Revision 2 1 Word count: 9548 2 Raman scattering of omphacite at high pressure: towards its possible application to 3 elastic geothermobarometry 4 Lisa Baratelli<sup>1</sup>, Mara Murri<sup>2</sup>, Matteo Alvaro<sup>2</sup>, Mauro Prencipe<sup>3</sup>, Boriana Mihailova<sup>4</sup>, Fernando 5 Cámara<sup>1</sup> 6 <sup>1</sup>Department of Earth Sciences, University of Milan, Via Sandro Botticelli 23, 20133 Milan, Italy. 7 <sup>2</sup>Department of Earth and Environmental Sciences, University of Pavia, Via Adolfo Ferrata 1, 27100 8 9 Pavia, Italy. <sup>3</sup>Department of Earth Sciences, University of Turin, Via Valperga Caluso 35, 10125, Turin, Italy. 10 <sup>4</sup>Department of Earth Sciences, Universität Hamburg, Grindelallee 48, 20146 Hamburg, Germany. 11 12 Corresponding author: Lisa Baratelli. Contact: lisa.baratelli@unimi.it 13 14 15 Abstract Due to their widespread occurrence in several geological settings, omphacite inclusions 16 could be used for elastic Raman geothermobarometry. However, the Raman scattering of complex 17 silicate minerals entrapped in a host depends on both the chemical composition and elastic strain 18

developed during the metamorphic pathway, which makes the task very challenging. Here, as a very first step to probe the potential of omphacite to be used as a mineral inclusion in elastic geothermobarometry, we report the pressure dependence of the Raman spectra of omphacite crystals with the same composition, approximately  $Jd_{45}Di_{55}$ , but having different symmetry because of the existence (*P2/n*) or absence (*C2/c*) of chemical order at the six- and eightcoordinated cation sites. The experimental results are complemented by ab initio quantum

mechanical simulations on fully ordered omphacite (Jd<sub>50</sub>Di<sub>50</sub>). We demonstrate that the position of 25 the well-resolved Raman peak near 688 cm<sup>-1</sup>, arising from Si-O-Si bond bending, is very sensitive to 26 pressure but independent of the state of chemical order, which makes it promising to be utilized in 27 Raman geobarometry. The width of this peak varies with chemical order but not with pressure and 28 therefore can be used to constrain the temperature of inclusion entrapment, because the 29 chemical order is indicative of the closure temperature of the cation-exchange reaction. However, 30 further detailed analyses on the compositional variation of the Raman spectra of omphacite is 31 required before considering omphacite-in-garnet systems as suitable for Raman elastic 32 geothermobarometry. 33 34 Keywords: Omphacite, site-occupancy order, Raman spectroscopy, high pressure, ab initio 35 calculation 36 37 Introduction 38 Clinopyroxenes (general formula M2M1T<sub>2</sub>O<sub>6</sub>) are key minerals, due to their widespread 39

occurrence in the Earth's crust and upper mantle (Skelton and Walker 2015), to characterize 40 various geological settings and processes. They can be found as both rock-forming and accessory 41 minerals in several terrestrial rock types from diverse geological contexts, including mantle 42 xenoliths, magmatic rocks (e.g., Lentz et al. 2011; Murri et al. 2016, 2019b; Gianola et al. 2023) 43 and high-pressure/temperature metamorphic rocks (Philpotts and Ague 2009; Gilio et al. 2020). 44 Clinopyroxenes also occur as inclusions in diamonds (Meyer and Boyd 1972; Nestola et al. 2016; 45 Pasqualetto et al. 2022) and as rock-forming minerals in meteorites (e.g., Papike 1980; Alvaro et 46 al. 2015; Carli et al. 2023). Among clinopyroxenes, omphacite, a solid solution of jadeite 47 (NaAlSi<sub>2</sub>O<sub>6</sub>) (20-80%), augite [Ca(Mg,Fe<sup>2+</sup>)Si<sub>2</sub>O<sub>6</sub>] (20-80%), and aegirine (NaFe<sup>3+</sup>Si<sub>2</sub>O<sub>6</sub>) (0-50%) 48

(Morimoto 1988) with the general formula (Ca,Na)(Mg,Fe<sup>2+</sup>,Al)Si<sub>2</sub>O<sub>6</sub>, is extremely important for 49 constraining pressure and temperature conditions in eclogitic metabasites where quartz is not 50 present as an equilibrium phase at peak conditions. It is well known that omphacites may exhibit 51 site-occupancy cation order, which leads to a splitting of both M1 and M2 sites into (M1, M11) and 52 (M2, M21), respectively (Figure 1). Consequently, the chemical order reduces the space group 53 symmetry from C2/c to P2/n, (e.g., Carpenter et al. 1990a, b; Boffa Ballaran et al. 1998a, b). Thus, 54 the determination of the state of chemically ordered omphacite can constrain the closure 55 temperature and cooling rate experienced by the host rock. The degree of cation order of 56 omphacite is commonly determined by means of single-crystal X-ray diffraction (XRD) 57 58 measurements followed by structural refinements, but this analysis generally requires large (~50-150 µm) good-quality single crystals. Moreover, chemically ordered regions form antiphase 59 domains that are detectable by transmission electron microscopy, therefore their size variation 60 has been proposed to serve as a geothermometer (Phakey and Ghose 1973; Carpenter 1980, 61 1981a, b). However, the dimensions of chemically ordered regions depend also on the chemical 62 composition as well (Cámara et al. 1998) and they can be further affected by post-crystallization 63 deformation processes (Van Roermund and Lardeaux 1991). Raman spectroscopy can verify the 64 existence of M-site cation order in omphacite by the splitting of the SiO<sub>4</sub>-stretching Raman peak 65 near 1017 cm<sup>-1</sup>, due to the doubling of the primitive unit cell (Katerinopoulou et al. 2008). Besides, 66 Raman spectroscopy has the advantage to probe micrometer-size spatial regions without the 67 necessity to extract the crystal from the host. Consequently, this analytical method is among the 68 most popular experimental techniques used in elastic geothermobarometry (Korsakov et al. 2009; 69 Anzolini et al. 2018; Nestola et al. 2018; Zhong et al. 2019; Campomenosi et al. 2021). By using this 70 technique, it is possible to estimate the equilibration pressure and temperature of a mineral. The 71 pressure and temperature conditions of the inclusion entrapment ( $P-T_{trap}$ ) can be estimated 72

knowing the inclusion pressure  $(P_{inc})$  at the conditions at which it is exhumed to the surface (P-73  $T_{end}$ ) and the elastic properties of the host and the inclusion, that allow to calculate isomekes (Gilio 74 et al. 2021). An isomeke is a line in the pressure-temperature space along which the volume strain 75 of the inclusion and the surrounding host are the same (Campomenosi et al. 2022). Then, 76 isomekes are used to determine the over- or under-pressure conditions developed in the 77 inclusion, and, therefore, the final inclusion pressure. Thus, to utilize omphacite inclusions as 78 geothermobarometry markers in metamorphic rocks, detailed analyses of the atomic dynamics of 79 omphacite as a function of chemical composition, temperature and pressure is required. However, 80 Raman-scattering data on omphacite are available only at ambient conditions, mainly in 81 comparison with other chain silicates (Buzatu and Buzgar 2010; Andò and Garzanti 2014; Franz et 82 al. 2014). 83

Here, as a very first step to probe the potential of omphacite to be used as a mineral inclusion in elastic geothermobarometry, we report the Raman spectra of omphacite crystals with and without *M*-site cation order, while having the same composition  $^{-1}Jd_{45}Di_{55}$ , at pressures up to  $^{-10}$  GPa, along with Hartree-Fock/Density-Functional-Theory (HF/DFT) simulations on fully ordered Jd<sub>50</sub>Di<sub>50</sub> omphacite in the same pressure range.

