub> + cor <sub>ss</sub>
M843	14	1400	1.5	O <sub>5</sub> + cor <sub>ss</sub>
<b>Fe<sub>3</sub>Fe<sub>3</sub>AlO<sub>9</sub></b>				
M844	18	1400	1.5	O <sub>5</sub> + cor <sub>ss</sub> + (sid)
M784	18	1500	1	O <sub>5</sub> + cor <sub>ss</sub> + (sid)

Notes: \* Experiment performed with a Pt-capsule; O<sub>4</sub>: cubic Fe(Al,Fe)<sub>2</sub>O<sub>4</sub>; O<sub>5</sub>: orthorhombic Fe<sub>2</sub>(Al,Fe)<sub>2</sub>O<sub>5</sub> phase; O<sub>9</sub>: monoclinic Fe<sub>3</sub>(Al,Fe)<sub>4</sub>O<sub>9</sub> phase; wü: wüstite; UQ=unquenchable phase that dissociated into Fe<sup>3+</sup>-bearing corundum + nearly pure magnetite; sid= siderite; hematite-corundum solid solutions are referred to as cor<sub>ss</sub> or hem<sub>ss</sub>; phases in parentheses = only present in traces; phase in brackets = assumed high-pressure polymorph that reacted back to defect-magnetite during quenching

**Table 2. Unit cell parameters of Fe(Fe,Al)<sub>2</sub>O<sub>4</sub>, Fe<sub>2</sub>(Fe,Al)<sub>2</sub>O<sub>5</sub> and Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub>**

Experiment	nAl	lattice parameters				
	[c.p.f.u.]	<i>a</i> [Å]	<i>b</i> [Å]	<i>c</i> [Å]	V [Å <sup>3</sup> ]	β [°]
<b><u>Fe(Fe,Al)<sub>2</sub>O<sub>4</sub></u></b>						
M798	0.38(1)	8.3464(1)			581.423(14)	
M830	0.39(3)	8.3387(1)			579.822(10)	
M840	0.41(1)	8.3379(1)			579.66(2)	
M808	0.44(2)	8.3421(1)			580.54(2)	
M842	0.85(1)	8.2901(1)			569.75(3)	
<b><u>defect-Fe(Fe,Al)<sub>2</sub>O<sub>4</sub></u><sup>1</sup></b>						
Z1953o	0.10(1)	8.3783(1)			588.11(2)	
S7316	0.11(1)	-			-	
M823	0.11(1)	8.3830(1)			589.11(2)	
M841	0.14(1)	8.3697(4)			586.31(9)	
<b><u>Fe<sub>2</sub>(Fe,Al)<sub>2</sub>O<sub>5</sub></u></b>						
M788	0.07(1)	2.8928(1)	9.7820(3)	12.5678(4)	355.632(12)	
M789	0.09(1)	2.8945(1)	9.7750(2)	12.5671(2)	355.572(8)	
M790	0.11(1)	2.8938(1)	9.7676(2)	12.5633(2)	355.109(8)	
M825	0.12(1)	2.8946(1)	9.7632(2)	12.5593(2)	354.934(8)	
Z2039u	0.12(1)	2.8933(1)	9.7634(3)	12.5557(3)	354.683(10)	
M843	0.12(1)	2.8937(1)	9.7725(2)	12.5640(2)	355.289(7)	
M826	0.13(1)	2.8939(1)	9.7654(2)	12.5580(2)	354.900(7)	
M844	0.14(1)	2.8942(1)	9.7582(3)	12.5582(3)	354.670(11)	
M784	0.17(1)	2.8941(1)	9.7545(2)	12.5491(2)	354.270(6)	
M804	0.17(1)	2.8933(1)	9.7588(5)	12.5558(5)	354.51(2)	
M815	0.20(1)	2.8929(1)	9.7495(3)	12.5484(4)	353.923(14)	
Z2030o	0.23(1)	2.8934(1)	9.7449(2)	12.5435(2)	353.667(7)	
Z2031o	0.23(1)	2.8937(1)	9.7699(3)	12.5647(3)	355.223(10)	
S7317	0.30(1)	2.8924(1)	9.7316(5)	12.5334(5)	352.780(18)	
<b><u>Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub></u></b>						
M815	-	2.8936(3)	9.8707(12)	15.3316(14)	437.89(5)	
<b><u>Fe<sub>3</sub>(Fe,Al)<sub>4</sub>O<sub>9</sub></u></b>						
Fe <sub>7</sub> O <sub>9</sub> <sup>2</sup>	0.00	9.696(2)	2.8947(6)	11.428(3)	314.10(12)	101.69(2)
M841	0.22(1)	9.6784(3)	2.8991(1)	11.4112(4)	313.545(14)	101.685(3)
Z2040u	0.23(1)	9.6717(4)	2.8961(1)	11.4222(4)	313.248(14)	101.741(3)
H5668	0.28(1)	9.6732(2)	2.8960(1)	11.4127(3)	313.042(9)	101.726(2)
M822	0.33(1)	9.6683(3)	2.8930(7)	11.3925(3)	312.00(1)	101.729(2)
H5745	0.42(2)	9.6604(9)	2.8919(2)	11.3965(9)	311.74(3)	101.725(6)

