

Influence of chemistry on the pyroelectric effect in tourmaline

KATE D. HAWKINS*

Centre for Microscopy and Microanalysis, University of Queensland, Brisbane, Queensland 4072, Australia;
and School of Mechanical and Manufacturing Engineering, Queensland University of Technology,
P.O. Box 2434, Brisbane, Queensland 4001, Australia

IAN D. R. MACKINNON

Centre for Microscopy and Microanalysis, University of Queensland, Brisbane, Queensland 4072, Australia

HELMUT SCHNEEBERGER**

Institut für Kristallographie und Mineralogie, Ludwig-Maximilians-Universität, Theresienstraße 41, Munich, Germany

ABSTRACT

Pyroelectric coefficients were measured from a series of natural tourmaline crystals between ~170 and ~500 K to quantify the variation of the pyroelectric effect with chemical composition. The amount of Fe in tourmaline has a prominent influence on the pyroelectricity. Fe content linearly decreases the pyroelectric coefficient in the composition range between 0.01(1) and 14.6(2) wt% FeO. Thus, to a first approximation, tourmaline pyroelectric coefficients may be predicted directly from the chemical composition derived by routine electron probe microanalysis.

The relationships between pyroelectricity and chemistry indicate that the pyroelectric coefficient is influenced to different extents by the occupancies of the X, Y, and Z cation sites in the tourmaline structure. The octahedral Y site occupancy strongly influences the pyroelectric coefficient due to the preference of Fe for this site. This work further suggests that the addition of Fe and Mg cations to the smaller Z octahedral site causes the pyroelectric coefficient to increase. However, because an extended suite of samples is not available in which the Z site contains ions other than Al, this proposed trend has not been experimentally determined. The chemistry of the ninefold coordinated X site and the population of this site do not influence the pyroelectric coefficients of tourmaline.

INTRODUCTION

Tourmalines are naturally occurring minerals with the general formula $XY_3Z_6(BO_3)_3Si_6O_{18}(O,OH,F)_4$, where usually X = Na, Ca, or a vacancy, Y = Al, Li, Fe^{2+} , Fe^{3+} , and many other cations, and Z = Al, Mg, and Fe^{3+} . It has been known for centuries (Dietrich, 1985) that tourmaline crystals are pyroelectric materials that develop an electrostatic charge when heated or cooled. Careful measurements of the combined primary and secondary pyroelectric effects in these minerals were first reported by Ackermann (1915). The results of that work showed that tourmaline exhibits pyroelectricity over a wide temperature range. Tourmaline is stable to high temperatures and may find application as an infrared detector where ferroelectric materials such as $LaTiO_3$ fail (Hamid, 1980). Importantly, Ackermann's (1915) measurements show that differently colored tourmalines exhibit different degrees of pyroelectricity. For example, black varieties are the weakest whereas rose colored tourmalines are the most

strongly pyroelectric. Subsequent investigations support these observations and show that tourmaline pyroelectric coefficients (p_3) range between ~1.8 and 5.4 $\mu C/(m^2 \cdot K)$ at 296 K (Hayashi, 1912; Röntgen, 1914; Ackermann, 1915; Gladkii and Zheludev, 1965; Gavrilova, 1965; Fabell and Henish, 1971; Drozhdin et al., 1975; Gavrilova et al. 1989; the present work). Related studies have shown that piezoelectricity, the inducement of an electric charge by an applied stress, is also stronger in translucent colored tourmalines than in certain opaque black varieties (Cady, 1946).

None of these previous studies has addressed the variation in the pyroelectric coefficients with either the precise chemical composition or crystal chemistry of differently colored tourmalines. Tourmaline crystal chemistry is complex because the size and anion coordination of the X, Y, and Z cation sites are different. As a result, many different cations can be accommodated in tourmaline by coupled isomorphous substitution (e.g., Henry and Guidotti, 1985; Foit et al., 1989; Burt, 1989). An indication of this complexity is given by the summary of natural and synthetic tourmaline end-members as well as by the references to their structural descriptions in Table 1.

* Present address: CSIRO Division of Minerals, P.O. Box 124, Port Melbourne, Victoria 3207, Australia.

** Present address: Bahnhofstraße 10a, 86551 Aichach, Germany.

TABLE 1. Tourmaline end-member compositions

End-member	X	Y	Z	First description	Refined structure
Natural end-members					
Buergerite	Na	Fe ₃ ⁺	Al ₆	Mason et al. (1964)	Barton (1969)
Chromdravite	Na	Mg ₃	(Cr,Al)	Rumantseva (1983)	Nuber and Schmetzer (1979)
Dravite	Na	Mg ₃	Al ₆	Tschermak (1885)	Buerger et al. (1962)
Elbaite	Na	(Al,Li) ₃	Al ₆	Vernadsky (1913)	Donnay and Barton (1972)
Feruvite	Ca	Fe ₃ ⁺	(Al,Mg) ₆	Grice and Robinson (1989)	n.d.
Liddicoatite	Ca	(Al,Li) ₃	Al ₆	Dunn et al. (1977)	Nuber and Schmetzer (1981)
Povondraite*	Na	Fe ₃ ⁺	Fe ₃ ⁺	Walenta and Dunn (1979)	Grice et al. (1993)
Schorl	Na	Fe ₃ ⁺	Al ₆	Mathesius (1564)	Fortier and Donnay (1975)
Tsilaisite**	Na	(Mn,Al) ₃	Al ₆	Schmetzer and Bank (1984)	Nuber and Schmetzer (1984)
Uvite	Ca	Mg ₃	(Al ₆ Mg)	Dunn (1977)	Schmetzer et al. (1979)
Synthetic end-members					
Alkali-defect dravite "Sodium aluminum" tourmaline†	— Na	(Mg ₂ Al) Al ₆	Al ₆ Al ₆	Rosenberg and Foit (1979) Rosenberg et al. (1986)	Rosenberg and Foit (1979) Rosenberg et al. (1986)

Note: n.d. = not determined.

* Povondraite was initially termed "ferridravite" (Walenta and Dunn, 1979) with Y₃ = Mg₃ before the formula was altered to Y₃ = Fe₃⁺ following crystal structure refinement (Grice et al., 1993).

** Tsilaisite = manganian elbaite.

† Also called aluminobuergerite (Foit, 1975) and olenite (Burt, 1989).