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# Materials and methods

91 Sample description

We used omphacite crystals from Münchberg Mass, Bavaria (Matthes and Schmidt 1974; O'Brien 1993), which belong to the same omphacite sample that has been previously analyzed with infrared spectroscopy by Boffa Ballaran et al. (1998a, b) (sample 74AM41) and by highpressure single-crystal XRD by Pandolfo et al. (2012a, b). We selected single crystals with low iron content to avoid further chemical effects on the Raman spectra (Katerinopoulou et al. 2008). For

97	in situ high-pressure Raman spectroscopic measurements, we used two natural samples (natural-8
98	and natural-11), which exhibited a high degree of cation order (Boffa Ballaran et al. 1998a), and
99	two samples annealed at 1173 K for 168 h (annealed-3 and annealed-7) to achieve a state of long-
100	range cation disorder. The size of samples natural-8, natural-11 and annealed-3 was approximately
101	90×150×200 $\mu$ m <sup>3</sup> , whereas size of sample annealed-7 was ~50×50×60 $\mu$ m <sup>3</sup> .
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103	Chemical composition
104	Wavelength-dispersive electron microprobe analyses (EMPA) were performed on the very
105	same specimens used in the high-pressure Raman-scattering experiments. After conducting the
106	Raman experiments, the omphacite crystals were embedded into epoxy resin, polished, and
107	carbon coated for electron microprobe analyses. The chemical composition was determined with a
108	Jeol 8200 Super Probe electron microprobe at the Department of Earth Sciences of the University
109	of Milan, operating at an acceleration voltage of 15 kV and beam current of 5 nA (beam size = $3$
110	$\mu$ m). A ZAF correction (Z = atomic number, A = absorption coefficient, F = fluorescence coefficient),
111	using the PAP routine (Pouchou and Pichoir 1985) was applied. Standards, spectral lines, and
112	analytical crystals used were: grossular (Si, Al $K\alpha$ , TAP; Ca $K\alpha$ , PET), ilmenite (Ti $K\alpha$ , PET), metallic Cr
113	(CrK $\alpha$ , LIF), fayalite (FeK $\alpha$ , LIF), rhodonite (MnK $\alpha$ , LIF), forsterite (MgK $\alpha$ , TAP), omphacite (NaK $\alpha$ ,
114	TAP) and K-feldspar (KK $\alpha$ , PET).

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# 116 Single-crystal X-ray diffraction

Single-crystal XRD experiments were performed at the Department of Earth Sciences of the University of Milan to determine the degree of long-range M-site cation ordering in the crystals selected for high-pressure Raman spectroscopy. Samples natural-8, natural-11, annealed-3 were analyzed with a Xcalibur-1-Oxford diffractometer equipped with a MoKα sealed tube source and a

charge-coupled device (CCD) detector. Due to its smaller size (see above), sample annealed-7 was 121 analyzed with a Rigaku XtaLAB Synergy-S diffractometer equipped with a HyPix-6000HE HPC area 122 detector, using a monochromatic X-ray incident beam with an energy of 17.4 keV ( $\lambda$ =0.71073 Å, 123 i.e., MoK $\alpha$ ), which is characterized by a higher brilliance. Structure refinements were performed 124 using the program SHELXL (Sheldrick 2015). The structure of ordered and disordered omphacite 125 was refined in P2/n and C2/c space groups, respectively, starting from the models provided by 126 Pavese et al. (2000) and McCormick (1986). Scattering curves were taken from the International 127 Tables for Crystallography (Wilson 1995). Neutral vs. ionized scattering factors were used to refine 128 the occupancy for all sites that are not involved in chemical substitutions (i.e. O sites and <sup>T1</sup>Si in 129 P2/n) (Hawthorne et al. 1995), while scattering factors of ionized atoms were used for <sup>T2</sup>Si and the 130 six-coordinated M1 and M11 sites alongside the eight-coordinated M2 and M21 sites. The site 131 partitioning was obtained using the refinement model proposed by Boffa Ballaran et al. (1998a) 132 and the chemical constraints from our EMPA data. Crystallographic information files (CIFs) are 133 134 available as Online Materials.

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# 136 High-pressure Raman spectroscopy

Raman spectra of the four selected crystals were collected using a Horiba Jobin-Yvon S.A.S. 137 T64000 triple monochromator spectrometer at the Mineralogisch-Petrographisches Institut, 138 Universität Hamburg, equipped with an Olympus BX41 confocal microscope and a Symphony 139 liquid-N<sub>2</sub>-cooled CCD detector. Raman scattering was excited with the 514.532-nm or the 488.0-140 nm lines of a Coherent Innova 90C Fred Ar<sup>+</sup> laser, using an Olympus LM Plan FLN 50× long-working 141 distance objective with a numerical aperture of 0.5. The spectrometer was calibrated to the  $T_{2g}$ 142 mode at 520.5 cm<sup>-1</sup> of a Si wafer. The spectral resolution was ~2 cm<sup>-1</sup>, while the instrumental 143 accuracy in determining the peak position was  $\sim 0.35$  cm<sup>-1</sup>. 144

In situ high-pressure Raman experiments were conducted with Boehler-Almax diamond 145 anvil cells equipped with diamonds with a culet diameter of 600 µm. The sample chamber was 146 made by a stainless-steel gasket with a thickness of 0.25 mm, which was initially indented to  $\sim$  95 147 µm and then drilled with an Almax-easyLab spark eroder with tungsten carbide tips, to obtain a 148 hole with a diameter of 300 µm. A 16:3:1 methanol-ethanol-water mixture was used as a 149 pressure-transmitting medium, which remains hydrostatic up to 10.5 GPa (Angel et al. 2007). The 150 pressure P was determined via the ruby photoluminescence R1 line (Munro et al. 1985) with a 151 precision of ~ 0.1 GPa. 152

Parallel polarized Raman spectra (polarization of incident light **E**<sub>i</sub> parallel to the polarization 153 of the scattered light  $E_s$ ) were collected in the range 50-1250 cm<sup>-1</sup> with acquisition times of 60 or 154 180 s averaged over three accumulations. The variations in the relative intensities indicated that 155 the samples were oriented with the crystallographic c axis approximately parallel (natural-8 and 156 annealed-3) or perpendicular (natural-11 and annealed-7) to  $E_i$ ; the orientational dependence of 157 158 the Raman intensities was established by measuring an omphacite grain for which single-crystal 159 XRD showed that the prism axis coincides with the crystallographic **c** axis. In both cases, the direction of the propagation of the incident light  $\mathbf{k}_i$  was  $\perp$  to  $\mathbf{c}$ . Therefore, in ( $\mathbf{c} \mid \mid \mathbf{E}_i \mid \mid \mathbf{E}_s$ ) 160 scattering geometry the Raman scattering is dominated by the (zz)-component of the Raman 161 polarizability tensor, whereas in ( $\mathbf{c} \perp \mathbf{E}_i \mid | \mathbf{E}_s$ ) scattering geometry the Raman spectrum is 162 determined by (xx)- and/or (yy)-components of the polarizability tensor, depending on the 163 orientation of the **b** axis with respect to  $\mathbf{k}_{i}$ . At each pressure step, the Raman scattering from the 164 diamond anvils was collected to estimate the degree of depolarization of Raman scattering due to 165 166 stress-induced changes in the optical properties of the diamond anvils (Mihailova et al. 2019). In addition, a background spectrum was measured aside from the sample while keeping the same 167 focus, and subsequently subtracted from the sample spectrum to eliminate the contribution from 168

169	the pressure-transmitting medium. Then the spectra were reduced by the Bose-Einstein
170	population factor to account for the temperature dependence of the Raman intensities, using the
171	relation: $I_{reduced} = I_{measured} / [(e^{\hbar\omega/k_BT} - 1)^{-1} + 1]$ , where $\hbar$ , $k_B$ , $\omega$ , and $T$ are the reduced
172	Planck's constant, Boltzmann's constant, phonon angular frequency (in s <sup>-1</sup> ) and temperature (in K),
173	respectively. The peak positions, full widths at half maximum (FWHMs) and integrated intensities
174	were determined by fitting the temperature-reduced spectra with pseudo-Voigt peak-shape
175	functions $PV=qL+(1-q)G$ , where L and G are Lorentzian and Gaussian functions and q is a variable
176	weighting coefficient. The OriginLab 2022 software package was used for data evaluation.