<sup>1</sup> defect-magnetite phase with relict features of an presumably hp-polymorph and an Al-poor composition

<sup>2</sup> endmember data from Sinmyo et al. (2016)



**Table 3. Unit cell parameters of corundum-hematite solid solutions**

Experiment	corundum				hematite			
	(Fe <sup>3+</sup> /Fe <sup>3+</sup> +Al)	<i>a</i> [Å]	<i>c</i> [Å]	<i>V</i> <sub>mol</sub> [cm <sup>3</sup> /mol]	(Fe <sup>3+</sup> /Fe <sup>3+</sup> +Al)	<i>a</i> [Å]	<i>c</i> [Å]	<i>V</i> <sub>mol</sub> [cm <sup>3</sup> /mol]
M842	0.040	-	-	-	0.985	-	-	-
M804*	0.045	4.7694(3)	13.014(1)	256.37(3)				
M815	0.045	4.7711(3)	13.018(2)	256.64(4)				
M826	0.045	4.7672(1)	13.0084(5)	256.030(9)				
Z2030o	0.045	4.7733(7)	13.012(3)	256.75(7)				
M825	0.055	4.7698(1)	13.0141(5)	256.42(1)				
Z2031o	0.065	4.7676(5)	13.018(2)	256.25(4)	0.970	-	-	-
M784	-	4.7749(2)	13.0310(8)	257.30(2)				
M823	0.065	4.7768(2)	13.0333(8)	257.55(2)				
M841	0.070	4.7800(3)	13.0428(11)	258.08(2)				
S7316	0.075	-	-	-				
M788	-	4.7820(3)	13.0407(2)	258.25(3)	0.850	4.9985(1)	13.6336(5)	295.000(11)
Z2039u	0.080	4.7698(2)	13.0148(10)	256.43(2)				
Z1953o	-	4.7845(4)	13.0290(10)	258.29(3)				
M843	0.090	4.7854(1)	13.0501(4)	258.810(8)				
M844	0.090	4.7797(3)	13.0439(12)	258.08(3)				
M822	0.095	4.7896(1)	13.0656(5)	259.570(9)				
M789	0.105	4.7900(2)	13.0564(10)	259.43(2)	0.815	4.9892(1)	13.6056(7)	293.30(2)
H5668	0.100	4.7897(2)	13.0612(10)	259.50(2)				
S7317	0.110	4.7858(8)	13.057(4)	258.99(7)				
H5745	0.115	4.7901(4)	13.0725(2)	259.76(3)				
Z2040u	0.120	4.7976(4)	13.0845(12)	260.82(3)	0.820	4.9714(3)	13.604(3)	291.18(6)
M790	0.145	4.7996(2)	13.0825(8)	260.99(2)	0.765	4.9741(2)	13.5698(9)	290.76(2)

\* wüstite has been detected with lattice parameters of *a* = 4.2718(2) Å and *V*<sub>mol</sub> = 77.953(12) cm<sup>3</sup>/mol