Primary pyroelectricity describes the development of an electric voltage at the poles of certain noncentrosymmetric crystals, such as tourmaline (space group $R3m$), as temperature is changed. A permanent electric dipole or spontaneous polarization is inherent along the $\langle 00.l \rangle$ axis in tourmaline. As temperature is varied, the charge distribution in the structure shifts to produce a voltage along this axis. This voltage dissipates as atmospheric molecules are adsorbed onto the surface, so the crystal soon reverts to electrical neutrality.

The primary pyroelectric coefficient is a vector property, isolated when the external electric field, applied stress, and applied strain on a crystal are constant or zero (Nye, 1957). However, thermal expansion in a crystal held under such conditions establishes a strain field. Thus, a component of the measured pyroelectric coefficient is caused by the piezoelectric effect. This component, known as secondary pyroelectricity, is important for this work, as it produces between 75 and 90% of the observed pyroelectric effect in tourmaline (Zheludev, 1971; Bhalla and Cross, 1981; Lang, 1974). It is impractical to prevent thermal expansion of a crystal by mechanical clamping. Thus, pyroelectric coefficients of tourmaline are measured parallel to the $\langle 00.l \rangle$ directions and the d_{33} piezoelectric tensor from unclamped crystals at constant stress (σ). The measured pyroelectric coefficient (p_3^g) is the sum of these primary and secondary coefficients.

A nonuniform strain field caused by a temperature gradient produces tertiary pyroelectricity (Nye, 1957). This effect produces erratic data. Therefore, tertiary pyroelectricity must be avoided by maintaining uniform heat distribution in the material during experimental determination of pyroelectric coefficients.

For this study, relationships between pyroelectricity and the chemistry of tourmaline are investigated for selected samples from ~ 170 to ~ 500 K. From these data, the effects of chemical composition on the p_3^g variation in

tourmalines are determined. These data complement previous studies, which correlated the crystal chemistry of tourmalines with other material properties such as color and refractive indices (e.g., Rossman et al., 1991; Deer et al., 1992) and magnetic and mechanical properties (e.g., Donnay et al., 1967; Tsang et al., 1971; Dobrovol'skaya and Kuz'min, 1975; Helme and King, 1978; Tatli, 1980, 1985; Tatli and Özkan, 1987).

EXPERIMENTAL METHODS

Preparation of single-crystal plates

Pyroelectric coefficients were obtained from single-crystal tourmaline plates with normals parallel to the $\langle 00.l \rangle$ crystallographic directions. Crystals detailed in Table 2 were aligned by back-reflection Laué diffractographs. The diffractographs showed the error in alignment, $\theta < 0.5^\circ$. This tolerance produces a 0.004% error between the measured pyroelectric coefficient (p_3^g) and the true coefficient (p_3^g) of the plate, by the following equation:

$$\frac{p_3^g - p_3^g}{p_3^g} = (1 - \cos \theta). \quad (1)$$

This equation is derived from Figure 1, which schematically illustrates the relationship between coefficient vectors and a plate surface. Plates ~ 300 and $900 \mu\text{m}$ thick were sliced from the crystal and then polished with flatness and parallelism within a $\pm 10\text{-}\mu\text{m}$ tolerance (Hawkins, 1993).

Translucent polished plates were examined with the petrographic microscope to select crystal volumes suitable for pyroelectric measurement. Material with cracks or color zoning was rejected. Regions with uniaxial conoscopic images were cut from these larger plates to form test plates for pyroelectric measurements, since biaxial fringes are evidence for internal strain or chemical zoning

TABLE 2. Tourmaline mineral samples

Sample	Color and ((00.l) axis length, basal dia) (cm)	Provenance	Museum catalogue no.
S1b	dark green (3.5, 0.5)	Maine, U.S.A.	S.K. Dobos
S3a	black (3.0, 0.9)	Harts Range, Australia	S.K. Dobos
S8	green-blue (1.0, 0.7)	Minas Gerais, Brazil	Daddow's Rock Shop
S19	black (0.5, 2.0)	Ratnapura, Sri Lanka	Smithsonian B14657
S21	black (3.8, 1.4)	Pakistan	QGRS
S32	dark green (2.5, 1.0)	Brazil	Harvard H#91844
S33a	black (1.2, 0.9)	Rollstone Hill, U.S.A.	Harvard (no cat. no.)
S35	dark blue (1.5, 0.7)	Hindu Kush, Afghanistan	AMNH 41859
S36	pale aqua (1.5, 1.3)	Plumbago Mine, U.S.A.	AMNH 40730
S37	dark blue (1.5, 1.0)	Usakos, Namibia	AMNH 24116
S40	mauve (1.0, 2.5)	Plumbago Mine, U.S.A.	AMNH 40732
S41	black (1.0, 3.5)	Tourmaline Queen, U.S.A.	AMNH 46244
S42	dark olive (0.5, 5.0)	Madagascar	Gemological Inst
S43	pale green (2.0, 0.5)	Himalaya Mine, U.S.A.	Gemological Inst
S49	black (3.0, 1.0)	Black Jack, Australia	Australian D14988
S50	black (5.1, 1.5)	unknown	D.J. Henry

Note: tourmaline samples were loaned and donated from the following sources: S.K. Dobos, Department of Earth Sciences, University of Queensland, Brisbane, Australia; Daddow's Rock Shop, Stanley Street, Woolloongabba, Brisbane, Queensland, Australia; Quality Gem Rough Supplies (QGRS), P.O. Box 129, Kyneton, Victoria, Australia; D.J. Henry, Department of Geology and Geophysics, Louisiana State University, Baton Rouge, Louisiana 70803-4101, U.S.A.; Harvard University Mineralogical Museum, 24 Oxford Street, Cambridge, Massachusetts 02138, U.S.A.; American Museum of Natural History (AMNH), Central Park West at Seventy-ninth Street, New York, New York 10024-5192, U.S.A.; National Museum of Natural History, Smithsonian Institution, Washington, DC 20560, U.S.A.; Gemological Institute, 1660 Stewart Street, Santa Monica, California 90404, U.S.A.; and the Australian Museum, 6-8 College Street, Sydney, New South Wales 2000, Australia.

(Foord and Mills, 1978). These flaws may produce tertiary pyroelectric effects. For plates containing no volumes of uniaxial structure, volumes of low biaxial distortion were selected. Test plates were cut from the opaque schorl without petrographic examination. These schorl samples (S3a, S21, and S49) have highly mosaic structures, as shown by neutron-rocking curves across individual Bragg reflections (Hawkins, 1993).

Surface areas of opaque and densely colored plates were measured from images captured with a Panasonic WV-BL600 video-rate CCD camera fitted with a Canon 50 mm macro lens and extender tube. The captured images were analyzed with the MD30-plus Image Analysis System on an IBM-PC 386. Conversely, the S36 plate did not provide sufficient contrast for image analysis. The surface area for this plate was obtained from its density measured by heavy liquids and its thickness and mass.