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# 178 Computational methods

Ab initio hybrid HF/DFT simulations have been conducted with the CRYSTAL17 software 179 (Dovesi et al. 2018) by using the WC1LYP functional. This functional has been shown to correctly 180 reproduce the elastic and vibrational properties of crystals (e.g. Prencipe et al. 2011, 2012; 181 Prencipe 2012, 2018; Stangarone et al. 2016). The WC1LYP functional is based on the generalized 182 gradient approximation (GGA) exchange functional WC (Wu and Cohen 2006), mixed with 16% of 183 the exact non-local Hartree-Fock exchange, and the Lee-Yan-Parr (LYP) correlation functional (Lee 184 et al. 1988). The grid for the numerical evaluation of the DFT exchange-correlation functionals was 185 chosen by the keyword XLGRID of CRYSTAL17 (Dovesi et al. 2018) and corresponds to a total of 186 900,780 points in the unit cell. A measure of the numerical accuracy provided by such a grid is the 187 evaluation of the total number of electrons in the unit cell, by the numerical integration of the 188 electron density over the cell volume. For omphacite, we obtained 415.99999 electrons out of 416 189 for the reference volume at the static limit (no zero-point and thermal pressures due to vibrational 190 effects included, see Prencipe et al. 2011) and at a static pressure of 0 GPa. The localized 191 contracted atomic basis sets used were: Na 8-511(1d)G; Al 85-11(1d)G; Ca 865-11(3d)G; Mg 85-192

11(1d)G; Si 88-31(1d)G; O 84-11(1d1d)G (Sophia et al. 2014; Valenzano et al. 2006). Within the 193 CRYSTAL code, the accuracy in evaluating the Coulomb and Hartree-Fock exchange series is 194 controlled by the keyword TOLINTEG, for which we set the four T1, T2, T3 and T4 parameters to 9 195 and T5 to 22 (Dovesi et al. 2018). The diagonalization of the Hamiltonian matrix was performed at 196 30 independent k vectors in the reciprocal space (Monkhorst net; Monkhorst and Pack 1976) by 197 setting the shrinking factor IS to 4 (Dovesi et al. 2018). Unit-cell parameters and fractional 198 coordinates were optimized by analytical gradient methods, as implemented in CRYSTAL17 199 (Civalleri et al. 2001; Dovesi et al. 2018). Geometry optimization was considered converged when 200 201 each component of the gradient (TOLDEG parameter in CRYSTAL17) was smaller than 0.00003 Hartree/Bohr and displacements (TOLDEX) were smaller than 0.00012 Bohr with respect to the 202 previous step. Simulations of the Raman spectra of omphacite at the static limit have been 203 performed at pressures ranging from 0 to 6 GPa (with step of 2 GPa from 0 to 4 GPa and then a 204 step of 1 GPa from 4 to 6 GPa) by using the keyword EXTPRESS. The wavenumbers of all normal 205 modes were calculated at the  $\Gamma$  point within the limit of the harmonic approximation, by 206 diagonalizing a mass-weighted Hessian matrix, whose elements are the second derivatives of the 207 full potential of the crystal with respect to the mass-weighted atomic displacements (Pascale et al. 208 2004). Furthermore, in order to directly compare the computational data with the experimental 209 ones, we followed the procedure reported in Ghignone et al. (2023) for the estimate of EoS 210 parameters at a temperature of 298 K. 211

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## Results

214 Crystal chemistry

The EMPA results for sample natural-8, annealed-3, and annealed-7 are reported in Table 1. Within uncertainties the compositions of all three specimens are the same as those reported by

Boffa Ballaran et al. (1998a). Sample natural-11 was unfortunately lost during the polishing procedure, so in the subsequent analyses we assumed that it has the same chemical composition of sample natural-8.

The structure of samples natural-8 and natural-11 is refined in P2/n, indicating chemical 220 ordering, whereas samples annealed-3 and annealed-7 exhibit C2/c space symmetry typical of 221 chemically disordered omphacites. The unit-cell parameters for all samples are given in Table 2, 222 along with those calculated by HF/DFT simulations for fully ordered Jd<sub>50</sub>Di<sub>50</sub> omphacite. The M1 223 site is commonly occupied by small cations such as Mg<sup>2+</sup>, Al<sup>3+</sup>, Fe<sup>3+</sup> and Fe<sup>2+</sup>, while larger cations 224 such as Ca<sup>2+</sup> and Na<sup>+</sup> are at the M2 site. In the case of chemical order-disorder phenomena, the 225 site occupancy factors are used to calculate the order parameters  $Q_{M1}$  and  $Q_{M2}$  (Carpenter et al. 226 1990a) as follows: 227

$$Q_{M1} = \frac{\left|\frac{(Al + Fe^{3+})_{M1} - (Al + Fe^{3+})_{M11}}{(Al + Fe^{3+})_{M1} + (Al + Fe^{3+})_{M11}}\right| + \left|\frac{(Mg + Fe^{2+})_{M1} - (Mg + Fe^{2+})_{M11}}{(Mg + Fe^{2+})_{M1} + (Mg + Fe^{2+})_{M11}}\right|}{2}$$
$$Q_{M2} = \frac{\left|\frac{Na_{M2} - Na_{M21}}{Na_{M2} + Na_{M21}}\right| + \left|\frac{Ca_{M2} - Ca_{M21}}{Ca_{M2} + Ca_{M21}}\right|}{2}$$

In natural samples the long-range order parameter at M1 is commonly twice larger than that at M2, i.e.  $Q_{M1} \sim 2Q_{M2}$ . (Boffa Ballaran et al. 1998a; Skelton and Walker 2015). The refined values of the site occupancy factors and subsequently calculated order parameters (see Table 3) are in a good agreement with those reported by Boffa Ballaran et al. (1998a). Sample natural-8 is slightly less ordered than natural-11 (Table 3). In Table S1 in the Online Materials the bond distances and the polyhedral volumes are reported.

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# 235 Raman scattering at ambient conditions

According to group theory (e.g., Kroumova et al. 2003), at the Brillouin-zone centre there are  $\Gamma_{opt} = 14A_g + 13A_u + 16B_g + 14B_u$  optical phonon modes in chemically disordered omphacite

(C2/c) and  $\Gamma_{opt} = 28A_g + 27A_u + 32B_g + 30B_u$  in chemically ordered omphacite (P2/n). The A<sub>u</sub> and B<sub>u</sub> 238 are infrared-active, while Ag and Bg modes are Raman-active. Only Ag modes should appear in 239 parallel polarized Raman spectra measured when the pyroxene prism axis is parallel or 240 perpendicular to the polarization of the incident light, as it was in our scattering geometry. 241 Therefore, 28 peaks are expected in the Raman spectra of samples natural-8 and natural-11 and 242 14 peaks in the spectra of samples annealed-3 and annealed-7. Indeed 14 peaks were observed for 243 chemically disordered omphacite (see Figure 2 and Table 4), but only 22 peaks were resolved for 244 chemically ordered omphacite. This can be due to peak overlapping (e.g. the peaks near 1021 and 245 1029 cm<sup>-1</sup>) or weak diagonal components (e.g. the peaks near 158, 281 and 452 cm<sup>-1</sup>) of the 246 Raman polarizability tensor, as indicated by the results of HF/DFT calculations. The calculated 247 phonon wavenumbers of  $A_g$  modes in P2/n at zero pressure match very well the observed Raman-248 peaks positions at atmospheric pressure (Table 4). The calculated and observed relative peak 249 intensities in the parallel-polarized Raman spectra are also in a very good agreement (see Fig. 3a). 250 251 A complete list of the Raman active phonon modes in fully ordered Jd<sub>50</sub>Di<sub>50</sub> omphacite as well as the Raman polarizability tensors derived from HF/DFT calculations at P = 0 GPa and T = 0 K are 252 given in the Online Materials (Table S2). 253