Figure 1

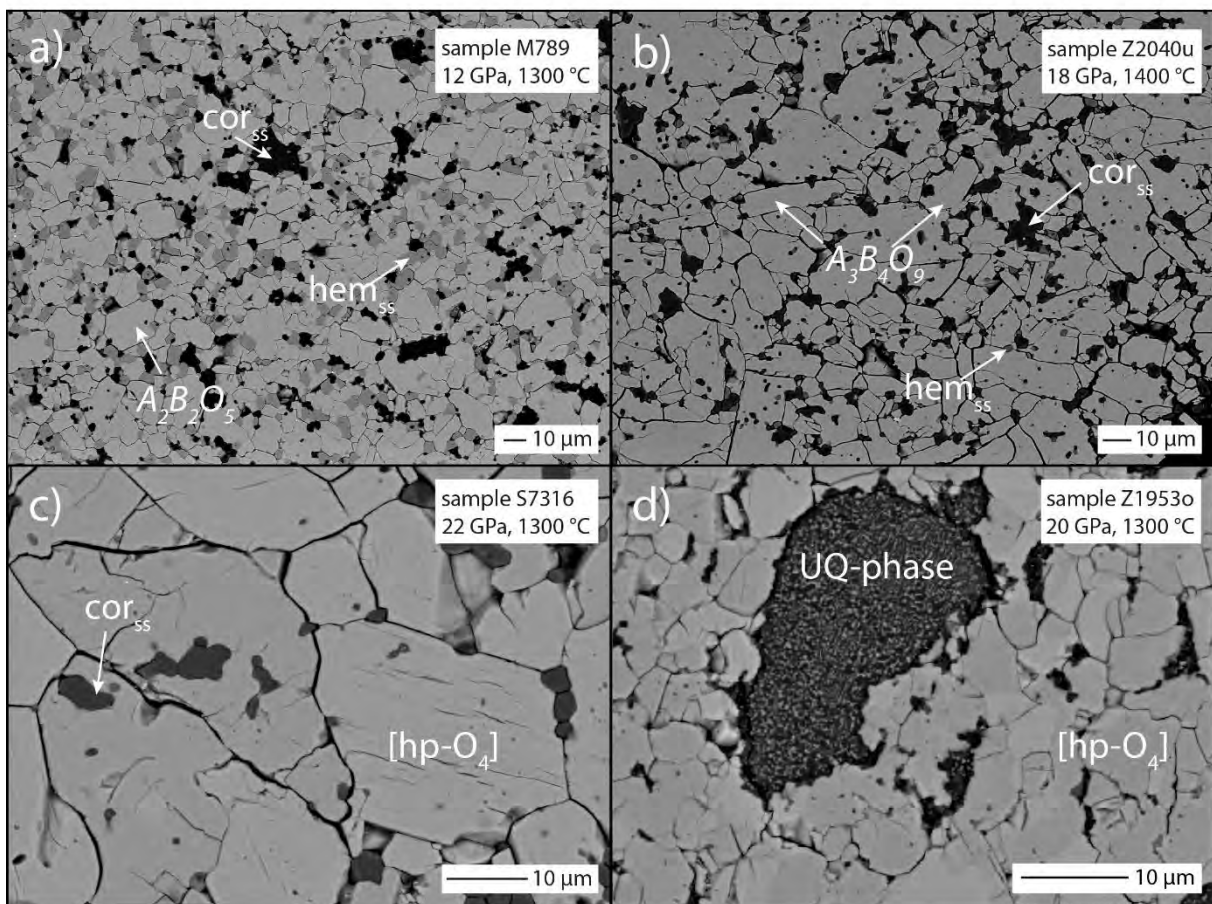


Figure 2