Pyroelectric coefficient measurement

Pyroelectric coefficients were measured by an electronic circuit (Schneeberger, 1992) built after Byer and Roundy (1972). A Au film electrode was evaporated onto both faces of each plate of surface area A and then mounted on a heating element and cooled to ~ 170 K. The temperature was increased at a constant rate (dT/dt) between ± 2 and ± 6 K/min to 500 K and then decreased at a constant rate. The varying pyroelectric current (I) measured at 100 points during each increasing and decreasing temperature ramp was used to calculate the pyroelectric coefficient at each temperature, $p_3^s(T)$, by the following equation (Byer and Roundy, 1972):

$$p_3^s(T) = \frac{I}{A(dT/dt)}. \quad (2)$$

The measurements were repeated for each plate with dif-

ferent heating rates. At least two plates from each tourmaline specimen were measured to check for reproducibility.

Electron probe microanalysis and structural formulae

The tourmaline plates were coated with a 200 Å C film. Chemical analyses were collected using a JEOL JXA-8800L automated electron microprobe with four wavelength dispersive spectrometers. Thirteen elements were calibrated against the following primary standards: SiO₂, spessartine garnet; TiO₂, TiO₂; Al₂O₃, spessartine garnet; Cr₂O₃, chromite; FeO, hematite; MnO, spessartine garnet; NiO, synthetic Ni₂SO₄; ZnO, ZnS; MgO, MgO; CaO, wollastonite; Na₂O, albite; K₂O, orthoclase; F, MgF₂. Secondary standards were analyzed prior to each session.

Line scans were made approximately normal to each other across each plate with a 15-kV, 15-nA beam. Analyses were taken at 100 or 200 μm intervals. These intervals are broad compared with the fine, 1–3 μm zonation features in some tourmalines (e.g., Grice and Robinson, 1989; Cavarretta and Puxeddu, 1990; Henry and Du-

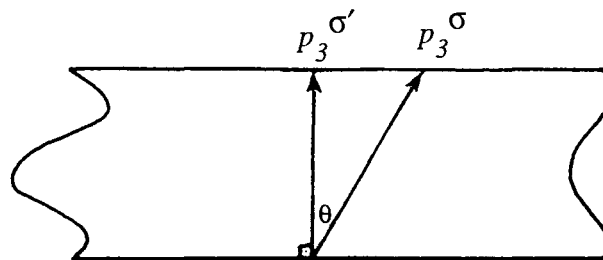


Fig. 1. Geometric relationship between the true (p_3^σ) and measured ($p_3^{\sigma'}$) pyroelectric coefficients and the surface of a crystal plate.

TABLE 3. Average compositions and structural formulae of the pyroelectric test plates

Sample	S1b	S3a	S8	S19	S21	S32	S33a	S35
SiO ₂	37.32(50)	35.04(41)	37.04(42)	36.12(58)	35.16(44)	37.04(55)	35.12(47)	37.74(84)
TiO ₂	0.18(5)	0.63(7)	0.01(2)	1.18(20)	0.60(13)	0.27(4)	0.80(21)	0.02(2)
Al ₂ O ₃	36.32(34)	31.06(26)	37.47(29)	27.13(35)	32.67(16)	35.74(29)	32.88(60)	36.61(58)
Cr ₂ O ₃	0.01(2)	0.01(2)	0.01(2)	0.08(4)	0.01(1)	0.01(1)	0.01(1)	0.01(1)
FeO	3.59(24)	14.29(15)	2.40(7)	0.83(12)	11.07(70)	5.98(13)	11.41(19)	6.74(16)
MnO	0.98(8)	0.59(4)	1.41(5)	0.01(2)	0.45(3)	0.34(4)	0.15(2)	0.52(4)
NiO	0.01(2)	0.02(3)	0.01(2)	0.01(3)	0.01(2)	0.01(1)	0.01(2)	0.01(1)
ZnO	0.04(4)	0.08(5)	1.88(7)	0.03(4)	0.41(7)	0.05(5)	0.07(4)	0.09(4)
MgO	0.19(3)	1.98(12)	0.00(1)	13.76(30)	2.23(56)	1.10(6)	3.30(27)	0.10(1)
CaO	2.11(3)	0.46(5)	0.04(2)	3.94(50)	0.11(2)	0.11(2)	0.52(16)	0.12(2)
Na ₂ O	1.73(14)	2.28(7)	2.08(11)	0.85(8)	2.04(8)	2.58(8)	1.98(12)	2.69(11)
K ₂ O	0.02(1)	0.07(1)	0.01(1)	0.01(1)	0.04(1)	0.03(1)	0.05(1)	0.03(1)
F	1.29(29)	0.59(5)	0.51(15)	0.70(10)	0.44(13)	1.33(24)	0.46(14)	1.15(8)
-F = O	-0.54	-0.25	-0.21	-0.29	-0.19	-0.21	-0.19	-0.48
Total	83.25(70)	86.86(58)	82.64(65)	84.34(96)	85.31(77)	84.4(1.1)	86.58(88)	85.3(1.3)
Ions per formula unit								
Si	6.00	6.00	6.00	6.00	6.00	6.00	6.00	6.00
B	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00
Na	0.54	0.76	0.65	0.27	0.67	0.81	0.66	0.83
Ca	0.36	0.08	0.01	0.70	0.02	0.02	0.10	0.02
K	0.00	0.02	0.00	0.00	0.01	0.01	0.01	0.01
Σ X site	0.90	0.86	0.66	0.97	0.70	0.84	0.77	0.85
Ti	0.02	0.08	0.00	0.15	0.08	0.03	0.10	0.00
Al	0.88	0.37	1.15	0.00	0.57	0.82	0.62	0.86
Cr	0.00	0.00	0.00	0.01	0.00	0.00	0.00	0.00
Fe	0.48	1.95	0.33	0.12	1.58	0.81	1.63	0.90
Mn	0.13	0.09	0.19	0.00	0.07	0.05	0.02	0.07
Ni	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00
Zn	0.00	0.01	0.22	0.00	0.05	0.01	0.01	0.01
Mg	0.05	0.51	0.00	2.72	0.57	0.27	0.84	0.02
Li*	(1.44)	—	(1.11)	—	(0.08)	(1.01)	—	(1.14)
Σ Y site	1.56	3.01	1.89	3.00	2.92	1.99	3.22	1.86
Al	6.00	5.90	6.00	5.31	6.00	6.00	6.00	6.00
Mg	0.00	0.00	0.00	0.69	0.00	0.00	0.00	0.00
Fe	0.00	0.10	0.00	0.00	0.00	0.00	0.00	0.00
Σ Z site	6.00	6.00	6.00	6.00	6.00	6.00	6.00	6.00
F	0.66	0.32	0.26	0.37	0.24	0.68	0.25	0.58
OH	3.11	3.48	3.45	3.38	3.62	3.32	3.50	3.50

Note: n.a. = not analyzed. Values in parentheses are 1 esd of the microprobe traverse analyses.