The assignment of each Raman peak to a definitive atomic vibration can be achieved by direct inspection of the calculated atomic vector displacements. The strongest experimentally observed Raman signals are then assigned as follows:

- The  $A_g$  modes in the range 950 and 1050 cm<sup>-1</sup> are related to the Si-O<sub>nb</sub> bond stretching (O<sub>nb</sub> = non-bridging oxygen, which is not shared between two TO<sub>4</sub>). Chemically disordered omphacite exhibits one peak observed near 1017 cm<sup>-1</sup>, whereas ordered omphacite is characterized by the splitting of this peak into two components at about 1007 and 1029 cm<sup>-1</sup>, due to the reduction of the translational symmetry.

262	- The strong peak at 688 cm <sup>-1</sup> , typical of both chemically ordered and disordered
263	omphacite, arises from A <sub>g</sub> mode comprising Si–O <sub>b</sub> –Si bond bending vibrations.
264	- The $A_g$ modes at 375, 382, 402 and 411 cm $^{-1}$ are related to M1O $_6$ and M11O $_6$ bending and
265	$TO_4$ rotation around <b>c</b> .
266	- The A <sub>g</sub> mode at 337 cm <sup>-1</sup> in ordered omphacite is related to M1-cation vibrations parallel
267	to the <b>b</b> axis and TO <sub>4</sub> rotation around the <b>c</b> axis. Note that the tilt angle is defined as the out-of-
268	plane tilting of the basal face of the tetrahedra with respect to the plane (100) (Cameron et al.
269	1973).
270	- The $A_g$ mode at 300 cm <sup>-1</sup> in chemically ordered omphacite is due to M1-cation and $^{M2}$ Na
271	vibrations parallel to <b>b</b> .
272	- The $A_g$ mode at 271 cm <sup>-1</sup> is associated with vibrations of M-site cations along <b>b</b> and $TO_4$
273	translation along <b>a</b> .
274	- The $A_g$ mode at 215 cm <sup>-1</sup> is mainly related to SiO <sub>4</sub> rotation around <b>a</b> .
275	Sketches of selected vibrational modes are presented in Figure S1 in the Online Materials.
276	It should be emphasized that the peak FWHM is affected by the ordering of the cations at
277	the six- and eight-coordinated sites and it increases with cation disorder, as reported by
278	Katerinopoulou et al. (2008).
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280	Pressure dependence of Raman spectra
281	Figure 4 shows the pressure evolution of Raman spectra of chemically ordered omphacite.
282	No change in the number of observed peaks occurs in the investigated pressure range. In
283	accordance with theory, the strongest peak at 688 cm <sup>-1</sup> gradually increases in wavenumber upon
284	pressure increase, while its FWHM increases slightly, but within the spectral resolution (see Figure
285	5). The former is due to the pressure-induced decrease in the atomic distances and the

subsequent increase in the effective interatomic force constants, whereas the latter is related to 286 the fact that the phonon decay and hence the peak FWHM should not change with pressure in the 287 absence of structural transformations. Figure 5 also demonstrates that the phonon compressibility 288  $\beta_{\omega} = \frac{1}{\omega_{0}} \frac{d\omega}{dP}$  of the A<sub>g</sub> mode near 688 cm<sup>-1</sup> does not depend on the chemical order, which makes 289 this peak a potential candidate for wavenumber-pressure calibration. The linear fit  $\omega(P) = \omega_0 + \frac{d\omega}{dP}$ 290 P to the experimental data points gives  $\omega_0 = 680.3(2)$ ,  $\frac{d\omega}{dP} = 3.32(3)$ , and  $\beta_{\omega} = 0.00488(5)$  GPa<sup>-1</sup>. 291 Then using the relation  $P = \frac{\omega - 680.3}{3.32}$ , the pressure can be determined with an accuracy of ~0.1 GPa. 292 Further experiments can improve the statistics of  $\omega(P)$  data points, which in turn will reduce the 293 errors in the coefficients of polynomial fit and consequently, will reduce the propagating 294 uncertainties in  $P_{inc}$ , that is the residual pressure. 295

A few Raman peaks for chemically ordered omphacite show anomalous behavior near 4.5-296 5 GPa: (i) the pressure dependence of the average wavenumber of the two components related to 297 the Si-O<sub>nb</sub> bond stretching shows a discontinuity near 5 GPa (Figure 6a), although the  $\frac{d\omega}{dR}$  value 298 remains nearly the same above 5 GPa; (ii) the slope of  $\omega(P)$  of the peaks near 375 cm<sup>-1</sup> and 337 cm<sup>-1</sup> 299 <sup>1</sup>, both arising from M1O<sub>6</sub> vibrations and SiO<sub>4</sub> rotations, changes around 4.5 GPa (Figure 6b and c) 300 and above this pressure the corresponding phonon compressibility equals to that of disordered 301 omphacite; (iii) the FWHM of the peak near 271 cm<sup>-1</sup>, generated by M-cation vibrations and SiO<sub>4</sub> 302 translations, becomes smaller above 5 GPa (Figure 6d). 303

In order to simulate the pressure dependence of the phonon modes, unit-cell parameters (reported in Table 2) and fractional coordinates were optimized by analytical gradient methods and the equation of state (EoS) at room temperature was modelled. As can be seen in Table 5, the calculated EoS parameters are in good agreement with previously published experimental results (Pavese et al. 2001; Pandolfo et al. 2012b; Zhang et al. 2016).

The pressure dependence of the phonon wavenumbers of fully ordered omphacites was simulated in the pressure range 0-11 GPa, while the Raman intensities were calculated at 0, 2, 4, 5 and 6 GPa. The correspondence between experimental and calculated spectra at high pressure is very good (Figure 3b). The calculated  $\frac{d\omega}{dP}$  of the main peaks also match very well the corresponding experimentally observed  $\omega(P)$  slopes (see Figure 7).