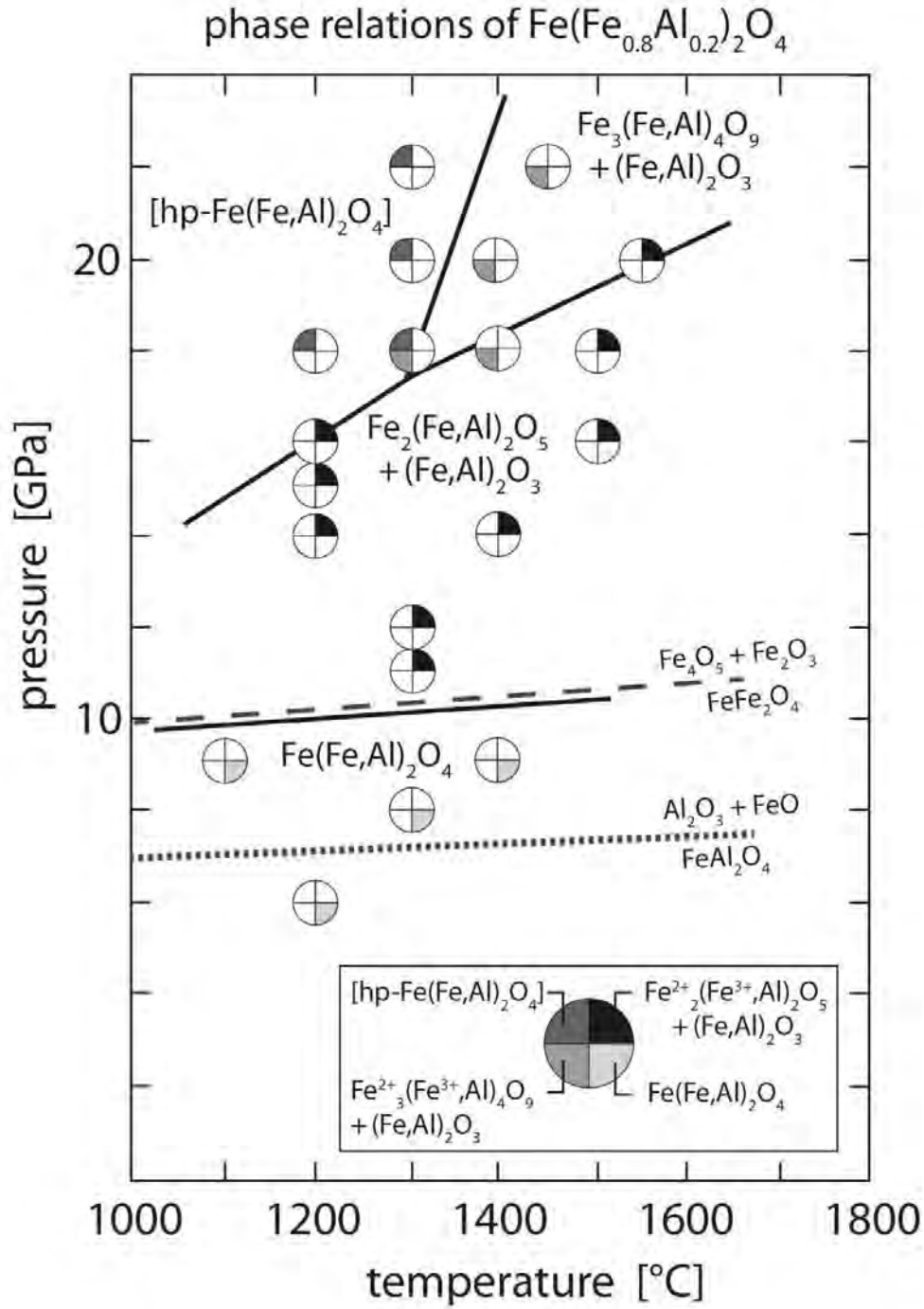


Figure 3