* Li ions pfu are the difference from full Y site occupancy.

trow, 1992). The intervals used in this work allowed each plate to be investigated within the constraints of instrument time, yet gave sufficient data to obtain an average composition with estimated standard deviations (esd) representative of chemical variation across the plates.

Structural formulae were normalized to six Si atoms per formula unit. The assumptions for cation site assignments are similar to those summarized by Rosenberg and Foit (1979). B is assumed to be present in stoichiometric amounts. Na, Ca, and K are assigned to the ^[9]X site. Multivalent cations are assumed to be in the oxidation states Fe²⁺, Mn²⁺, and Ti⁴⁺. Al is assigned to the smaller ^[6]Z site and excess Al is assigned to the larger ^[6]Y site. All remaining cations are assigned to the Y site. If there is a deficiency of Al, the smaller cations such as Mg and excess Fe²⁺ (in lieu of Fe³⁺) are assigned to the Z site, using crystal ion radii (Shannon, 1976) as a guide. No attempt is made to partition the same ion between both octahedral sites. An exception is the S3a schorl in which Fe is partitioned between the Y and Z sites, using the ratio derived from Mössbauer studies and refined during single-crystal X-ray structure determination (Hawkins, 1993).

Some Y site vacancies are present when these criteria are followed. However, structural refinement has shown that Y site vacancies are unlikely in tourmaline (Donnay and Barton, 1972; Foit and Rosenberg, 1977; Rosenberg and Foit, 1979; Hawkins, 1993). Therefore, Y site vacancies are filled by Li if the mineral has a lithia-rich provenance, i.e., if it contains elbaite or liddicoatite components. The number of H ions is estimated from bond valence sums (Brese and O'Keeffe, 1991).

RESULTS

Test plate quality

Chemical variation within the tourmaline test plates may affect their suitability for electrical measurements. Thus, two statistical tests were applied to the sets of traverse analyses to assess (1) whether there is chemical zonation within the plates and (2) whether two or more plates from the same crystal have the same average chemistry.

First, the variation in concentrations along a traverse was compared with the counting statistics for each element. For this study, microprobe traverses do not show zoning at a 95% confidence level if the relative variation

TABLE 3.—Continued

Sample	S36	S37	S40	S41	S42	S43	S49	S50
SiO ₂	38.83(26)	37.68(33)	39.45(76)	34.7(1.0)	37.87(61)	38.18(15)	35.39(80)	36.91(48)
TiO ₂	0.01(1)	0.01(2)	0.01(1)	0.03(4)	0.07(4)	0.03(2)	0.41(23)	0.02(2)
Al ₂ O ₃	38.32(23)	36.98(38)	40.39(26)	35.77(25)	38.36(63)	38.61(24)	33.2(1.0)	35.49(56)
Cr ₂ O ₃	0.01(1)	0.01(2)	0.01(2)	0.01(2)	0.01(2)	0.01(1)	0.01(2)	0.01(3)
FeO	2.13(21)	4.84(22)	0.01(1)	8.84(18)	2.29(14)	0.79(6)	14.04(94)	8.32(22)
MnO	0.57(5)	0.80(5)	0.18(4)	1.72(8)	0.34(6)	0.77(4)	0.16(5)	0.31(4)
NiO	0.01(1)	0.01(2)	0.01(2)	0.01(3)	0.01(3)	0.01(2)	0.01(2)	0.01(2)
ZnO	0.13(3)	0.27(10)	0.02(4)	0.89(15)	0.06(7)	0.05(4)	0.07(8)	0.20(6)
MgO	0.02(1)	0.03(2)	0.00(1)	0.05(2)	0.03(2)	0.01(1)	0.63(11)	0.11(2)
CaO	0.52(8)	0.32(3)	0.49(5)	0.03(2)	0.44(13)	1.48(8)	0.10(6)	0.06(2)
Na ₂ O	2.18(5)	2.59(7)	1.65(11)	1.93(9)	2.13(10)	1.69(4)	1.77(20)	2.66(9)
K ₂ O	0.02(1)	0.02(1)	0.01(1)	0.03(2)	0.02(2)	0.01(1)	0.04(2)	0.03(1)
F	0.92(6)	1.13(7)	n.a.	0.49(16)	0.87(24)	1.10(6)	0.46(20)	1.60(28)
-F = O	-0.39	-0.47	—	-0.21	-0.37	-0.46	-0.43	-0.67
Total	83.29(47)	84.21(60)	82.2(1.2)	84.3(1.0)	82.1(1.0)	82.27(32)	86.1(1.4)	85.05(73)
Ions per formula unit								
Si	6.00	6.00	6.00	6.00	6.00	6.00	6.00	6.00
B	3.00	3.00	3.00	3.00	3.00	3.00	3.00	3.00
Na	0.65	0.80	0.49	0.65	0.65	0.51	0.58	0.84
Ca	0.09	0.05	0.08	0.01	0.07	0.25	0.02	0.01
K	0.00	0.00	0.00	0.01	0.00	0.00	0.01	0.01
Σ X site	0.74	0.85	0.57	0.67	0.72	0.76	0.61	0.86
Ti	0.00	0.00	0.00	0.00	0.01	0.00	0.05	0.00
Al	0.98	0.94	1.24	1.28	1.16	1.15	0.64	0.80
Cr	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00
Fe	0.28	0.64	0.00	1.28	0.30	0.10	1.99	1.13
Mn	0.07	0.11	0.02	0.25	0.05	0.10	0.02	0.04
Ni	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00
Zn	0.01	0.03	0.00	0.11	0.01	0.01	0.01	0.02
Mg	0.00	0.01	0.00	0.01	0.01	0.00	0.16	0.03
Li*	(1.66)	(1.28)	(1.74)	(0.08)	(1.46)	(1.64)	(0.13)	(0.98)
Σ Y site	1.34	1.72	1.26	2.92	1.54	1.36	2.87	2.02
Al	6.00	6.00	6.00	6.00	6.00	6.00	6.00	6.00
Mg	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00
Fe	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00
Σ Z site	6.00	6.00	6.00	6.00	6.00	6.00	6.00	6.00
F	0.45	0.57	n.a.	0.27	0.44	0.55	0.25	0.82
OH	3.40	3.53	3.88-F	3.62	3.38	3.31	3.62	3.22

in concentration for each element does not exceed twice the relative variation in counting statistics. By this criterion, all plates except S40 and S43 showed zoning in some elements.