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# Discussions

The hybrid HF/DFT simulations reveal that the anomalous  $\omega(P)$  behavior of the peak ~337 316 cm<sup>-1</sup> for chemically ordered omphacite can be explained by enhanced phonon-phonon 317 interactions near 4.7 GPa. The ab initio simulations reveal that in this spectral range there are 318 actually two Ag modes, at 337 and 349 cm<sup>-1</sup>. Having the same symmetry and similar energies, 319 these Ag modes can interact and exchange intensity via the so-called avoided intersection, also 320 referred to as energy-level anticrossing (Dresselhaus et al. 2008); that is, two modes of the same 321 symmetry approach one another in energy but without energy crossing. Indeed, the wavenumbers 322 of the two  $A_g$  modes become very close to each other at a volume of about 413  $\textrm{\AA}^3$  (see the insert 323 in Figure 7), corresponding to a pressure of 4.7 GPa, while at higher pressure they diverge. These 324 two Raman peaks could be experimentally distinguished for sample natural-8 when measured in 325 air, but not under high pressure in DAC, so that only one peak at about 337 cm<sup>-1</sup> was used in the 326 fitting model for high-pressure spectra. However, as can be seen in Figure 7, the experimental and 327 simulated pressure trend for the Ag mode near 337 cm<sup>-1</sup> are in a very good agreement. 328

It should be emphasized that above the pressure of  $A_g$ -phonon anticrossing, the phonon compressibility of the mode near 337 cm<sup>-1</sup> in chemically ordered omphacite becomes the same as that in chemically disordered omphacite. The same is valid for the  $A_g$  mode near 375 cm<sup>-1</sup>. Both phonon modes involve SiO<sub>4</sub> tilting and M-cation vibrations and the change in  $\beta_{\omega}$  should be related

to a change in the M-O-Si interactions. External pressure alone cannot mobilize M-site cations to 333 form a chemically disordered state above 4.5 GPa. However, pressure can trigger a redistribution 334 of the local elastic strains related to the difference in the ionic sizes of the M-site cations, leading 335 to unified  $MO_6$  volumes and consequently, averaged effective M-O-Si force constants that appear 336 to be the same as a state of site-occupancy disorder. In addition, both the zenithal and azimuthal 337 components of the out-of-plane basal tilts of tetrahedra, which change as a function of the degree 338 of order (for exact definitions see Boffa Ballaran et al. 1998a and Cámara et al. 1998), may also 339 change as a consequence of compression, related also to the unified volume compressibility of the 340 M1 and M11 octahedra. Chemical disorder is difficult to implement in ab initio simulations. 341 However, based on the combined experimental and HF/DFT results presented here, we speculate 342 that the pressure-enhanced phonon-phonon interactions near 4.5-5 GPa trigger the 343 homogenization of local elastic strains associated with chemical order at the (M1, M11) and (M2, 344 M21) sites, respectively. This phenomenon should be responsible also for the anomaly in the 345 averaged  $\omega(P)$  of the Si-O<sub>nb</sub> bond stretching mode observed near 1017 cm<sup>-1</sup>, because M-O 346 interactions have a secondary effect on the energy of this phonon mode. The experimentally 347 observed reduction of the FWHM of the 271-cm<sup>-1</sup> peak, resulting primarily from M-cation 348 vibrations, is also consistent with the assumption for a smaller dispersion in the MO<sub>6</sub> size above 349 ~4.5 GPa. 350

For ordered omphacite a slight anomaly in the pressure dependence of the M1,110<sub>6</sub> volumes has been reported at about 4 GPa by Pandolfo et al. (2012b), which was attributed to a pressure-induced change of the tilt angle for both T10<sub>4</sub> and T2O<sub>4</sub> tetrahedra. This observation corroborates with the anomalies reported here for the phonon modes at about 1017, 375, 337, and 271 cm<sup>-1</sup>. Moreover, we show that the position of the well resolved Raman peak at 688 cm<sup>-1</sup>, generated by Si-O<sub>br</sub>-Si bond bending A<sub>g</sub> mode, exhibits a strong pressure dependence, which is not

affected by the presence or absence of chemical order. That is, the temperature-induced change 357 of the translational symmetry due to site-occupancy order-disorder processes does not influence 358 the wavenumber of the Si-O<sub>br</sub>-Si bond bending and therefore, this phonon mode properly reflects 359 the strained state of an omphacite inclusion associated with the metamorphic conditions. Besides, 360 HF/DFT calculations indicate that this mode exhibits the strongest polycrystalline isotropic Raman 361 intensity (see Table 4), ensuring that the peak near 688 cm<sup>-1</sup> can be experimentally detected in any 362 random orientation of an omphacite inclusion. Thus, as garnets do not show Raman scattering in 363 this spectral range (Kolesov and Geiger 1998), this peak seems to be a promising candidate to be 364 used for estimating the residual pressure for omphacite-in-garnet inclusion-host systems. On the 365 other hand, the FWHM of this peak does not change with pressure, but it is indicative of the 366 absence or presence of chemical cation order. When an instrumental resolution of 2 cm<sup>-1</sup> is used, 367 disordered omphacite shows a FWHM for the 688-cm<sup>-1</sup>-peak around 16 cm<sup>-1</sup>, whereas for 368 omphacite with  $Q_{M1} \cong 2Q_{M2} \cong 0.8$  the peak FWHM is around 12 cm<sup>-1</sup>. A cation order parameter 369  $Q_{M1} \cong 0.83$  corresponds to a closure temperature of about 600 K (Carpenter et al. 1990a). 370 Therefore, a FWHM above 12 cm<sup>-1</sup> implies that the omphacite inclusion has been trapped at T > 371 372 600 K. However, the Raman spectrum of omphacite and in particular the position of the wavenumber of the Si-O-Si bond bending mode depends on the chemical composition (Wang et al. 373 2001; Prencipe et al. 2012). Hence, further comprehensive studies on the compositional effect on 374 the phonon compressibility for omphacites with diverse chemistry is required before utilizing 375 omphacite-in-garnet systems for Raman elastic geothermobarometry applications. 376

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## Implications

379 Conventional thermobarometry techniques are commonly used in geological sciences for 380 determining pressure and temperature conditions of mineral formation. These methods rely on

mineral assemblages and chemical compositions, providing valuable insights into metamorphic 381 processes, but they are limited by assumptions of chemical thermodynamic equilibrium under 382 hydrostatic pressure (Powell and Holland 2008; Gilio et al. 2022). On the other hand, Raman 383 elastic thermobarometry provides a non-destructive in situ approach for estimating pressure and 384 temperature conditions based on the elastic properties of an inclusion and its host. This technique, 385 which is mostly independent of chemical equilibration, holds great promise for providing accurate 386 thermobarometric data, but it is restricted to a few host-inclusion systems. The lack of calibrations 387 of Raman shifts with pressure, as well as of elastic properties at variable pressure and 388 temperature, requires further experiments to improve and validate this method in geological 389 studies (Kohn et al. 2023). 390

The present study represents the first step towards enabling the use of omphacite 391 inclusions for elastic geothermobarometry purposes using Raman spectroscopy. We demonstrate 392 that knowing the EoS and the composition of the inclusion, determined by EMPA analysis, it will 393 394 be possible to a first approximation to estimate its residual pressure using Raman spectroscopy. 395 For this purpose, one should comprehensively study the effect of the relevant thermodynamic variables on the phonon wavenumbers, namely elastic stress (or hydrostatic pressure), chemical 396 composition, degree of chemical order. That is, one should: (i) establish relationships between 397 phonon wavenumber and external elastic stresses/hydrostatic pressure, i.e.  $\omega(P)$ ; (ii) a robust 398 relationship between the chemical composition, denoted here for simplicity with  $\{x\}$ , and the 399 phonon wavenumber, i.e.  $\omega({x})$ , (iii) the effect of chemical composition on phonon 400 compressibilities, i.e.  $\beta_{\omega}({x})$ ; and (iv), the effect of chemical order on  $\omega$  and  $\beta_{\omega}$ . At the current 401 stage, this can only be done using empirical calibrations such as  $\omega(P) = \omega_0 + (d\omega/dP)P$  for selected 402 phonon modes, which have been proved to be insensitive to the degree of chemical order, so that 403 404 at least one parameter influencing the phonon frequencies can be eliminated. Further ab initio

calculations will enable, at least for the ordered structure, to calculate the Grüneisen tensor for a 405 correct estimate of the residual strains (see Murri et al. 2018 for further details). The subsequent 406 key point to enable the use of omphacite for elastic geobarometric purposes is the contrast in 407 their elastic properties with respect to those of the host. Garnet is a significantly stiffer material 408 compared to omphacite. Synthetic  $Prp_{60}AIm_{40}$  garnet has a bulk modulus ( $K_{T0}$ ) of 167.2 (17) GPa 409 (Milani et al. 2015); on the other hand, omphacite has a lower  $K_{T0}$ , ranging from 122(1) to 119(2) 410 GPa (Pandolfo et al. 2012a, b), thus implying a larger relaxation of omphacite volume than its host 411 412 garnet and the building up of a significant remanent inclusion pressure from which it is possible to calculate the entrapment conditions (Korsakov et al. 2009). Furthermore, omphacite has a softer 413 bulk modulus compared to zircon, but it is stiffer than quartz. The combination of their 414 thermoelastic properties contributes to the isomekes trends: the isomekes of omphacite resemble 415 those of quartz, making it a potential barometer. 416