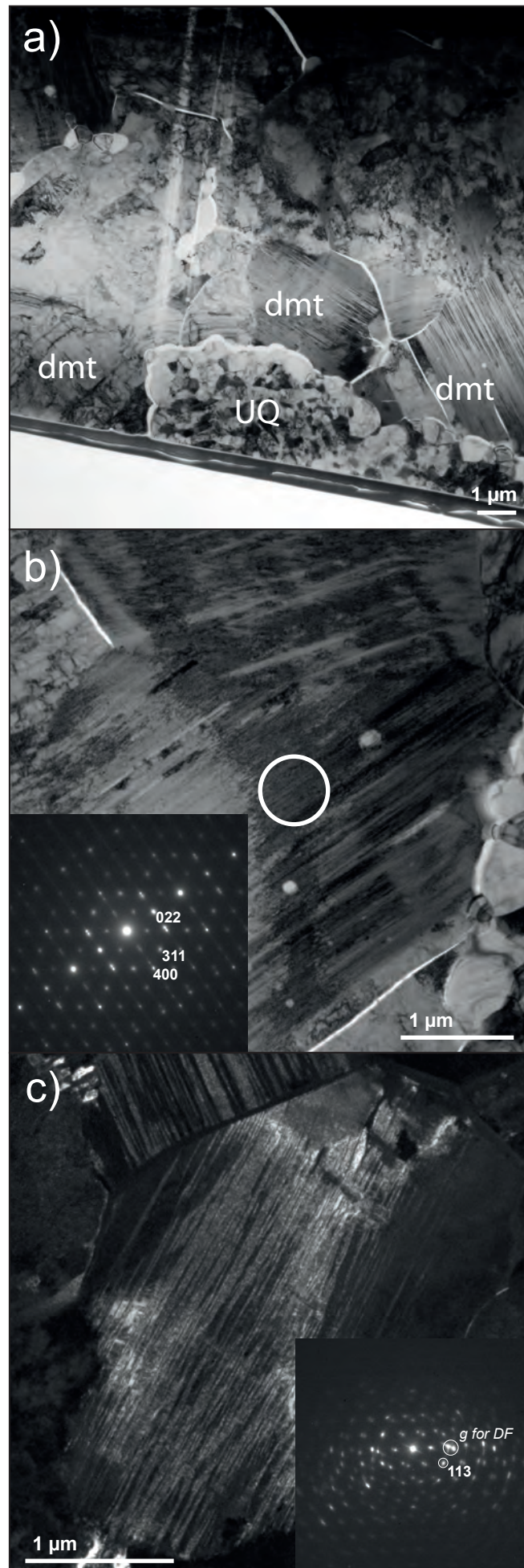




Figure 4

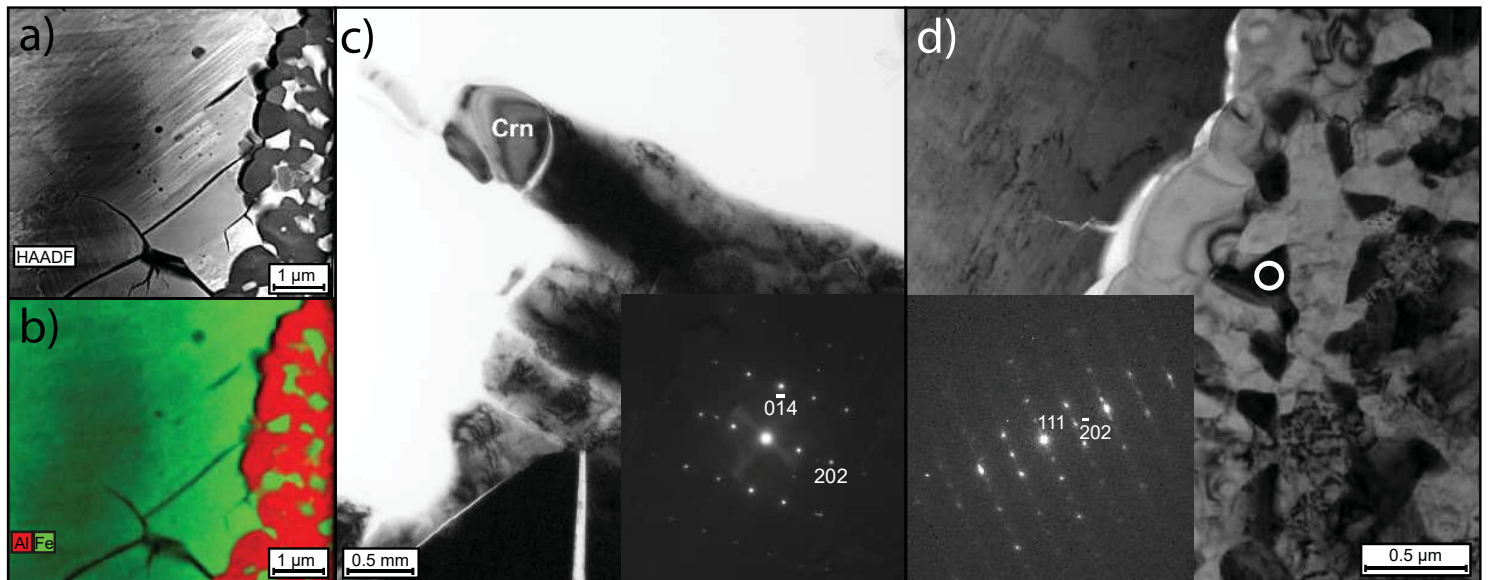