Second, differences between the average chemical compositions of plates cut from the same tourmaline specimen were determined for each element within a 95% confidence interval by a Student *t*-test (Walpole and Myers, 1978). In most cases, the compositions of all elements were consistent between plates. Thus, the analyses shown in Table 3 are the averages of microprobe traverses within these confidence limits for each sample.

Pyroelectricity vs. temperature curves

In this temperature range, the pyroelectric response for tourmaline is a convex curve, as illustrated in Figure 2. Curves for the repeated temperature cycles and for several plates from the same crystal were fitted with a quadratic function. The error in p_3 shown in Table 4 is calculated from the spread of the data from the fitted function (Schneeberger, 1992). In most cases, plates from the same specimen gave the same curve within these error margins. Thus, the dimensional tolerances, chemical zonation, and flaws in these tourmaline plates were ac-

ceptable for reproducible measurement of p_3 . Repeat data collected from the same crystal were added to form the quadratic functions for p_3 vs. T given in Table 4. The S1b and S41 samples are exceptions to the reproducibility generally observed within samples. The two S1b samples gave consistently different intersecting curves, as shown by the equations in Table 4. An explanation for these differences is not obvious because both plates are of similar quality. Thus, S1b results are omitted from further analysis. Similarly, one S41 plate was omitted from the average curve for this sample because it produced pyroelectric coefficients that were consistently ~10% higher than those for three other S41 plates. This difference may be attributed to inaccurate surface-area measurement of one plate.

The low-temperature form of the pyroelectric response curve is shown in Figure 2. The lowest temperature obtained in this work was limited by the liquid N₂ coolant, and so the polarity inversion in tourmaline p_3 at <20 K (e.g., Drozhdin et al., 1975; Gavrilova et al., 1989) was not detected.

The high-temperature region of the curve adopts one of two forms. First, there is an upward inflection of the curve shown in Figure 2. This behavior has been noted

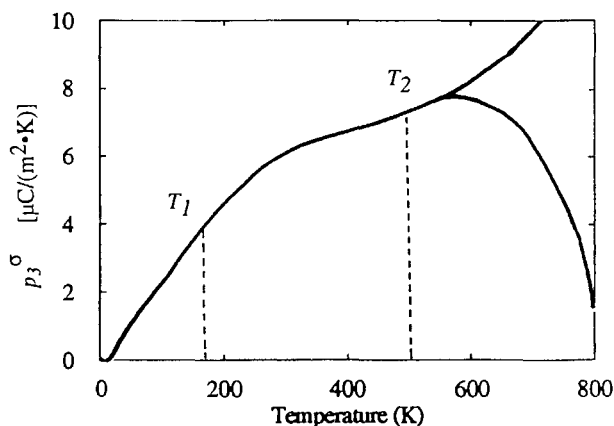


Fig. 2. The general form of the pyroelectric response with temperature change for tourmaline. The approximate temperature range for the present measurements is between T_1 and T_2 . A low-temperature polarity inversion is shown at <20 K (after Gavrilova et al., 1989).

for only a pale rose sample above 450 K (Ackermann, 1915; Gavrilova et al., 1983, 1989). However, this type of behavior was observed during one experiment of the S1b/5 plate above ~ 420 K and for both S49 plates above ~ 400 K. A similar inflection has been predicted for the thermal expansion vs. temperature relationship in other minerals. For example, strong anharmonic thermal vibration above 1300 K in forsterite produces an upward inflection of the calculated thermal expansion coefficient curve (Reynard et al., 1992). Thermal expansion causes the secondary pyroelectric effect, which is a strong effect in tourmaline. Thus, this region of the curve in Figure 2 may reflect the effects of anharmonic vibration on pyroelectricity. These results indicate that the onset of pyroelectricity that is caused by anharmonic vibration of ions in the structure and high thermal expansion may occur at a lower temperature in S1b/5 and S49 than in other tourmalines.

Second, a smooth decrease in the pyroelectric coefficient was observed in the schorl S33a plate above ~ 430 K. This behavior, which is probably caused by the onset of electrical conductivity, has not been previously reported in tourmaline. Such conductivity is different from the electric current generated in the external circuit from the pyroelectric voltage. This tourmaline is omitted from further analysis because it produced an anomalously low p_3^s . This problem arose from a difficulty in the accurate determination of surface area because of chipped edges.

Further upper temperature limits were encountered with S21, S32, and S41 plates, as shown in Table 4. The signals from these plates contained high levels of electronic noise at higher temperatures.

Pyroelectric coefficient variation with bulk chemistry

Clearly, tourmaline bulk chemistry determines the pyroelectric response of this mineral. To investigate trends between the pyroelectric coefficients and chemistry, pyroelectric coefficients were calculated from equations for 14 samples in Table 4 at two temperatures. Low-temperature behavior is represented by data at 193 K. Although most samples were measured to ~ 500 K, some plates gave erratic responses above ~ 400 K. Therefore, higher temperatures are represented by data at 383 K, so all 14 samples could be included in the data set. The variations in tourmaline pyroelectric coefficients with bulk chemistry are shown by the graphs in Figure 3.