To the aim of recalculating the entrapment conditions, knowing the elastic behavior as a 417 function of composition and ordering state becomes crucial. Previous high-pressure single-crystal 418 419 XRD analyses on omphacite from the same location as that of omphacite used in our study revealed  $K_{T0}$  = 122(1) GPa and K' = 5.1(3) for ordered omphacite, and  $K_{T0}$  = 119(2) GPa and K' = 420 5.7(6) for the disordered omphacite (Pandolfo et al. 2012a, b), i.e. within the uncertainties 421 ordered and disordered omphacite exhibit a very similar volume compressibility up to 7.5 GPa 422 (Table 5). Regarding the composition, Pandolfo et al. (2012a) discussed the bulk modulus 423 dependence on the composition in terms of jadeite and diopside solid solution and proposed the 424 following equation: 425

$$K_{T0} = 106(1)GPa + 0.28(2) \times Jd_{(mol\%)}$$

426 Our preliminary results based on the calculated EoS of omphacite indicate that the slope of 427 omphacite-in-garnet isomeke is close to the slope of quartz-in-garnet isomeke (Figure 8) which

makes omphacite a good candidate for elastic geobarometry. Here we present an example of 428 garnet host with omphacite inclusions because it is a widespread host-inclusion system in several 429 geological settings. For example, omphacite in garnet could be extremely useful in low-silica 430 systems like metabasites (the most widespread lithotype of eclogites) where quartz is not stable 431 and omphacite becomes a suitable common phase for retrieving metamorphic peak conditions. 432 Furthermore, omphacite-in-garnet host-inclusion system will provide fundamental hints for low-433 pressure and high-temperature geological settings where both quartz and zircon inclusions in 434 435 garnet have considerable shortcomings. The first comes from the high temperature  $\alpha$ -quartz  $\leftrightarrow \beta$ quartz phase transition and related effects on the atomic dynamics (e.g., Murri et al. 2019a) and 436 437 the latter because of the high-temperature elastic reset in zircon (Campomenosi et al. 2022).

While the slope of the isomekes and the different character of the phase transition provides significant advantages for omphacite-in-garnet with respect to quartz- and zircon-ingarnet because it extends significantly the PT space of potential applicability, however the large spacing between the isomekes, far greater than that of quartz and zircon, may be a significant shortcoming. Because of the large spacing, omphacite inclusions in garnet will require higher precision in determining  $P_{inc}$  from Raman spectroscopy.

In this paper we show that below ~4.5 GPa the phonon compressibility of MO<sub>6</sub> octahedra is larger for the chemically ordered omphacite than for its disordered counterpart, whereas above 4.5 GPa the elastic behavior is unified, probably because the external pressure leads to homogeneous distribution of the internal chemically-induced local stresses. The strong Raman peak near 688 cm<sup>-1</sup>, arising from Si-O<sub>br</sub>-Si bond bending, seems to be a promising Raman marker for residual pressure via its position, and indicative of closure temperature via its width. However, further comprehensive analysis of the effect of composition and the cation order degree on all

451	omphacite Raman-active modes and their pressure behavior is required to develop a protocol for
452	the proper utilization of omphacite-in-garnet systems in Raman elastic geothermobarometry.
453	
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## 668 Captions

- **Table 1.** Electron microprobe analyses in oxide wt% and chemical formulae in atoms per formula
- unit, based on six oxygen atoms
- **Table 2.** Unit-cell parameters of the analyzed crystals, compared to the ab initio calculated ones
- **Table 3.** Site occupancy factors, order parameters and mean bond distances (given in Å) refined
- 673 from single-crystal XRD data with chemical constraints
- **Table 4.** Calculated and experimental wavenumbers ( $\omega$ , in cm<sup>-1</sup>) of the A<sub>g</sub> modes and relative intensities normalized to 688 cm<sup>-1</sup> mode (*I*, in arbitrary units) of omphacite
- **Table 5.** Bulk modulus K<sub>0</sub>, its pressure derivative K', the zero-pressure volume V<sub>0</sub> as well as the
- order parameters  $Q_{M1}$  and  $Q_{M2}$ , pressure (P) and temperature (T) conditions of each study. The
- EoSs were determined using single-crystal XRD for all experimental studies except for Pavese et al.
- 679 (2001), who used powder XRD
- Figure 1. A structural fragment of ordered omphacite (space group symmetry P2/n). SiO<sub>4</sub> tetrahedra are shown in dark blue, M1 sites in orange, M11 sites in yellow, M2 in blue, and M21 in light blue. The structure was prepared using the VESTA software package (Momma and Izumi 2008).
- Figure 2. Raman spectra of chemically ordered (red and grey lines) and chemically disordered (blue and green lines) omphacite, collected in  $(c ||E_i||E_s)$  (samples natural-8 and annealed-3) or (c  $\pm E_i || E_s$ ) (samples natural-11 and annealed-7) scattering geometry at room temperature and atmospheric pressure. The signals marked by \* are not related to omphacite Raman-active phonon modes.
- Figure 3. Experimental and calculated Raman spectra of ordered omphacite (P2/n), at ~0 GPa (a) and at ~5 GPa (b).

Figure 4. Pressure evolution of  $(c||E_i||E_s)$  Raman spectra of chemically ordered omphacite (sample natural-11).

**Figure 5.** Pressure dependence of the wavenumber and FWHM of the strongest Raman peak at 688 cm<sup>-1</sup>. Filled symbols are related to the ordered omphacites, while open symbols to the disordered omphacites.

**Figure 6.** Pressure dependence of the wavenumber or FWHM of the peaks that show an anomaly at about 5 GPa in ordered omphacite, along with the corresponding trends for disordered omphacites: the average wavenumber of the two peaks near to 1007 and 1029 cm<sup>-1</sup> (a); the slope of the peaks at about 375 cm<sup>-1</sup> (b) and 337 cm<sup>-1</sup> (c) changes around 4.5 GPa; the FWHM of the peak near 271 cm<sup>-1</sup> becomes smaller above 5 GPa (d).

Figure 7. Experimental and calculated pressure dependence of the wavenumber of the  $A_g^{337}$  mode.

The slopes of the curves and the pressure at which they change are reported. The insert shows the calculated wavenumbers of the two  $A_g$  modes that are involved in the anticrossing.

**Figure 8.** Calculated isomekes for quartz, zircon and omphacite inclusion in an almandine host, obtained from the  $P_{inc}$  (equal to 0.0, 0.2, and 0.4 GPa) with the EoSFit–Pinc software (Angel et al. 2017b), using the equation of state of quartz (Angel et al. 2017a), zircon (Ehlers et al. 2022) and almandine (Angel et al. 2022), while the HF/DFT calculated EoS was used for omphacite (Jd<sub>50</sub>Di<sub>50</sub>). The α-β transition of quartz is represented in a red dashed line, line and the shaded vertical strip represents the *P2/n*-to-*C2/c* omphacite transition.