Figure 5

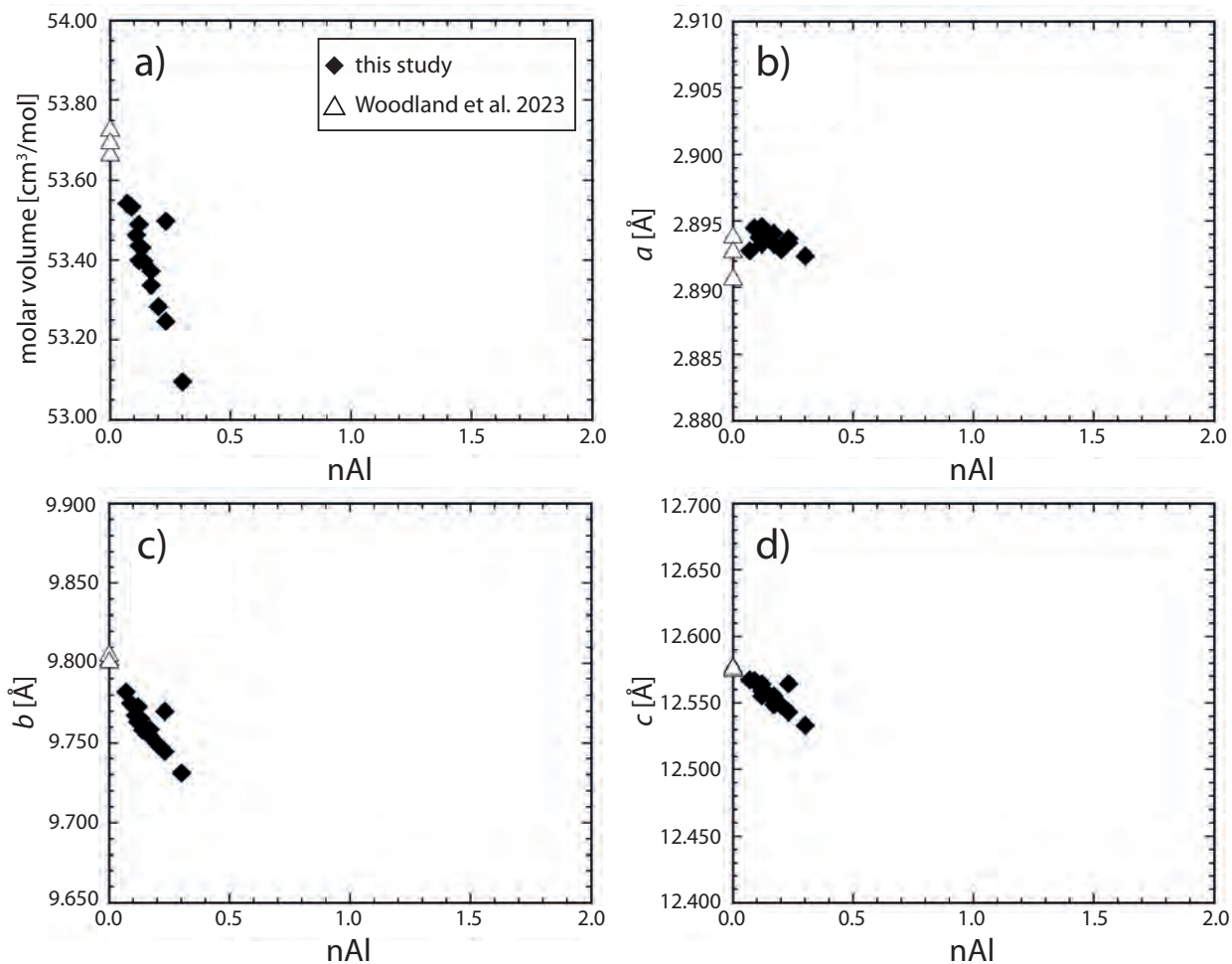