FeO. The linear decrease in pyroelectric coefficient with increase in Fe content (as wt% FeO), shown in Figure 3a, is the most important result. The data at 193 and 383 K are strongly linear, as the linear least-squares correlation coefficient r is ~ 0.9 in both cases. This high correlation is statistically significant, since the random probability for 14 data points with a correlation coefficient between 0.85 and 0.90 is $<0.1\%$ (Taylor, 1982). The higher temperature data are more strongly linear, and the slope of this line is higher than for the 193 K data. Thus, an increase in Fe content has a stronger influence on tourmaline py-

TABLE 4. Empirical equations for the temperature dependence of p_3^s

Sample	p_3^s [$\mu\text{C}/(\text{m}^2 \cdot \text{K})$]	Error	T range (K)
S1b/5*	$-0.2379 + 1.732 \times 10^{-2}T - 9.54 \times 10^{-6}T^2$	0.10	170-423
S1b/6*	$4.379 \times 10^{-2} + 1.178 \times 10^{-2}T - 8.133 \times 10^{-6}T^2$	0.10	170-500
S3a	$-0.3686 + 1.112 \times 10^{-2}T - 4.259 \times 10^{-6}T^2$	0.17	170-500
S8	$-0.979 + 2.194 \times 10^{-2}T - 1.873 \times 10^{-5}T^2$	0.07	170-500
S19	$-0.8659 + 1.97 \times 10^{-2}T - 1.652 \times 10^{-5}T^2$	0.20	170-500
S21	$-2.661 + 2.849 \times 10^{-2}T - 3.170 \times 10^{-5}T^2$	0.15	170-463
S32	$-1.588 + 2.257 \times 10^{-2}T - 1.976 \times 10^{-5}T^2$	0.22	170-483
S33a*	$-0.6975 + 1.056 \times 10^{-2}T - 6.903 \times 10^{-6}T^2$	0.15	170-433
S35	$-1.88 + 2.308 \times 10^{-2}T - 1.976 \times 10^{-5}T^2$	0.07	170-500
S36	$-1.19 + 2.33 \times 10^{-2}T - 1.987 \times 10^{-5}T^2$	0.11	170-500
S37	$-1.310 + 2.024 \times 10^{-2}T - 1.578 \times 10^{-5}T^2$	0.10	170-500
S40	$-0.4096 + 1.816 \times 10^{-2}T - 1.226 \times 10^{-5}T^2$	0.10	170-500
S41	$-0.8918 + 1.950 \times 10^{-2}T - 1.620 \times 10^{-5}T^2$	0.16	170-443
S42	$-0.9894 + 1.979 \times 10^{-2}T - 1.465 \times 10^{-5}T^2$	0.07	170-500
S43	$-0.7167 + 2.246 \times 10^{-2}T - 1.883 \times 10^{-5}T^2$	0.14	170-500
S49	$-0.9472 + 1.500 \times 10^{-2}T - 9.123 \times 10^{-6}T^2$	0.10	170-403
S50	$-1.585 + 1.954 \times 10^{-2}T - 1.566 \times 10^{-5}T^2$	0.06	170-500

* Pyroelectric equations from these samples are not included in the analysis of p_3^s variation with chemistry.

roelectricity at higher temperature. This effect could be caused by an increase in electrical conduction of Fe-rich tourmaline at elevated temperatures and is consistent with the proposed onset of electrical conductivity in schorl. The p_3^s may approach a plateau for the Fe-rich tourmalines S3a, S21, and S49, as indicated by the term 'schorl' in Figure 3a. This small invariant region is more evident at low temperature.

SiO₂. The p_3^s variation with SiO₂ content is linear, as shown in Figure 3b. However, this variation results from the heavy element content such as Fe, since there is only minor substitution for Si in tourmaline (e.g., Walenta and Dunn, 1979; Henry and Guidotti, 1985; Gallagher, 1988). An inverse proportionality between weight percent SiO₂ and weight percent FeO is seen by comparison of Figure 3a and 3b.

MnO₂. A weak linear relationship between p_3^s and weight percent MnO₂ is shown in Figure 3c.

Al₂O₃. There are two regions denoted by 1 and 2 in circles in Figure 3d. In the Al-rich region (1), p_3^s increases with Al₂O₃ concentration. This region is detailed in Figure 3e. A minimum p_3^s value corresponding to the schorl samples S3a, S21, and S49 identified in the FeO relationship in Figure 3a occurs around 31–33 wt% Al₂O₃. The data in this minimum region fit the linear relationship shown in Figure 3e without degrading the correlation coefficient. Therefore, the data points at this minimum are included in region 1. However, r may appear artificially high because the data set is small. With further depletion of Al from tourmaline, the pyroelectric coefficient increases again (Fig. 3d, region 2). The data point shown near 27 wt% Al₂O₃ represents the S19 uvite-dravite, supporting the data in Figure 3f and indicating that incorporation of large amounts of Mg in the tourmaline lattice increases p_3^s .

MgO. There is a discontinuity in data between ~3 wt% MgO and the S19 uvite-dravite at 14 wt% MgO in Figure 3f. At MgO concentrations below the sharp inflection around 0.10–0.50 wt% MgO, the relationship between p_3^s and weight percent MgO is strongly linear, despite the large composition error bars (Fig. 3g). However, a paucity of suitable dravite or uvite tourmalines prevented detailed investigation of the effect of Mg incorporation on the pyroelectric coefficients.

ZnO, Na₂O, CaO, and F. Zn, F, and the X-site cations do not significantly affect the pyroelectric coefficients. ZnO content vs. p_3^s displays a very weak inflection around 0.1–0.5 wt% ZnO in a trend similar to that of MgO vs. p_3^s . There is a weak correlation between Na₂O and p_3^s , as shown in Figure 3h.

DISCUSSION

Pyroelectric coefficients of Fe-containing tourmalines

These data form a basis for the prediction of tourmaline pyroelectric coefficients directly from electron microprobe FeO analyses. All tourmaline compositions studied conform to the linear relationship between weight percent FeO and p_3^s shown in Figure 3a. Fe-rich tourmalines are