711 Table 1

	Natural-8	Annealed-3	Annealed-7
	(10 pt)	(5 pt)	(3 pt)
SiO <sub>2</sub>	56.31(26)	55.79(25)	54.83(45)
$AI_2O_3$	11.30(10)	11.24(6)	11.09(3)
TiO <sub>2</sub>	0.12(4)	0.14(3)	0.12(2)
$Cr_2O_3$	0.04(4)	0.04(3)	0.29(1)
FeO	2.56(6)	2.57(10)	2.41(7)
MnO	0.01(2)	0.01(2)	0.02(3)
MgO	9.87(11)	9.84(11)	9.46(12)
CaO	15.27(11)	15.27(6)	14.72(15)
Na <sub>2</sub> O	5.97(10)	5.99(15)	5.87(18)
K <sub>2</sub> O	0.01(1)	0.03(5)	0.00(1)
Total	101.45(36)	100.93(39)	98.57(89)
Si	1.966(3)	1.961(2)	1.962(1)
Al	0.465(3)	0.466(2)	0.470(3)
Ti	0.003(1)	0.004(1)	0.003(1)
Cr	0.001(1)	0.001(1)	0.008(1)
Fe <sup>2+</sup>	0.075(2)	0.076(3)	0.072(1)
Mn	0.000(0)	0.000(0)	0.001(1)
Mg	0.514(5)	0.516(5)	0.507(2)
Ca	0.571(5)	0.575(4)	0.567(1)
Na	0.404(6)	0.408(8)	0.409(9)
К	0.000(0)	0.002(2)	0.000(0)
Total	4.000(3)	4.007(3)	4.000(7)

713 Table 2

Sample	е	Natural-8	Natural-11	Annealed-3	Annealed-7	Calculated (HF/DFT)
Space gro	oup	P2/n	P2/n	C2/c	C2/c	P2/n
	a (Å)	9.5962(2)	9.5888(2)	9.6019(2)	9.5935(2)	9.614
	b (Å)	8.7897(2)	8.7852(2)	8.7781(3)	8.7725(2)	8.781
Unit-cell parameters	<i>c</i> (Å)	5.2586(2)	5.2587(1)	5.2539(2)	5.2508(1)	5.287
parameters	β (°)	106.789(2)	106.855(2)	106.692(2)	106.696(2)	107.22
	<i>V</i> (Å <sup>3</sup> )	424.65(2)	423.96(2)	424.17(2)	423.27(2)	426.3

_	Table	3						
	San	nple	Natural-8	Natural-11			Annealed-3	Annealed-
Space g		group	P2/n	P2/n			C2/c	С2/с
	T1	Si	1.000(19)	1.000(18)	Т	Si	0.9599(13)	0.9613(9)
		eps	14.000	14.000		Al	0.0401(13)	0.0387(9)
		eps <sub>0</sub>	14.000	14.000		eps	13.96(3)	13.961(17)
	<t1-0></t1-0>		1.6338(8)	1.6330(8)		eps <sub>0</sub>	14.000	14.000
	T2	Si	0.9741(13)	0.9756(13)		<t-0></t-0>	1.6333(10)	1.6319(5)
		Al	0.0259(13)	0.0244(13)	M1	Mg	0.235(3)	0.802(2)
		eps	13.97(3)	13.98(3)		Al	0.728(2)	0.102(2)
		eps <sub>0</sub>	14.000	14.000		Fe <sup>2+</sup>	0.037(2)	0.096(2)
	<t2< td=""><td>-0&gt;</td><td>1.6323(8)</td><td>1.6318(8)</td><td></td><td>eps</td><td>13.26(7)</td><td>13.43(5)</td></t2<>	-0>	1.6323(8)	1.6318(8)		eps	13.26(7)	13.43(5)
	M1	Mg	0.882(2)	0.894(2)		eps <sub>0</sub>	13.61(9)	13.56(9)
		Al	0.031(2)	0.015(2)	M2	<m1-0></m1-0>	2.0115(10)	2.0108(6)
		Fe <sup>2+</sup>	0.087(2)	0.091(2)		Ca	0.559(6)	0.560(3)
		eps	13.25(6)	13.30(7)		Na	0.441(6)	0.440(3)
		eps <sub>0</sub>	13.31(7)	13.37(8)		eps	16.03(14)	16.04(7)
	<m1-0></m1-0>		2.0761(8)	2.0789(7)		eps <sub>0</sub>	15.73(14)	15.65(12)
	M11	Mg	0.100(2)	0.081(2)		<m2-0></m2-0>	2.4892(10)	2.4871(5)
		Al	0.846(2)	0.867(2)	•			
		Fe <sup>2+</sup>	0.054(2)	0.052(2)				
		eps	13.60(6)	13.59(6)				
		eps <sub>0</sub>	13.52(7)	13.43(7)				
	<m1< td=""><td>1-0&gt;</td><td>1.9528(8)</td><td>1.9462(8)</td><td></td><td></td><td></td><td></td></m1<>	1-0>	1.9528(8)	1.9462(8)				
		$Q_{M1}$	0.828	0.864				
	M2	Ca	0.3951(19)	0.3629(19)				
		Na	0.6049(19)	0.6370(19)				
		eps	14.56(5)	14.27(5)				
		eps <sub>0</sub>	14.39(10)	13.97(12)				
	<m2-0></m2-0>		2.4725(8)	2.4714(8)				
	M21	Ca	0.7801(19)	0.8034(19)				
		Na	0.2199(19)	0.1966(19)				
		eps	18.02(5)	18.23(5)				
	eps <sub>0</sub>		17.86(10)	17.93(12)				
	<m21-0></m21-0>		2.5096(8)	2.5095(8)				
_	Q <sub>M2</sub>		0.397	0.453				
	<i>Note</i> : <i>e</i> constra	-	tron per site; <i>ep</i>	$s_0 = \text{electron period}$	er site,	calculated	d without chem	ical

717 Table 4

	Calculator					Meas	ured			
	Calculated (HF/DFT)		Nat	ural-8	Natu	ıral-11	Anne	aled-3	Anne	aled-7
Mode	ω	Ι	ω	Ι	ω	Ι	ω	Ι	ω	1
$A_g$	142.34	10.69	146(1)	95.39	147(2)	44.38	146(1)	107.56	Not of	oserved
$A_g$	157.86	0.85	Not r	esolved	Not re	esolved				
$A_g$	190.94	5.63	194(3)	60.21	193(2)	18.97	189(1)	5.22	187(2)	1.72
$A_g$	215.48	6.47	220(3)	44.88	220(1)	77.24	225(2)	18.79	215(2)	67.30
$A_g$	229.60	1.67	238(2)	39.22	Not o	oserved				
$A_g$	271.38	20.65	270(1)	27.57	271(1)	55.52	270(2)	2.32	267(1)	16.83
$A_g$	281.19	3.09	Not r	esolved	Not re	esolved				
$A_g$	300.37	5.72	301(2)	18.53	303(2)	16.88				
$A_g$	329.34	44.44	Not r	esolved	Not re	esolved				
$A_g$	337.48	123.53	335(1)	313.85	338(1)	293.94				
$A_g$	348.91	29.83	354(2)	202.48	Not re	esolved	350(1)	963.71	349(1)	190.36
$A_g$	375.24	43.96	Not r	esolved	372(1)	352.11				
$A_g$	381.97	130.59	380(1)	425.63	Not re	esolved	385(1)	175.16	382(2)	74.46
$A_g$	401.91	8.08	399(3)	348.59	397(2)	37.35				
$A_g$	410.63	136.60	414(1)	172.85	413(2)	86.02	410(1)	192.69	407(1)	94.08
$A_g$	452.38	0.88	Not r	esolved	Not re	esolved				
$A_g$	519.33	20.57	514(1)	6.32	516(1)	166.91	516(1)	65.84	513(2)	57.85
$A_g$	544.95	23.77	541(15)	210.92	541(1)	46.30				
$A_g$	562.54	51.87	563(1)	294.75	563(1)	217.96	561(1)	167.02	561(1)	145.63
$A_g$	609.11	49.85	595(2)	41.25	601(2)	14.54	594(2)	11.33	598(1)	19.62
$A_g$	688.08	1000.00	680(1)	1000.00	681(1)	1000.00	679(1)	1000.00	678(1)	1000.00
$A_{g}$	734.64	17.16	737(2)	3.28	734(2)	126.12				
Ag	899.22	35.84	882(2)	90.90	894(3)	133.15	877(2)	75.67	881(6)	41.35
$A_g$	916.33	78.28	911(2)	110.11	Not re	esolved	905(3)	90.16	906(8)	15.61
Ag	1007.44	601.98	1007(2)	508.12	1008(1)	608.87	1016(1)	424.02	1017(1)	946.97
Ag	1020.81	215.24	Not r	esolved	Not re	esolved				
Ag	1028.96	840.22	1023(2)	816.55	1026(1)	518.84				
Ag	1067.45	25.90	1096(2)	211.75	1076(3)	164.86				