Figure 6

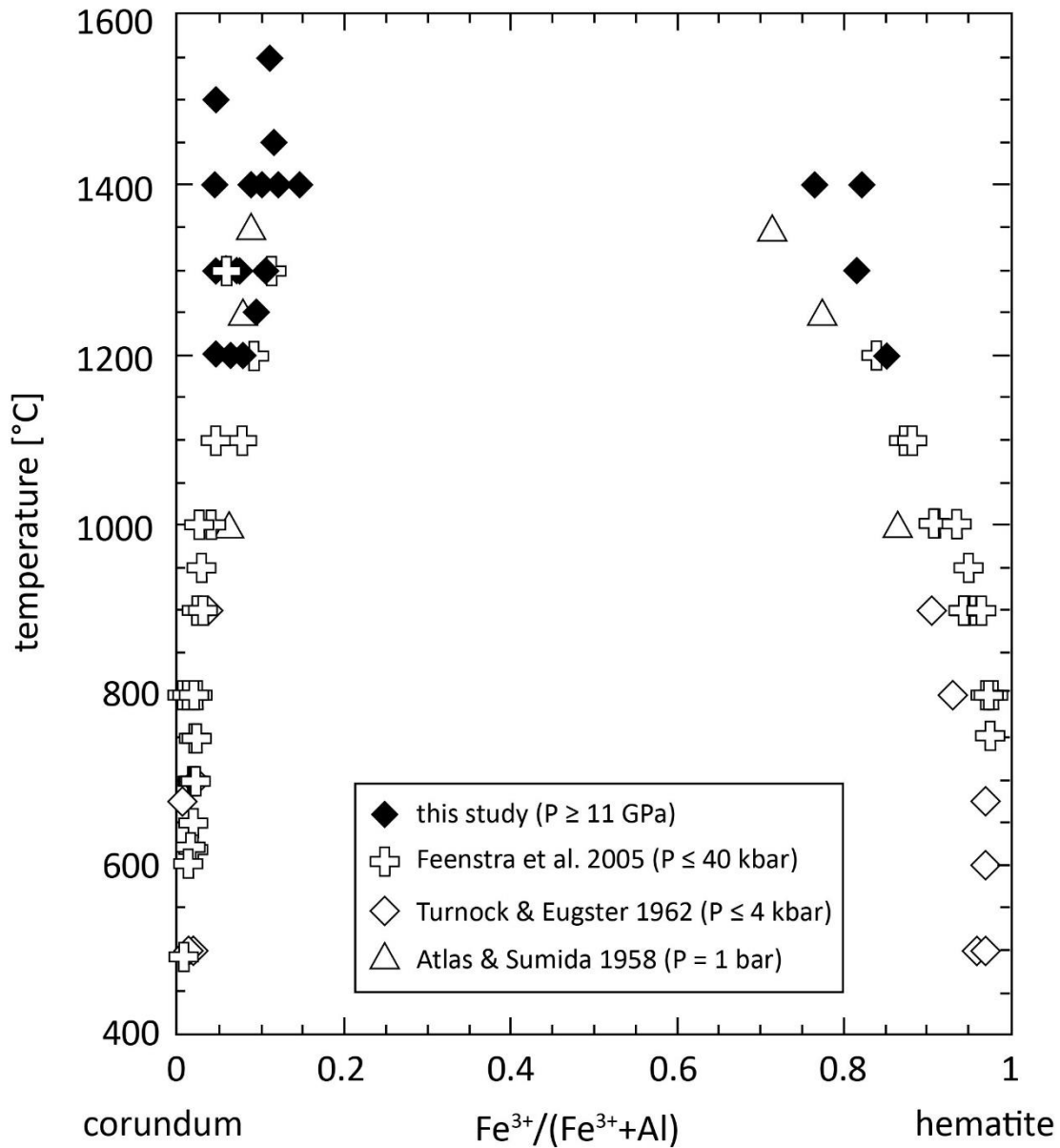


Figure 7