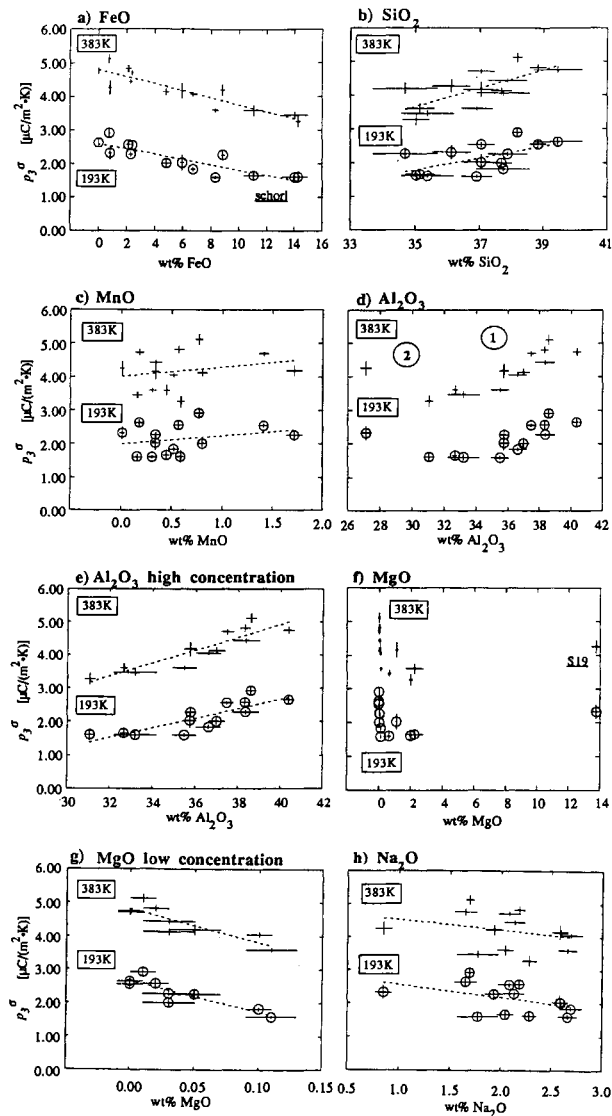


Fig. 3. Relationships between p_3^s and the bulk chemistry of tourmaline. Error bars are ± 1 esd for chemical composition (Table 3) and p_3^s (Table 4). Equations have the form $p_3^s = A + Bx$, where x is the weight percent oxide; and r is the correlation coefficient of the linear least-squares fit of the line to the data. (a) FeO 193 K: $A = 2.59$, $B = -0.08$, $r = -0.86$; 383 K: $A = 4.79$, $B = -0.10$, $r = -0.90$. (b) SiO₂ 193 K: $A = -4.51$, $B = 0.18$, $r = 0.61$; 383 K: $A = -5.91$, $B = 0.27$, $r = 0.73$. (c) MnO 193 K: $A = 1.99$, $B = 0.25$, $r = 0.27$; 383 K: $A = 4.02$, $B = 0.28$, $r = 0.24$. (d) Al₂O₃. (e) Al₂O₃ high concentration 193 K: $A = -2.99$, $B = 0.14$, $r = 0.83$; 383 K: $A = -2.94$, $B = 0.20$, $r = 0.90$. (f) MgO. (g) MgO low concentration 193 K: $A = 2.65$, $B = -9.14$, $r = -0.88$; 383 K: $A = 4.81$, $B = -9.92$, $r = -0.86$. (h) Na₂O 193 K: $A = 2.94$, $B = -0.39$, $r = -0.44$; 383 K: $A = 4.84$, $B = -0.31$, $r = -0.28$.

often shown in this study to be black or darkly colored. Thus, this relationship supports the general observation (e.g., Dietrich, 1985) that black tourmalines exhibit weaker pyroelectric behavior than pale colored tourmalines. However, the S19 uvite-dravite is Fe-poor and has a cor-

TABLE 5. Predicted pyroelectric coefficients p_3^s [$\mu\text{C}/(\text{m}^2 \cdot \text{K})$] and errors for selected tourmaline end-member compositions

End-member	p_3^s at 193 K (Fig. 4)	p_3^s at 193 K (Eq. 3)	p_3^s at 296 K (Eq. 4)	p_3^s at 383 K (Eq. 5)
Tourmaline (FeO = 0%)	—	2.59(0.50)	3.95(0.51)	4.79(0.52)
Schorl (FeO = 20.46%)	—	1.02(0.56)	2.03(0.57)	2.68(0.58)
Fe-free tourmalines				
Uvite	2.45	—	—	—
Elbaite	3.08	—	—	—
Aluminobuergerite	5.75	—	—	—

respondingly high p_3^s value, despite the dark brown or black color. Therefore, these results confirm that color, which has been used previously to characterize tourmalines, is not as reliable as chemistry for the prediction of the pyroelectric coefficients.

Predictive equations between the p_3^s and FeO content at three temperatures were devised by simple regression analysis, with weight percent oxide as the independent variable and p_3^s as the dependent variable. These equations include the following:

$$\text{at 193 K} \quad p_3^s = 2.594 - 0.077(\text{wt\% FeO})$$

$$\mu\text{C}/(\text{m}^2 \cdot \text{K}) \quad (3)$$

$$\text{at 296 K} \quad p_3^s = 3.949 - 0.094(\text{wt\% FeO})$$

$$\mu\text{C}/(\text{m}^2 \cdot \text{K}) \quad (4)$$

$$\text{at 383 K} \quad p_3^s = 4.791 - 0.103(\text{wt\% FeO})$$

$$\mu\text{C}/(\text{m}^2 \cdot \text{K}) \quad (5)$$

Pyroelectric coefficients and errors calculated at 193, 296, and 383 K for an ideal schorl end-member containing 20.46% FeO (Dietrich, 1985) and a tourmaline with FeO = 0% are listed in Table 5. Pyroelectric coefficients predicted for tourmalines at 296 K are in good agreement with values for similar samples reported at ambient temperature. For example the p_3^s value of 2.03(57) $\mu\text{C}/(\text{m}^2 \cdot \text{K})$ at 296 K for the schorl agrees extremely well with the value of 2.2 $\mu\text{C}/(\text{m}^2 \cdot \text{K})$ for a black tourmaline (Gladkii and Zheludev, 1965). This black tourmaline is probably close to a schorl in composition. Further, the p_3^s of 4.4 $\mu\text{C}/(\text{m}^2 \cdot \text{K})$ at ambient temperature reported for pink tourmalines (Hayashi, 1912; Ackermann, 1915) is within the error of the value 3.95(51) $\mu\text{C}/(\text{m}^2 \cdot \text{K})$ at 296 K predicted for Fe-free tourmaline. Pink tourmaline is usually an elbaite with minor Mn substitution.

The errors in p_3^s predicted from Equations 3–5 are calculated as a 90% confidence interval based on a Student t -test (Walpole and Myers, 1978). These intervals are large percentage errors in p_3^s . Such errors may be reduced only by increasing the number of samples in the data set. However, it is very difficult to obtain tourmaline crystals of the quality required for this measurement technique. For

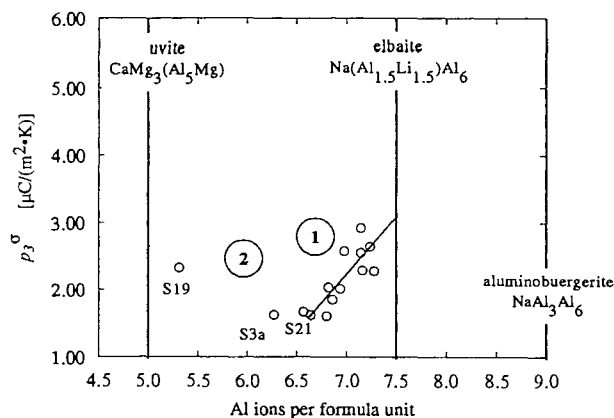


Fig. 4. The number of Al ions per tourmaline formula unit vs. p_3^s at 193 K.

example, the 14 samples used in this analysis represent only 27% of the tourmaline crystals collected for this work (Hawkins, 1993).