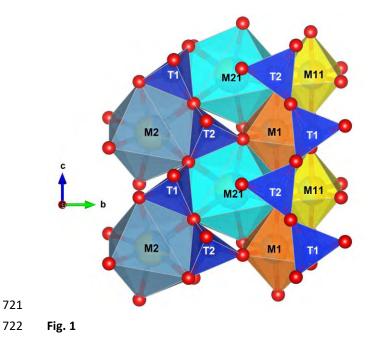
*Note*: the calculated intensities correspond to polycrystalline material, whereas the listed experimental intensities for natural-8 and annealed-3 correspond to the square of (zz)-component of the Raman polarizability tensor measured in scattering geometry ( $\mathbf{c} \mid \mid \mathbf{E}_s$ ), whereas the experimental intensities for natural-11 and annealed-7 correspond to a weighted sum of squared (xx)- and (yy)-components of the Raman polarizability tensor measured in scattering geometry ( $\mathbf{c} \mid \mathbf{E}_s$ ), with weight coefficients depending on the **b**-axis orientation with respect to the laser-beam propagation (random in our study).

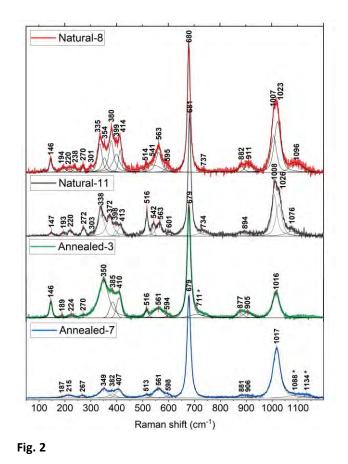
## 719 Table 5

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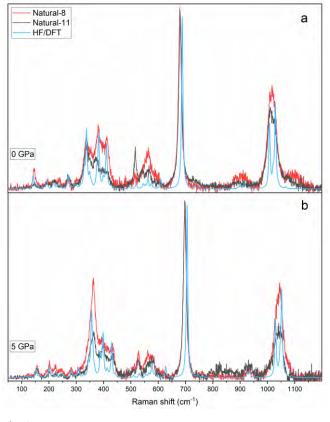
	Calculated	Pandolfo et	Pavese et	Zhang et
	(HF/DFT)	al., 2012b	al., 2001	al., 2016
Di (wt%)	50.00	48.0	47.4	51.5
Jd (wt%)	50.00	52.0	46.4	46.5
Ae (wt%)	0.00	0.0	6.2	2.0
K <sub>0</sub> (GPa)	119(2)	122(1)	117(3)	116(2)
K'	4.9(5)	5.1(3)	6.0(1)	4.3(2)
V <sub>0</sub> (Å <sup>3</sup> )	426.52(3)	421.43(4)	422.2(1)	423.9(3)
$Q_{M1}$	1	0.8956	0.84	0.81
$Q_{M2}$	1	0.4993	0.42	0.39
P (GPa)	0-6	0-7.5	0-13	0-47
T (°C)	25	25	25	25

*Note*:  $Q_{M1}$  and  $Q_{M2}$  of ordered omphacite are derived from ab initio simulations and experimentally determined in previous XRD studies (Di = diopside, Jd = jadeite, Ae = aegirine). For Zhang et al. (2016)  $Q_{M2}$  is calculated based on the site occupancy factor, while  $Q_{M1}$  as a function of the bond distances (Carpenter et al. 1990a).

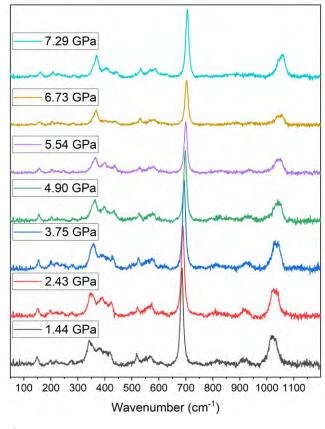






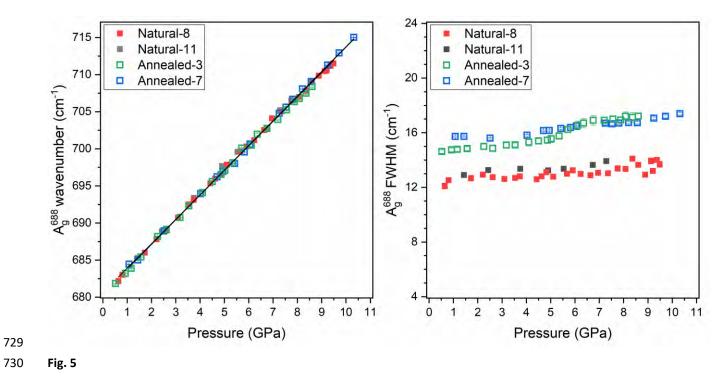


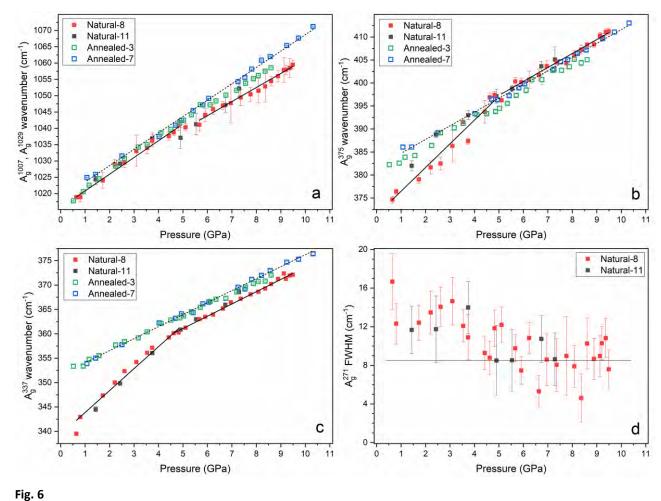




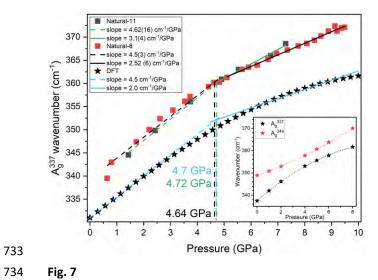


728 Fig. 4

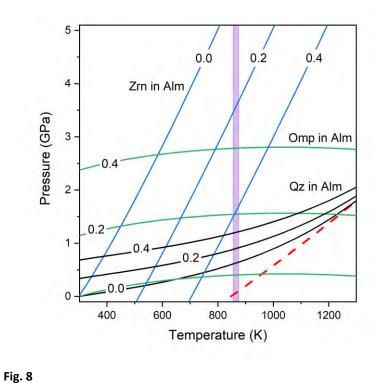




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