It is surprising that the correlation between weight percent FeO and p_3^s is so high, despite the assumption that Fe is divalent for all tourmaline and that the effects of Fe partitioning between the Y and Z sites have not been taken into account.

The strong effect of FeO concentration may obscure the effects of other ions on p_3^s . Thus, an investigation of the complex crystal chemical substitutions in tourmaline is used to show the dependence of the pyroelectric response on such ions.

Cation substitution effects on the pyroelectric coefficient

Y site. The relationship between the number of Al ions per formula unit (pfu) and p_3^s is shown in Figure 4. Above 6.5 Al ions in region 1, p_3^s increases with Fe removal by the schorl-elbaite substitution on the Y crystallographic site: $2\text{Fe}_Y^{2+} \rightleftharpoons \text{Li}_Y^+ + \text{Al}_Y^{3+}$. This substitution corresponds to the dependence of p_3^s on weight percent FeO. Therefore, the schorl-elbaite substitution is probably the dominant mechanism for Fe removal to produce high p_3^s coefficients in Al-rich tourmalines. In these tourmalines, Z is usually fully occupied by Al.

A similar linear increase in p_3^s with increased Mn suggests that Fe is also removed by the Y site cation substitution: $\text{Fe}_Y^{2+} \rightleftharpoons \text{Mn}_Y^{2+}$. However, this cation exchange is not important for all tourmaline compositions studied, since the amount of Mn incorporated by this substitution is small, <2 wt% MnO. Further, the relationship is weak, as the linear-fit correlation coefficients for data at 193 and 383 K are ~ 0.25 , as shown in Figure 3c. At high Mg concentrations the $\text{Fe}_Y^{2+} \rightleftharpoons \text{Mg}_Y^{2+}$ substitutions deplete Fe from the structure. The p_3^s value then increases, as shown in Figure 3f.

Around the minimum inflection of the weight percent Al_2O_3 vs. p_3^s relationship in Figure 4, Fe removal from the Y site and substitutions for Al on the Z site occur

The effect of Al content on p_3^o is probably a result of the substitution of Al for Y and Z site cations, particularly Fe. However, there may be properties of Al, Fe, and other ions that affect polarity in the structure and pyroelectric behavior.

Z and X sites. The regions in Figure 4 that are low and intermediate in Al content, shown as region 2, indicate that a further set of cation substitutions may affect the p_3^o values in these tourmalines. A line cannot be drawn from these data because there are insufficient data points. Thus, the data from samples S19, S3a, and S21 (labeled) are discussed individually. The low Al content of these tourmalines suggests that there are other cations on the Z site. For example, the S19 uvite-dravite structural formula (given in Table 3) is depleted in Al. Thus, 0.69 Mg ions and 5.31 Al ions are incorporated in the Z site by the uvite substitution $\text{Na}_x^+ + \text{Al}_z^{2+} = \text{Ca}_z^{2+} + \text{Mg}_z^{2+}$. Powder electron-spin resonance data collected at ~ 130 K (Hawkins, 1993) indicate that there is also minor Fe^{3+} on the Z site of S19. In addition, the S3a schorl has been shown by Mössbauer studies and population parameter refinement during least-squares structure analysis (Hawkins, 1993) to contain 0.10 Fe ions per six Z sites. This analysis suggests that a 13% substitution of Mg for Al and a 1.7% substitution of Fe for Al on the Z site in the S19 uvite-dravite and S3a schorl, respectively, may strongly increase the pyroelectric coefficient. Other ions, for example Mn (Johnston and Duncan, 1975), partition to the tourmaline Z site. Thus, there are many possible Z site substitutions for Al that may affect p_3^o .

Pyroelectric coefficients for Fe-free tourmaline

There are several Fe-free end-members shown in Table 1. Pyroelectric coefficients of the elbaite [$\text{XY}_3\text{Z}_6 = \text{Na}(\text{Al}_{1.5}\text{Li}_{1.5})\text{Al}_6$], uvite [$\text{CaMg}_3(\text{Al}_3\text{Mg})$], and aluminobuergerite (NaAl_3Al_6) end-members are shown by extrapolation in Figures 4 and 5. There are sufficient data for the schorl-elbaite substitution series in Figure 4 to extrapolate a value of p_3^o for elbaite, which is included in the summary of p_3^o values in Table 5. Similarly, the p_3^o for uvite is extrapolated in region 2 of Figure 5, although there are fewer data points than in region 1. It may be difficult to obtain samples in this composition range with further work because there is a miscibility gap reported in the Al-Mg (elbaite-dravite) phase field for natural tourmalines (e.g., Donnay and Barton, 1972; Taylor and Slack, 1984; Henry and Guidotti, 1985; Cavarretta and Puxeddu, 1990; Kassoli-Fournaraki, 1990).

The prediction of aluminobuergerite pyroelectric behavior from Figure 4 is justified only if the same linear trend continues from the schorl-elbaite region to higher Al compositions. It may be difficult to explore this relationship further using the present pyroelectric apparatus because large crystals are required. Aluminobuergerite is a synthetic end-member (Foit, 1975) produced when Al content on the Y site is increased by the liddicoatite, aluminobuergerite, and olenite substitutions (Henry and Guidotti, 1985; Burt, 1989). Only crystals <1 mm in size

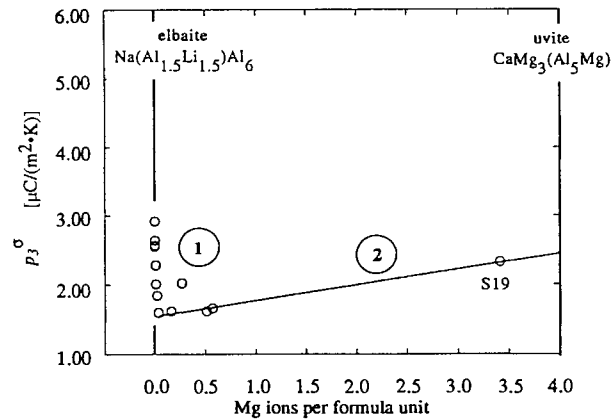


Fig. 5. The number of Mg ions per tourmaline formula unit vs. p_3^o at 193 K.

are produced by hydrothermal synthesis (e.g., Taylor and Terrell, 1967; Foit, 1975; Vorbach, 1989).

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